

# Enhancement of the field emission properties of low-temperature-growth multi-wall carbon nanotubes by KrF excimer laser irradiation post-treatment

Chia-Hung Li <sup>a,\*</sup>, Han-Chi Liu <sup>a</sup>, Shih-Chun Tseng <sup>c</sup>, Yi-Ping Lin <sup>b</sup>, Shih-Pu Chen <sup>b</sup>,  
Jung-Yu Li <sup>b</sup>, Kwang-Hsiung Wu <sup>a</sup>, Jenh-Yih Juang <sup>a</sup>

<sup>a</sup> Department of Electrophysics, National Chiao Tung University, Hsinchu, Taiwan

<sup>b</sup> Advanced Energy Technology Laboratory, ERL, ITRI, Chutung, Hsinchu, Taiwan

<sup>c</sup> Mechanical and Systems Research Laboratories, M500, MSL/ITRL, Chutung, Hsinchu, Taiwan

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## Abstract

Multi-wall carbon nanotube (CNT) films were fabricated by microwave plasma chemical vapor deposition at low temperatures ( $\sim 500$  °C). The films when properly post-treated by laser irradiation exhibited a factor of 2–3 enhancement in the emission current, while the turn-on field ( $E_{on}$ ) was reduced from 4.89–5.22 to 2.88–3.15 V/ $\mu\text{m}$ . The introduction of excessive oxygen during laser irradiation, however, degrades the performance of field emission properties drastically. Raman spectroscopy measurements revealed the intimate correlation between the parameter  $I_D/I_G$  (intensity ratio between the two representative Raman peaks seen in carbon nanotubes) and the field emission performance. The scanning electron microscopy (SEM) and transmission electron microscopy (TEM) analyses showed that the irradiation-induced modification of the tube morphology and crystallinity might be responsible for the observations.

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## 1. Introduction

Carbon nanotubes (CNTs) [1] are one-dimensional (1-D) nano-materials with excellent technological application potential, including field emission [2], microelectronic devices [3] and micro-gun [4], due to their high field emission current density, low turn-on field and stable currents. Colbert and Smalley [5] observed the enhancement of field emission of CNTs by using laser irradiation and they interpreted this effect as being due to the presence of localized plasma induced by instant vaporization of CNTs and ionization of the species. In this study, the microwave plasma chemical vapor deposition (MP-CVD) system was used to synthesize carbon nanotubes. We then used a KrF excimer laser to practice the post-treatment on the obtained CNTs. The effects of the post-treatment parameters such as laser power density, count number of the delivered laser pulses, and

precursor atmosphere on the field emission characteristics of the CNTs are discussed.

## 2. Experiments

For fabricating CNTs, we first prepared a fully cleaned p-(100) Si substrate, and then layers of 40 nm Ti and 20 nm Ni were deposited sequentially on the Si substrate using E-gun vapor deposition at a base pressure of  $10^{-6}$  Torr. The coated substrate was then immediately loaded into the MP-CVD chamber for hydrogen plasma pre-treatment. The pre-treatment was conducted by applying microwave with the power of 800 W to the chamber with a 90 sccm flowing hydrogen gas (corresponding to a background pressure of 24 Torr) while keeping the substrate temperature at 500 °C for 15 min. Finally, a continuous process was practiced for growing the multi-wall carbon nanotubes (MWNTs) by raising the microwave power to 1200 W and switching the reaction gas to mixed  $\text{H}_2/\text{CH}_4$  with a ratio of 9:1. The substrate temperature remained at 500 °C and no bias was intentionally applied to the substrate.

\* Corresponding author. Tel.: +886 3 5712121 56187.

E-mail address: [leechia2002@yahoo.com.tw](mailto:leechia2002@yahoo.com.tw) (C.-H. Li).

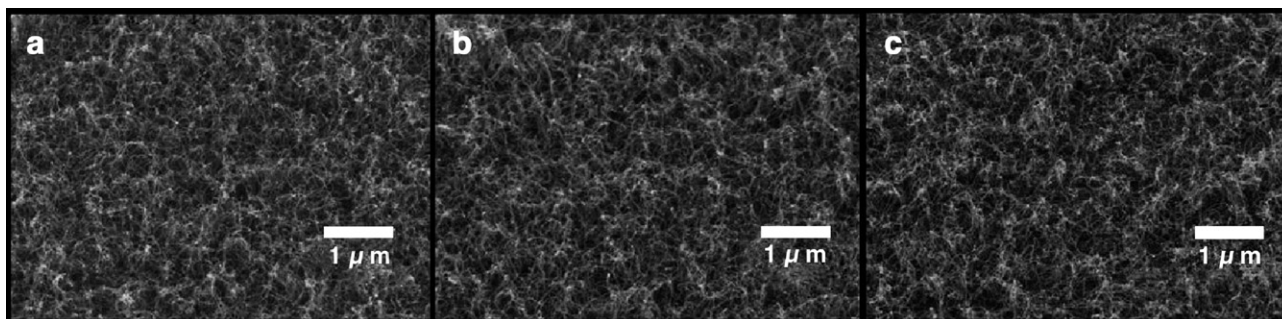


Fig. 1. SEM images of MWNT films: (a) as-grown films; (b) after irradiated with laser power density of 9 mJ/cm<sup>2</sup>, and (c) 14 mJ/cm<sup>2</sup>.

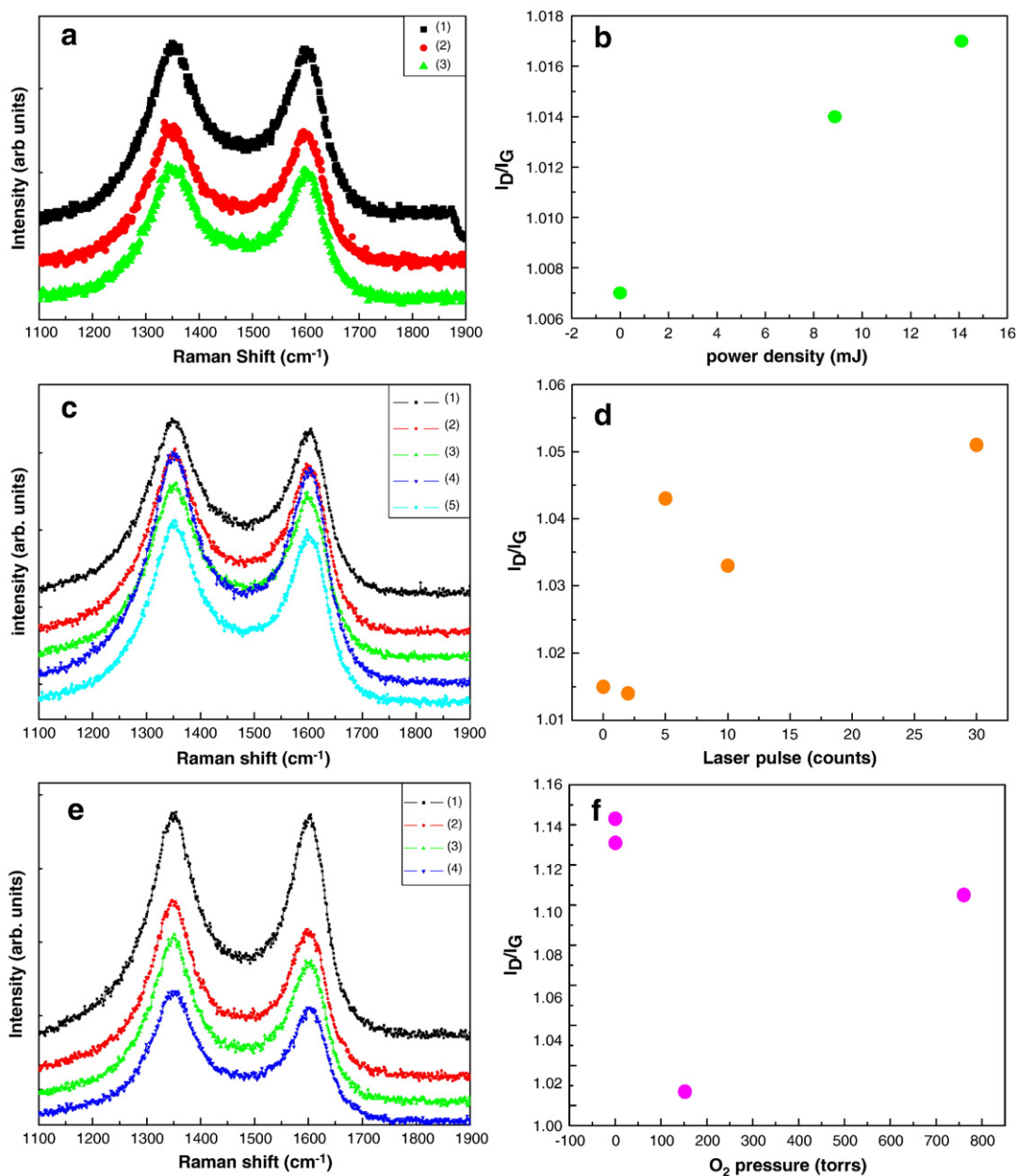


Fig. 2. The Raman spectra for MWNT films treated with different parameters. (a) The influence of laser power density; curves (1)–(3) show the spectra for the as-grown film, laser irradiated in air at 8.87 mJ/cm<sup>2</sup>, and 14.23 mJ/cm<sup>2</sup>, respectively. (b) The  $I_D/I_G$  variation with the irradiated laser power density. (c) Effects of delivered laser pulse counts ( $n$ ); curves (1)–(5) represent the result for as-grown,  $n=2$ ,  $n=5$ ,  $n=10$ , and  $n=20$  cases. (d) The corresponding  $I_D/I_G$  for the conditions depicted in (c). (e) Effects of oxygen pressure ( $PO_2$ ) during irradiation; curves (1)–(4) are for the as-grown,  $PO_2=5 \times 10^{-5}$ ,  $5 \times 10^{-1}$ , and 760 Torr, respectively. (f) The  $I_D/I_G$  for the corresponding conditions in (e). Note: each curve has been displaced vertically for clarity.

For post-treatment, the MWNT films were irradiated by a KrF (248 nm) excimer laser with power density and pulse counts being varied from 8 to 15 mJ/cm<sup>2</sup> and pulse counts from 2 to 40 counts, respectively. Unless specified, the irradiation was carried out in air with the incident laser beam being defocused. In some cases, oxygen was intentionally introduced during irradiation. Three background pressures were studied, namely  $5 \times 10^{-5}$ , 0.5, and 760 Torr. For field emission measurements, the 180  $\mu\text{m}$ -spacer diode structure was made on the ITO-coated glass substrate. The emission characteristics were obtained in a vacuum chamber under a pressure of  $10^{-5}$  Torr. The Keithley 237 source unit was used for measuring the current of emitted electrons with the voltage varying from 1 to 1100 V. The JEOL 7000F FE-SEM was employed to investigate the morphologies of the MWNT films. The excitation source of the Raman spectrometer is a Nd:YAG laser with the wavelength of 532 nm. For Raman spectroscopy the intensity of the two characteristic Raman shifts  $I_D$  (defect-mode) and  $I_G$  (graphite-mode) was measured. The high-resolution transmission electron microscopy (HRTEM) analysis was performed on Philips Tecnai 20 operated at 200 keV to examine the microstructure of CNT films.

### 3. Results and discussion

Fig. 1 shows the SEM images of the CNT films prior to and after the laser irradiation. There is no noticeable difference between these images. The results suggest that while laser irradiation might damage the individual CNT body, it does not alter the film morphology. We note here that changing the atmosphere during irradiation did not introduce noticeable film morphology modifications, either.

Fig. 2(a) shows the intensity of Raman shifts for MWNT films irradiated by different laser powers. The  $I_D/I_G$  as a function of laser power is plotted in Fig. 2(b). Since the ratio indicates the relative density of dangling bonds and defects to the more crystalline graphite structure [6–9], the increased  $I_D/I_G$  ratio with increasing laser power, thus, reflects the increase in the density of dangling bonds and crystal defects due to the energetic incident photons. It appears that the effect of the incident energetic photons, though may be inadequate to cause changes to the MWNT films in a macroscopic sense (e.g. morphology and distribution of the tubes), may induce some changes microscopically (e.g. graphite structure of CNTs body). The effect of delivered laser pulses on the  $I_D/I_G$  ratio also shows a similar trend, as evidenced in Fig. 2(c) and (d). However, the effects of oxygen introduced during irradiation displayed rather scattered results, as can be seen in Fig. 2(e) and (f). We suspect that, in addition to the physical bombardments, the latter cases may also involve some complicated chemical reactions.

To correlate these spectroscopic evidences with the emission properties, the current density vs electric field ( $J$ – $E$ ) curves were measured. Fig. 3(a) shows the  $J$ – $E$  curve for MWNT films prior to and after laser irradiation treatment. As is evident from the results, for film treated with 14.1 mJ/cm<sup>2</sup> laser intensity, the emission current density increases from 0.98 mA/cm<sup>2</sup> at an applied voltage of 1100 V to 2.46 mA/cm<sup>2</sup> at the same applied voltage, while the turn-on field decreases from 4.98

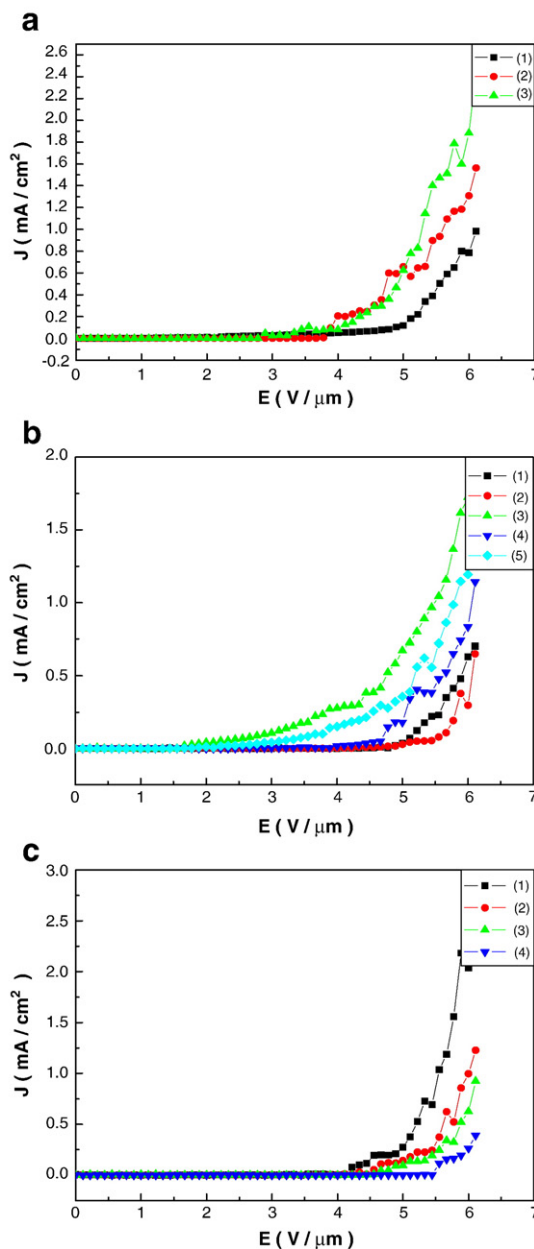


Fig. 3.  $J$ – $E$  curves for MWNT films treated with the different parameters described in the captions of Fig. 2.

to 3.15 V/ $\mu\text{m}$ . The turn-on electrical field is defined as the applied field strength at which the current density reaches  $J=0.1$  mA/cm<sup>2</sup>. Comparing the  $I_D/I_G$  correlation with the field emission performance indicates that increasing  $I_D/I_G$  results in lower turn-on field and higher emission current density. Both effects are considered to be beneficial to the field emission properties of MWNT films.

Comparing Fig. 2(d) with Fig. 3(b) indicates that the more incident laser pulse counts delivered resulted in more damages (reflected in the increasing  $I_D/I_G$  ratio), which, in turn, gives rise to an enhancement of emission current density from 0.7 mA/cm<sup>2</sup> to 1.8 mA/cm<sup>2</sup> at an applied voltage of 1100 V. Our results agree with those reported by Zhao et al. [8], wherein they use UV laser to improve the field emission characteristics.

Similar to the previous results, the turn-on field reduced from 5.2 to 2.9 V/ $\mu\text{m}$ . It appears that the delivered laser pulses have effects similar to laser energy density in enhancing the film emission performance. Moreover, these enhancements, unlike those obtained previously by applying either a bias voltage to the substrate or by adding other gases, are having rather different natures. In the previously practiced method, the microstructure and morphology of the films may have changed simultaneously and were difficult to control. On the other hand, the laser irradiation treatment appears to be more viable in modifying the  $I_D/I_G$  ratio in a systematic manner. In particular, the correlation between  $I_D/I_G$  ratio and emission performance has also been clearly demonstrated.

The above point can be further elaborated by comparing the values of  $I_D/I_G$  of the MWNT films after irradiation (1.143, 1.131, and 1.105) to that of the as-grown films ( $I_D/I_G=1.007$ ) shown in Fig. 2(e) and (f). It implies that laser irradiation can induce damages to the MWNTs regardless of atmospheric conditions (e.g. from  $5 \times 10^{-5}$  to 760 Torr), except that the extent of

damage might be reduced by dynamic recovery occurring at some particular oxygen pressure. This could be due to the fact that more oxygen burns out more amorphous carbon and the heat arising from inflammation brings about the annealing effect to make a better crystallite structure [8]. However, a further increase of oxygen partial pressure may induce further damage to the MWNT structure again. The  $J$ - $E$  curve displayed in Fig. 3(c), nonetheless, showed overall degradation in emission performance for irradiation under pure oxygen atmosphere. The current density at  $V=1100$  V changes from 2.62 mA/cm<sup>2</sup> for the as-grown film to 1.23, 0.92, and 0.39 mA/cm<sup>2</sup> for the film irradiated at oxygen pressure of  $5 \times 10^{-5}$ , 0.5, and 760 Torr, respectively. Similarly, the turn-on field ( $E_{\text{on}}$ ) changes from 4.33 V/ $\mu\text{m}$  for the as-grown film to 4.7, 5, and 5.6 V/ $\mu\text{m}$  for films irradiated under the corresponding oxygen pressures mentioned above.

Finally, in order to confirm the conjectures on the correlations between the post-treatment parameters and the emission performance, it is necessary to seek for direct support from

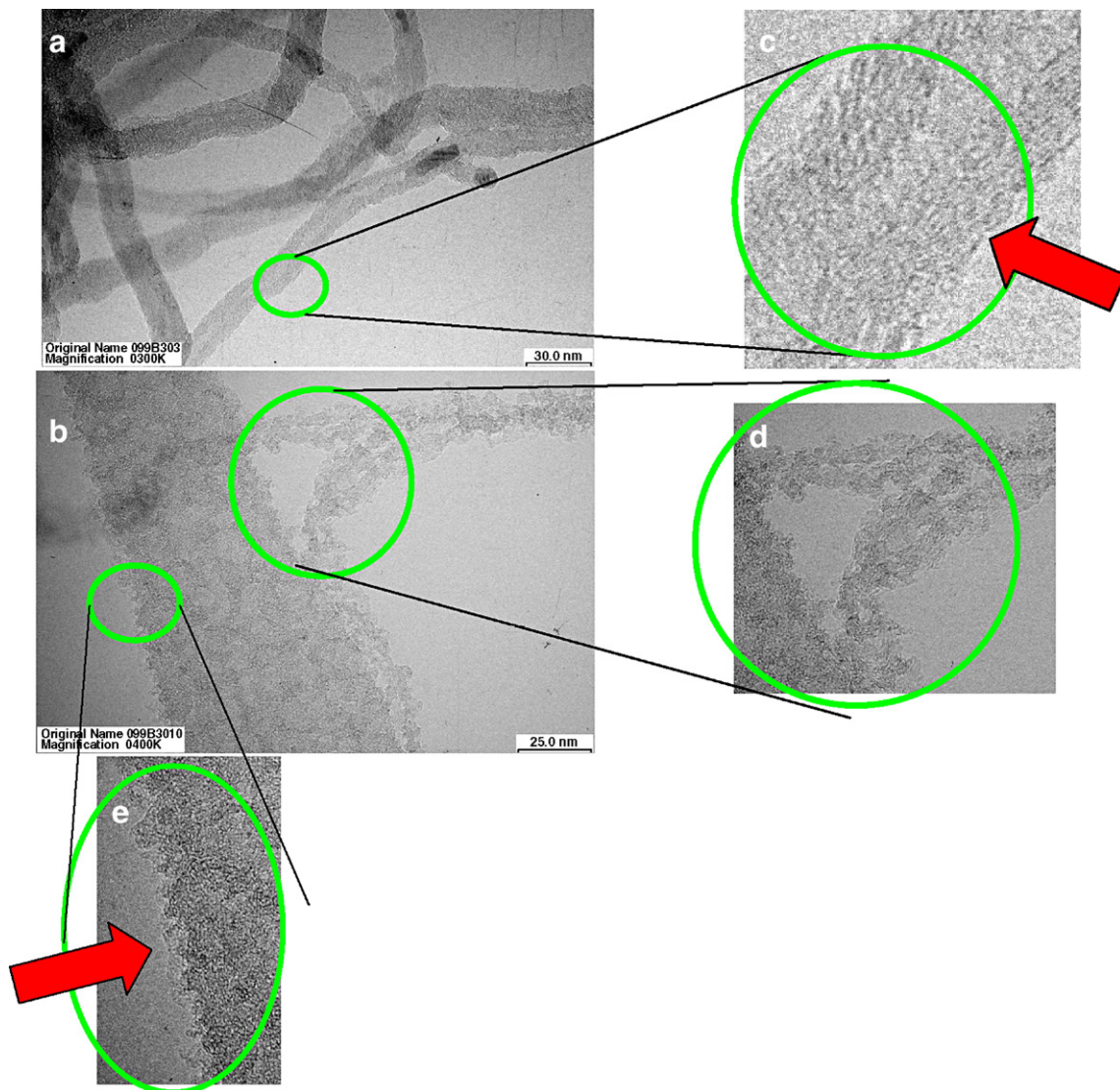


Fig. 4. HRTEM images for MWNT films (a) before, and (b) after laser irradiation. (c)–(e) Displays the enlarged view of the corresponding circled areas (see text) (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

microscopic structure analyses. Fig. 4 shows the high-resolution transmission electron microscopy (HRTEM) images for MWNT films prior to (a) and after laser irradiation (b), respectively. Fig. 4(c)–(e) demonstrates the enlarged view of the circled corresponding areas. It is evident from Fig. 4(c) that the multi-wall structure (indicated by the red arrow) of the as-grown film is clearly observable. However, after laser irradiation, not only the tube edge lost its smoothness and became rather irregular, but also the wall structure has been largely destroyed. This is quite consistent with our view of attributing the increased  $I_D/I_G$  ratio to the increased broken-bonds and structural damages caused by the bombardment of incident energetic photons. The enhancement of the emission performance could arise from either the increase of emission site or the enhancement of emission efficiency of particular sites or even both after laser treatment. At present, however, we are unable to reach the definite conclusion on this particular issue, and further investigations are needed.

#### 4. Summary

In summary, we have demonstrated in this report that the post-treatment by KrF excimer laser could effectively enhance the field emission properties of the multi-wall carbon nanotube films fabricated by low-temperature microwave plasma chemical vapor deposition. We obtained a factor of 2–3 enhancement in emission current density, and a significant reduction in the turn-on field for the MWNT films by the current process scheme. Introduction of oxygen during irradiation, however,

was found to degrade the performance of field emission properties. The improvement in the emission characteristics induced by the irradiation as revealed by the corresponding modifications in the  $I_D/I_G$  ratios suggested that the primary reason was probably due to the degenerating of the graphite structure induced by the photon bombardments [5,8,9]. The HRTEM images provide some direct evidences for the abovementioned conjectures.

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