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Citation: Applied Physics Letters 88, 263104 (2006); doi: 10.1063/1.2216881

View online: http://dx.doi.org/10.1063/1.2216881

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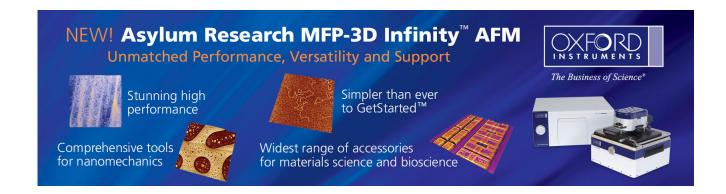
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Guided three-dimensional molecular self-assembly on silicon substrates

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(Received 2 April 2006; accepted 31 May 2006; published online 27 June 2006)

We demonstrate three-dimensional (3D) self-assemble growth of the metallothionein (Mn,Cd-MT-2) molecules on patterned semiconductor substrates. The MT molecules deposited on the patterned substrates were found to grow into 3D rod or ring-type nanostructures, depending on the shape of patterned nanostructures on the substrates. Dense arrays of 3D molecular nanorods or rings with an area density close to 10^{10} cm⁻² were demonstrated with a pore size of 20 nm and a pitch size of 100 nm. Those engineered molecular nanostructures provide an excellent opportunity for biological applications, sensing sources of nanodevices, biochemical reactions on surfaces, and even single molecule studies. © 2006 American Institute of Physics. [DOI: 10.1063/1.2216881]

During recent years, self-assembly has become one of the most important strategies used in biology to develop complex, functional structures and to arrive at one- to threedimensional ensembles of nanostructures. Self-assembly on surfaces is one of the approaches to self-organized structures that are particularly successful. By the coordination of molecules to surfaces, the molecular systems form ordered systems—self-assembled monolayers (SAMs). Thin films of diblock copolymers were found to self-assemble into ordered periodic structures at the molecular scale ($\sim 5-50$ nm) and have been used as templates to fabricate quantum dots, 1,2 nanowires, ^{3–5} and magnetic storage media. ⁶ More recently, in epitaxial assembly of block-copolymer films, molecular level control over the precise size, shape, and spacing of the order domains was achieved with advanced lithographic techniques.⁷ The development of methods for patterning and immobilizing biologically active molecules with micrometer and nanometer scale control has been proven integral to ranges of applications such as basic research, diagnostics, and drug discovery. Some of the most important advances have been in the development of biochip arrays that present either DNA, protein, or carbohydrates. The use of patterned substrates for components of microfluidic systems for bioanalysis is also progressing rapidly. 11-13

In recent years, there have been substantial attention focused on the surface modification and patterning at the nanoscale to anchoring protein molecules on silicon substrates and reactions of organic compounds with silicon surfaces. The major attraction is the combination of the richness of organic chemistry and the versatilities of semiconductor technology. It has been demonstrated that 2,2,6,6-tetramethylpiperidinyloxy (TEMPO) can bond with a single dangling bond on hydrogen-terminated Si(100) and Si(111)

surfaces. ¹⁴ Functional organic molecular layers were also found to self-assemble on metal ¹⁵ and semiconductor surfaces. ¹⁶ In this letter, we demonstrate methods of producing arrays of three-dimensional (3D) protein nanostructures on a Si surface. The combination of electron-beam (e-beam) lithography and self-assemble approaches provides not only a high throughput of arrays of protein nanostructures but also the highest precision.

Molecular structures of the metallothionein (MT-2) protein used in this experiment can be found in Ref. 17. The steps in synthesizing Mn,Cd-MT-2 molecules were described elsewhere. Spectra from both the UV absorption and the circular dichroism (CD) of our MT molecules are similar to those of the native MT-2. The effective diameter of Mn,Cd-MT-2, measured by dynamic light scattering (DLS) spectrophotometer, is 2.86±0.29 nm. The nanostructured templates

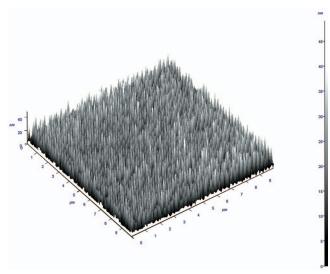


FIG. 1. (Color) AFM images of the dense molecular self-assembly arrays with a density of approximately $10^{10}\,\mathrm{cm^{-2}}$. The molecules have self-assembly grown into a rod shape.

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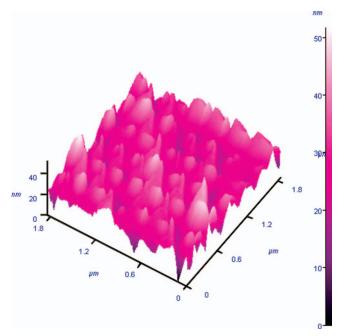


FIG. 2. (Color) Image of nanorods with AFM operated in contact mode.

were prepared on Si (001) substrates using e-beam lithography. The flow chart of the lithography and etching processes is given in Ref. 17 (supplementary, Fig. 2). A Si wafer was diced into 1 cm² substrates, which were cleaned with a modified RCA solution to remove organic contaminants and then dried at 150 °C for 1 h to drive off excess moisture. The fabrication process commences by spinning a thin layer of ZEP-520A photoresist onto the Si wafer using a Laurell coater and prebaking it at 180 °C for 2 min. Photoresist layers with different thicknesses were deposited onto Si substrates by varying the spin rates of the coater, and their thickness was determined from surface profile measurements. We were able to precisely control the photoresist thickness with accuracy within ±1 nm. Nanopore patterns were directly written in a square area of about $200 \times 200 \ \mu \text{m}^2$, with an e-beam writer operated at a fixed electron voltage of ~50 kV. Exposed samples were developed at 25 °C in ZED-N50 solution for a few minutes followed by residue ZEP-520A descumed through ULVAC ozone system. By using the patterned ZEP-520 as the hard mask, the exposed Si area was etched into the substrate with a depth of approximately 120 nm via the reactive ion etching (RIE) for about 80 s. Schematics of a completed nanostrucured template are also shown in Ref. 17 [supplementary, Fig. 2(b)]. By varying the thickness of the photoresist layers, e-beam exposure time and current, templates with nanopore sizes ranging from 20 to 150 nm and pitch sizes from 100 to 1200 nm were fabricated. Scanning electron microscope (SEM) images of the nanopores on one the templates were given in Ref. 17 (supplementary, Fig. 3). The size of the nanopores only increased by 1-2 nm after RIE processes. We were able to achieve a nanopore density as high as 10^{10} cm⁻² with a pitch size of 100 nm. The results indicated that we succeeded in preparing periodically distributed nanopore arrays on Si substrates. Their sizes and distributions are highly uniform and can be controlled with precision.

The self-assembled growth of the MT-2 proteins on the patterned templates is demonstrated as follows. The templates were immersed into the tris-HCL buffer solution with

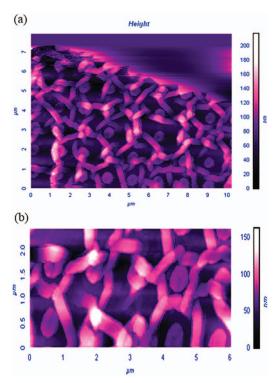


FIG. 3. (Color) (a) 3D AFM images of the molecular self-assembly on the template with unprotected Si surface. (b) Part of the image shown in (a) was enlarged by a factor of 2.

a MT-2 concentration of 1 mg/ml. An electric field, approximately 20 V/cm, was applied for 5 min to drive MT molecules into nanopores. The sample was then washed with de-ionized (DI) water twice to remove the unbounded MT molecules and salts on the surface (schematics of this process were shown in Ref. 17). After the deposition of MT molecules, surface profile and phase images of templates were measured with an atomic force microscope (AFM) operated in taping mode at room temperature. Figure 1 shows the image of the template surface with 40 nm nanopores and a pitch size of 100 nm after the deposition of MT molecules. Keep in mind that the top of the Si surface was still covered by photoresist after etching processes, which has prevented MT molecules to bond with the Si surface underneath. However, the surface inside nanopores inevitably contains dangling bonds after RIE processes. Therefore, the electrical field driven MT molecules were all anchored on the bare Si surface inside the pores. The MT molecules in each pore were found to self-assemble into a rod shape from the bottom of the pore and extended above the template surface. Those molecular nanorods have an average height of \sim 70 nm above the surface with a diameter equal to the size of the pore. Dense molecular nanorod arrays, as shown in Fig. 1, were prepared on the template with a high density $(\sim 10^{10} \text{ cm}^{-2})$ of nanopores. The molecular self-assembly appears to be quite "soft" and can be manipulated with the AFM tip. This was demonstrated in the AFM image shown in Fig. 2. With the AFM operated in contact mode due to larger scanning force applied on the molecular self-assembly compared to that in the taping mode, the nanorods were all bent down and leaned along the tip scanning direction.

Experiments on templates with pore sizes larger than 100 nm gave quite different results. Although the MT molecules which landed inside the larger pores were able to self-assemble, molecular self-assembly was not able to extend

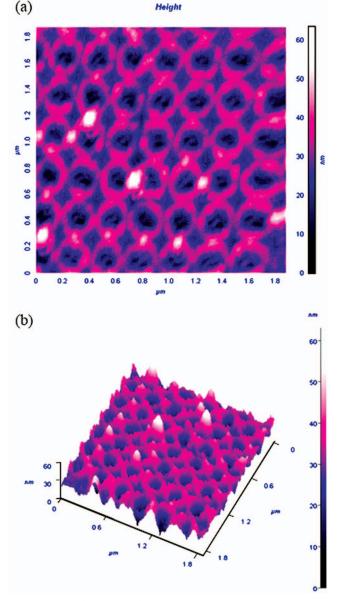


FIG. 4. (Color) (a) 2D and (b) 3D AFM images of the self-assembled molecular nanorings.

above the template surface even with a much longer deposition time. Therefore, we were not able to generate 3D images on this type of molecular self-assembly. We also uncover that the molecular self-assembly in the larger pores can diffuse laterally toward the neighboring pores. Judging from the AFM phase images, the MT molecules have self-assembled into a more dense structure in the larger pores compared to that in the small pores.

In order to observe the self-assembled growth of molecules on the unprotected Si surface, we intentionally stripped off the photoresist layer from one of the templates. Surprisingly, textile structures as shown in Fig. 3 were found on the template. It appears that, on the unprotected template, the molecular self-assembly inside the nanopores was able to extend themselves along the surface toward the neighboring pores and entangle with the others. In Fig. 3(b) we can see clearly that those molecular self-assembly located inside the nanopores were hollow. By using a photoresist layer with a thickness less than 120 nm and slightly longer e-beam exposure time, we were able to create a ring-type area with exposed Si surface around nanopores. On this particular temposed Si surface around nanopores.

plate, the MT molecules not only self-assemble inside the pores, they also assemble along the circumference of pores to form molecular nanorings. Figure 4 shows the two-dimensional (2D) and 3D AFM images of molecular rings.

It is desirable to understand the underlying self-assembled growth mechanism in order to gain better control of the formation of molecular nanostructures. The assembly of biological systems with complex structures is dictated by highly selective, noncovalent interactions, such as hydrogen bonding and van der Waals attractions. We believe that the first layer of proteins anchored inside the nanopores was bonded with the Si surface dangling bonds. They have provided building blocks for proteins which arrived at a later time. With the assistance of spatial confinement from the patterned nanostructures, the rest of the proteins are able to self-assemble via van der Waals interactions and form molecular self-assembly.

In conclusion, we have demonstrated methods of patterning molecular self-assembly and self-assembled growth of metallothionein (MT-2) molecules on nanostructured semiconductor surfaces. The MT molecules were shown to self-assemble into 3D nanostructures, depending on the structures patterned on the Si templates. Importantly, this work should not be limited to MT-2 and should be extended to other types of molecules and proteins.

This work was supported by the National Science Council of Taiwan under Contract No. NSC 94-2112-M-009-038 and by the MOE-ATU program.

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¹⁷See EPAPS Document No. E-APPLAB-88-204626 for the molecular structures of MT-2 proteins and preparation of Si templates. This document can be reached via a direct link in the online article's HTML reference section or via the EPAPS homepage (http://www.aip.org/pubservs/ epaps.html).

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