Journal of The Electrochemical Society, **153** (1) G44-G46 (2006) 0013-4651/2005/153(1)/G44/3/\$20.00 © The Electrochemical Society, Inc.



# Effect of Surface Passivation Removal on Planarization Efficiency in Cu Abrasive-Free Polishing

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Selective removal of surface passivation on protruded Cu film is a critical factor of Cu planarization. For a stress-free Cu abrasive-free polishing (Cu AFP) process, due to the lack of mechanical abrasion by abrasives, a polishing pad is used instead of abrasives to remove surface passivation during Cu planarization. Thus, the planarization efficiency in Cu AFP relates to the efficiency of surface passivation removed by a pad. Comparing Cu oxides with a non-native Cu-BTA (Cu-Benzotriazole) mono-layer used as surface passivation, this study found that an oxide-free Cu surface should be required in Cu AFP. When Cu oxides function as surface passivation in Cu AFP, they are removed with greater difficulty by a pad resulting in low planarization efficiency. Contrary to Cu oxides, high planarization efficiency can be obtained with non-native Cu-BTA as surface passivation in Cu AFP.

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Manuscript submitted April 7, 2005; revised manuscript received August 16, 2005. Available electronically November 22, 2005.

In a damascene chemical mechanical polishing (CMP) process, selective removal of surface passivation on protruded metal film is a critical factor of metal planarization. In Kaufman's model of metal CMP, a metal film is oxidized with oxidizers resulting in the formation of metal oxide on a metal surface during polishing.<sup>1</sup> In this case, metal oxide functions as surface passivation to reduce metal corrosion at the recess area. After selective removal of protruded metal oxide with mechanical abrasion by abrasives, the underlying metal is oxidized as metal oxide again leading to effective removal of protruded metal film. Consequently, global metal planarization can be obtained. However, the abrasives used to remove metal oxide would result in some mechanical damages in metal CMP, such as film peeling and scratches.

In recent years, in order to reduce mechanical damage, Cu abrasive-free polishing (Cu AFP) was developed as an alternative to Cu CMP for Cu planarization. In Kondo's study of Cu AFP, it was demonstrated that Cu planarization could be addressed without mechanical abrasion by abrasives.<sup>2</sup> In that case, a polishing pad was used instead of abrasives for selective removal of surface passivation on protruded Cu film. After effective removal of surface passivation with soft friction from a pad, an oxidizer is able to corrode the underlying Cu. But contrary to protruded Cu, there is no mechanical abrasion to remove surface passivation on recess Cu. Thus, due to great inhibition of passivation, the Cu removal at the recess region would be ineffective. Consequently, global Cu planarization is obtained due to effective removal of protruded Cu and great inhibition of surface passivation at recess Cu. Based on the above, the efficiency of surface passivation removed with a pad is a critical factor of planarization efficiency in Cu AFP. Even if different surface passivation has similar inhibition, their different removal efficiency on protruded Cu film would lead to a significant difference in planarization efficiency.

In order to verify the above viewpoint, in this study a non-native Cu-BTA (Cu-Benzotriazole) adlayer was used to compare with Cu oxides as surface passivation in Cu AFP. Based on the Pourbaix diagram, while Cu planarization is carried out with alkaline slurries, native Cu oxides would form on a Cu surface as surface passivation.<sup>3</sup> In contrast, a Cu surface is oxide-free during polishing in acid slurries. However, effective removal of Cu oxides requires great mechanical stress. In order to address stress-free Cu planarization, Cu AFP slurries are preferred to be kept in acid. Therefore, in this study, 1.1 M HNO<sub>3</sub> functioned as an oxidizer in order to keep a Cu AFP slurry in acid. BTA functioned as a corrosion inhibitor to form non-native Cu-BTA passivation through BTA molecules complex with Cu<sup>+</sup>.<sup>4</sup> Contrary to non-native Cu-BTA, Cu oxides were

used as another surface passivation in Cu AFP. By adding hydrogen peroxide  $(H_2O_2)$  in HNO<sub>3</sub> solution, a Cu surface would be oxidized as Cu oxides during polishing.

## Experimental

A patterned wafer with varying feature size patterns was used to evaluate the planarization efficiency of AFP slurries. The patterned wafer was prepared with a 150-mm-diam p-type silicon, (100) orientation wafer. After a standard RCA clean, 800-nm-thick SiO<sub>2</sub> was thermally grown on the silicon substrate in the furnace. Then, the thermal SiO<sub>2</sub> laid on the silicon substrate was patterned with various feature sizes (500, 300, 200, 150, 100, 50, 20, 10  $\mu$ m). By means of sputtering, a multilayer of 50-nm-thick tantalum under 1700-nm-thick Cu film was deposited on the patterned SiO<sub>2</sub> wafer. The original step height between protruded spaces and recess trenches was about 800 nm, by means of a surface profiler (Tencor P-10)

A Cu blanket wafer was used to evaluate the static Cu etch rate and Cu removal rate in AFP slurries. The blanket Cu wafer was prepared with a p-type, (100) orientation Si wafer. The Si wafer was used as the base on which 0.2-µm-thick SiO<sub>2</sub> was thermally deposited followed by a 50-nm-thick sputtered layer prior to depositing 1000-nm-thick Cu as the top.

A CMP machine (Westech model 372M), consisting of a circular polishing platen mounted with a Rodel IC 1400 K grooved (made of polyurethane impregnated polyester) pad, was used for the polishing process. The down force and back pressure were fixed at 3 and 0 psi, respectively. The rotating speeds of platen and carrier were 55 and 60 rpm, and the flow rate of slurry was 150 mL/min. In this study, Cu removal rate was evaluated with a four-point probe by measuring the change in Cu sheet resistance before and after polishing. However, for Cu patterned wafer polishing, the Cu removal at protruded spaces of various feature scales was hard to exactly evaluate with today's technology. Thus, the Cu removal in Cu patterned wafer polishing was estimated by measuring the variation of Cu sheet resistance at the blanket area on a Cu patterned wafer before and after polishing.

All polishing experiments were carried out with a new polishing solution comprised only of chemical agents such as oxidizer and corrosion inhibitor, but no abrasives. In this study, 1.1 M HNO<sub>3</sub> solution containing 20  $\mu$ M BTA was used as an AFP slurry in order to form non-native Cu-BTA passivation on a Cu surface during polishing. In 1.1 M HNO<sub>3</sub> solution, a 0.05 pH value helps keep the Cu surface oxide-free, which is necessary for BTA to form Cu-BTA complex. On an oxide-free Cu surface, Cho's study indicated that Cu-BTA complex would vertically stack on a Cu surface as passivation against HNO<sub>3</sub> corrosion.<sup>5</sup>

Contrary to non-native Cu-BTA, in this study, Cu oxides were

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Figure 1. Mechanism of the Cu AFP technique.

used as another surface passivation in Cu AFP. By adding 0.5 vol %  $H_2O_2$  in 1.1 M HNO<sub>3</sub> solution,  $H_2O_2$  can decompose OH<sup>-</sup> to oxidize a Cu surface as Cu oxides, as shown below<sup>6</sup>

 $\mathrm{H_2O_2} + 2e^- \rightarrow 2 \ \mathrm{OH^-}$ 

 $Cu + OH^{-} \rightarrow Cu(OH) + e^{-}, \ 2Cu(OH) \leftrightarrow Cu_{2}O + H_{2}O$ 

$$Cu + 2 OH^- \rightarrow Cu(OH)_2 + 2e^-, Cu(OH)_2 \leftrightarrow CuO + H_2O$$

However, the inhibition of Cu oxides against HNO<sub>3</sub> corrosion was not as effective as Cu-BTA. In order to maintain the two kinds of surface passivation to have similar inhibition, 1 mM BTA was added in 1.1 M HNO<sub>3</sub> solution containing 0.5 vol % H<sub>2</sub>O<sub>2</sub> to formulate an AFP slurry. Consequently, the pH value of an AFP slurry containing H<sub>2</sub>O<sub>2</sub> was about 0.05, which was the same as that of 1.1 M HNO<sub>3</sub> solution containing 20  $\mu$ M BTA. Furthermore, an AFP slurry containing H<sub>2</sub>O<sub>2</sub> had a static Cu etch rate of about 122 nm/min. The static Cu etching rate in 1.1 M HNO<sub>3</sub> solution containing 20  $\mu$ M BTA was about 107 nm/min.

Electrochemical analysis was used to observe Cu corrosion behavior in Cu AFP solution. All measurements were performed with an EG&G potentiostat model 273 with a rotating working electrode of 99.9% Cu, an AgCl reference electrode, and a Pt counter electrode.

#### **Results and Discussion**

For a stress-free Cu AFP process, selective removal of surface passivation on protruded Cu film could be carried out with a polishing pad. After selective removal of surface passivation on protruded Cu film with a pad, the underlying Cu corroded by oxidizers results in Cu planarization, as shown in Fig. 1.<sup>2</sup> However, this study found that the efficiency of surface passivation removed by a pad strongly depends on the characteristic of surface passivation. Indeed, even if different surface passivation has similar inhibition to reduce Cu corrosion at recess trenches, a difference in the efficiency of passivation removed on protruded spaces would lead to significantly different planarization efficiency in Cu AFP.

In this study, 1.1 M HNO<sub>3</sub> solution containing 20  $\mu$ M BTA is used as an AFP slurry for Cu patterned wafer polishing. After 462nm-thick Cu removal, the step height reduction of various feature Cu lines can be above 155 nm, as shown in Fig. 2. In this case, effective Cu planarization is supposed due to high Cu-BTA removal efficiency at protruded spaces and great Cu-BTA inhibition at recess trenches. Based on Pourbaix diagram, 1.1 M HNO<sub>3</sub> solution has a 0.05 pH value to keep the Cu surface oxide-free during polishing.<sup>3</sup> By means of AC impedance, due to an oxide-free Cu surface, the



Figure 2. Effective Cu planarization with an AFP slurry formulated with 1.1 M HNO<sub>3</sub> and 20  $\mu$ M BTA.

charge-transfer resistance in 1.1 M HNO<sub>3</sub> solution is almost neglected, as shown in Fig. 3. On an oxide-free Cu surface, BTA molecules complex with Cu<sup>+</sup> onto a Cu surface to form a Cu-BTA passivation layer against HNO3 corrosion. In Cho's study, FI-STM (field ion-scanning tunneling microscope) images showed that Cu-BTA passivation was a monolayer with an orderly stack on an oxidefree Ĉu surface.<sup>5</sup> Because Cu-BTA passivation is a monolayer, this study supposes that a polishing pad could effectively remove the Cu-BTA passivation on protruded spaces. The Cu static etch rate in 1.1 M HNO<sub>3</sub> solution can be above 330 nm/min. However, in the addition of 20 µM BTA, Cu-BTA monolayer forms on an oxide-free Cu surface leading to an increase in charge-transfer resistance from 0 to 1400  $\Omega$ . Then, static Cu etch rate is reduced from 330 to 120 nm/min. In other words, the formation of Cu-BTA passivation on a Cu surface can effectively reduce Cu corrosion at recess trenches.

Contrary to non-native Cu-BTA, another surface passivation used in Cu AFP is composed with Cu oxides and Cu-BTA. In this case, an AFP slurry is formulated with 1.1 M HNO<sub>3</sub>, 1 mM BTA, and 0.5 vol % H<sub>2</sub>O<sub>2</sub>. For Cu removal with an AFP slurry containing H<sub>2</sub>O<sub>2</sub>, a Cu surface is not oxide-free during polishing. 0.5 vol % H<sub>2</sub>O<sub>2</sub> can decompose OH<sup>-</sup> to oxidize a Cu surface as Cu oxides during polishing.<sup>6</sup> By means of X-ray photoelectron spectra, the



Figure 3. Charge-transfer resistance of the Cu surface in  $1.1 \text{ M HNO}_3$  solution with increasing BTA concentration.



Figure 4. X-ray photoelectron spectra of the Cu surface after polishing with an AFP slurry containing  $H_2O_2$ .

surface passivation in an AFP slurry containing  $H_2O_2$  is composed of Cu oxides and Cu-BTA, as shown in Fig. 4. Moreover, in an AFP slurry containing  $H_2O_2$ , the static Cu etch rate is about 122 nm/min and close to it in 1.1 M HNO<sub>3</sub> solution with 20  $\mu$ M BTA. For Cu patterned wafer polishing with AFP slurries, due to the lack of mechanical abrasion by abrasives, Cu removal at recess trenches depends on chemical formulation of AFP slurries. Therefore, similar Cu etch rate indicates that the effect of Cu removal at recess trenches on planarization efficiency in two AFP slurries can also be supposed almost the same.

However, even though the surface passivation in two AFP slurries has similar inhibition to reduce Cu removal at recess trenches, their different removal efficiency on protruded spaces leads to a significant difference in planarization efficiency in Cu AFP. For Cu patterned wafer polishing with an AFP slurry containing H<sub>2</sub>O<sub>2</sub>, after 1032-nm-thick Cu removal, the step height reduction of various feature Cu lines is still less than 100 nm, as shown in Fig. 5. Based on the Cu AFP model of Kondo, the planarization efficiency in Cu AFP strongly depends on the inhibition of surface passivation at recess trenches and selective removal of surface passivation on protruded spaces.<sup>5</sup> Because the effect of Cu removal at recess trenches on planarization efficiency in two AFP slurries is supposed the same, a difference in planarization efficiency should be due to different removal efficiency of surface passivation on spaces. In an AFP slurry containing H<sub>2</sub>O<sub>2</sub>, surface passivation is composed with Cu-BTA and native Cu oxides. However, effective removal of metal oxide needs strong mechanical stress. Hence, the reduction of planarization efficiency should be due to hard removal of Cu oxides by a pad. Otherwise, it can be observed that the pattern effect in an AFP slurry containing H<sub>2</sub>O<sub>2</sub> is slight, as shown in Fig. 5. That is a significant phenomenon to signify that low planarization efficiency is due to ineffective removal of Cu oxides. In a Cu damascene process, Cu



Figure 5. Cu patterned wafer polishing with an AFP slurry containing H<sub>2</sub>O<sub>2</sub>.

planarization suffers from the problem of pattern effect due to Cu removal rate proportional to effective local pressure. While the passivation removal is ineffective by a pad, it can be supposed that no mechanical abrasion contributed to Cu removal rate. Hence, Cu removal rate would strongly depend on Cu corrosion and Cu removal becomes pressure-independent. Consequently, patter effect is neglected and the step height reduction at all feature scales is similar. Based on the above, when surface passivation on protruded spaces cannot be effectively removed, the underlying Cu film cannot be corroded by HNO<sub>3</sub> leading to lower planarization efficiency. Indeed, an oxide-free Cu surface is necessary in Cu AFP. Furthermore, the surface passivation used in Cu AFP should be required to have high removal efficiency by a polishing pad.

## Conclusions

For a Cu AFP process, a Cu surface should be oxide-free during polishing. When Cu oxides are used as surface passivation in Cu AFP, their hard removal by a polishing pad would result in low planarization efficiency. In order to obtain high planarization efficiency, the surface passivation used in Cu AFP should be required to have high removal efficiency by a polishing pad.

National Chiao Tung University assisted in meeting the publication costs of this article.

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