

Dehydroxyl effect of Sn-doped silicon oxide resistance random access memory with supercritical CO₂ fluid treatment

Tsung-Ming Tsai, Kuan-Chang Chang, Ting-Chang Chang, Yong-En Syu, Kuo-Hsiao Liao, Bae-Heng Tseng, and Simon M. Sze

Citation: *Applied Physics Letters* **101**, 112906 (2012); doi: 10.1063/1.4750235

View online: <http://dx.doi.org/10.1063/1.4750235>

View Table of Contents: <http://scitation.aip.org/content/aip/journal/apl/101/11?ver=pdfcov>

Published by the [AIP Publishing](#)

Articles you may be interested in

[Mechanism of power consumption inhibitive multi-layer Zn:SiO₂/SiO₂ structure resistance random access memory](#)

J. Appl. Phys. **114**, 234501 (2013); 10.1063/1.4843695

[Electrical conduction mechanism of Zn:SiO_x resistance random access memory with supercritical CO₂ fluid process](#)

Appl. Phys. Lett. **103**, 083509 (2013); 10.1063/1.4819162

[A reactive magnetron sputtering route for attaining a controlled core-rim phase partitioning in Cu₂O/CuO thin films with resistive switching potential](#)

J. Appl. Phys. **113**, 183522 (2013); 10.1063/1.4804326

[Effect of hydrogen/deuterium incorporation on electroforming voltage of SiO_x resistive random access memory](#)

Appl. Phys. Lett. **101**, 183505 (2012); 10.1063/1.4765356

[Reducing operation current of Ni-doped silicon oxide resistance random access memory by supercritical CO₂ fluid treatment](#)

Appl. Phys. Lett. **99**, 263501 (2011); 10.1063/1.3671991



NEW! Asylum Research MFP-3D Infinity™ AFM
Unmatched Performance, Versatility and Support

OXFORD INSTRUMENTS
The Business of Science®

Stunning high performance
Simpler than ever to GetStarted™
Comprehensive tools for nanomechanics
Widest range of accessories for materials science and bioscience

The advertisement features several images: a blue textured surface, a brown textured surface, a grid of colorful rectangular samples, and the Asylum Research MFP-3D Infinity AFM instrument.

Dehydroxyl effect of Sn-doped silicon oxide resistance random access memory with supercritical CO₂ fluid treatment

Tsung-Ming Tsai,^{1,a)} Kuan-Chang Chang,¹ Ting-Chang Chang,^{2,3,a)} Yong-En Syu,³ Kuo-Hsiao Liao,¹ Bae-Heng Tseng,¹ and Simon M. Sze^{2,4}

¹Department of Materials and Optoelectronic Science, National Sun Yat-Sen University, 70 Lien-hai Road, Kaohsiung 804, Taiwan

²Department of Physics, National Sun Yat-Sen University, 70 Lien-hai Road, Kaohsiung 804, Taiwan

³Advanced Optoelectronic Technology Center, National Cheng Kung University, Taiwan

⁴Department of Electronics Engineering and Institute of Electronics, National Chiao Tung University, Hsinchu 300, Taiwan

(Received 29 March 2012; accepted 21 August 2012; published online 14 September 2012)

The tin-doped can supply conduction path to induce resistance switching behavior. However, the defect of tin-doped silicon oxide (Sn:SiO_x) increased the extra leakage path lead to power consumption and joule heating degradation. In the study, supercritical CO₂ fluids treatment was used to improve resistive switching property. The current conduction of high resistant state in post-treated Sn:SiO_x film was transferred to Schottky emission from Frenkel-Poole due to the passivation effect. The molecular reaction model is proposed that the defect was passivated through dehydroxyl effect of supercritical fluid technology, verified by material analyses of x-ray photoelectron spectroscopy and Fourier transform infrared spectroscopy. © 2012 American Institute of Physics. [<http://dx.doi.org/10.1063/1.4750235>]

To overcome the technical and physical limitation issues of conventional charge storage-based memories,¹⁻⁴ the resistance random access memory (RRAM) composed of an insulating layer sandwiched by two electrodes is a great potential candidate for next-generation nonvolatile memory due to their superior properties such as low cost, simple structure, fast operation speed, and nondestructive readout.^{5,6}

In our previous research, supercritical CO₂ (SCCO₂) fluid technology was used to improve the dielectric properties and performance of various thin film transistors (TFTs), such as hydrogenated amorphous-silicon TFTs and ZnO TFTs.⁷⁻¹³ The liquid-like and gas-like double properties of SCCO₂ fluids can be used to dissolve and transport H₂O molecules into the thin film and assist in oxidizing thin film at a low temperature. In addition, CO₂ is a nontoxic, non-flammable, and chemical-stable material. Although most RRAM devices have many superior properties of nonvolatile memory, the high operation current of RRAM during steady state is a major issue to nonvolatile memory for the application of portable electronic products. Therefore, the supercritical CO₂ is worthy to develop for improving the electrical properties of RRAM switching layer.

The tin-doped can supply conduction path to induce resistance switching behavior.¹⁴ However, the defect of tin-doped silicon oxide (Sn:SiO_x) would increase the extra leakage path lead to power consumption and joule heating degradation. In this work, the resistive switching layer of Sn-doped silicon oxide (Sn:SiO_x) was treated by SCCO₂ fluids to enhance its electrical properties. The Pt/Sn:SiO_x/TiN sandwiched devices were fabricated to investigate resistive switching properties of Sn:SiO_x after SCCO₂ treatment. In addition, the influence of SCCO₂ treatment on resistive switching behaviors of Sn:SiO_x was evaluated by

material and conduction mechanism analyses. Because the supercritical fluid has gaslike and high pressure properties to efficiently diffuse into nanoscale without damage,¹⁵ the current of post-treated Sn:SiO_x was reduced obviously due to the trap passivated by H₂O molecule of the SCCO₂ fluids.

The experimental samples were prepared as follows: the Sn:SiO_x thin film (about 30 nm) was deposited on the TiN/Ti/SiO₂/Si substrate by co-sputtering with the pure SiO₂ and Sn targets. The sputtering power was fixed at RF power 200 W and 3 W for SiO₂ and Sn targets, respectively. The co-sputtering was carried out in argon ambient (Ar = 30 sccm) with a working pressure of 6 mTorr at room temperature. In contrast, the Sn:SiO_x films were put into the reactive chamber of supercritical fluid system and then the SCCO₂ fluid mixed with 0.5 ml water were syringed into the reactive chamber to treat the sample. During the treatment, the water-mixed supercritical CO₂ fluids were heated and pressured to 120 °C and 3000 psi in the stainless steel chamber of supercritical fluid system for 1 h. Finally, the Pt top electrode of 200 nm thickness was deposited on Sn:SiO₂ film to form electrical devices with Pt/Sn:SiO_x/TiN sandwich structures by DC magnetron sputtering. The entire electrical measurements of devices with the Pt electrode of 250 μm diameter were performed using Agilent B1500 semiconductor parameter analyzer. In addition, the Fourier transform infrared spectroscopy (FTIR) measured by Bruker VERTEX 70v spectrometer in far infrared region and x-ray photoelectron spectroscopy (XPS) were used to analyze the chemical composition and bonding of these insulator materials, respectively.

The “forming process” is required to activate all of the Sn:SiO_x RRAM devices, using dc voltage sweeping with a compliance current of 2 mA. The leakage current of the Sn:SiO_x RRAM devices after SCCO₂ treatment was lower than that of pre-treatment devices (Figure 1(a)). This phenomenon is attributed to the improvement on dielectric properties

^{a)} Authors to whom correspondence should be addressed. Electronic addresses: tchang@mail.phys.nsysu.edu.tw and tmtsai@faculty.nsysu.edu.tw.

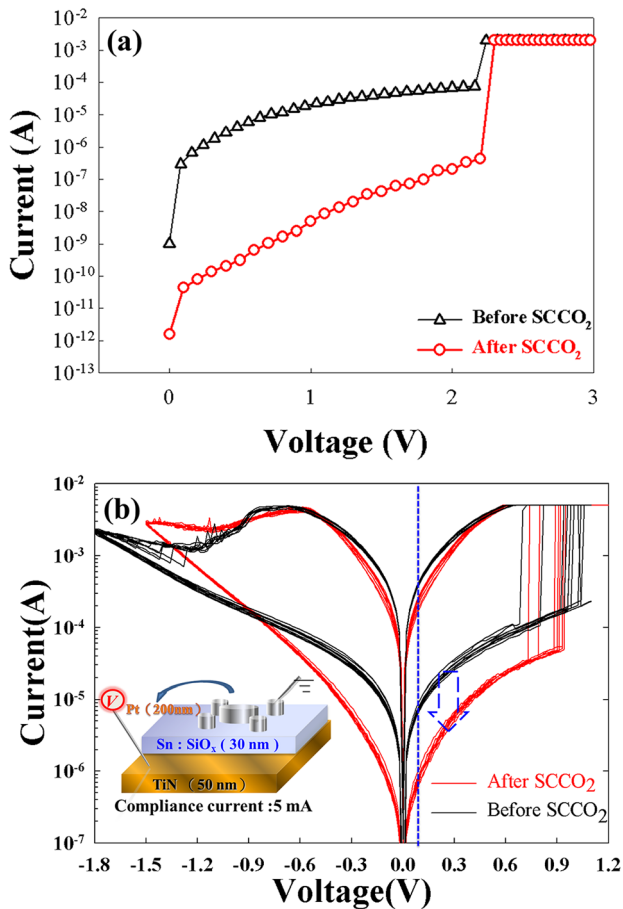


FIG. 1. (a) The forming current curves of the Sn:SiO_x RRAM devices before and after SCCO₂ treatment. (b) The black and red curves are the resistive switching characteristics of Sn:SiO_x film before and after SCCO₂ treatment, respectively. The current in high resistance state of post-treated Sn:SiO_x film is reduced about 15 times from 9 μA to 0.6 μA.

through SCCO₂ treatment, which has been reported by our previous study.⁸ After the forming process, the electrical current-voltage properties of the Sn:SiO_x devices were compared before and after SCCO₂ treatment (Figure 1(b)). The current of Sn:SiO_x devices is reduced at 0.1 V reading voltage after SCCO₂ treatment. Figure 1(b) shows the electrical current-voltage (I-V) properties of the Sn:SiO_x RRAM devices before and after SCCO₂ treatment. We can find that the current of Sn:SiO_x in high resistive state (HRS) is reduced from 9 μA to 0.6 μA at 0.1 V reading voltage after SCCO₂ treatment. The interesting phenomenon indicates that the increment of operation resistance for readout is about 15 times after SCCO₂ treatment. To investigate the current reduction mechanism, we analyzed the current conduction mechanism in HRS of Sn:SiO_x with and without SCCO₂ treatment as shown in Figure 2. The relationship in the curve of $\ln(I/V)$ versus the square root of the applied voltage ($V^{1/2}$) is linear. According to the relationship of Frenkel Poole conduction, $I = \frac{qN_c\mu}{d} V \exp\left[\frac{q}{kT} \left(2\sqrt{\frac{qV}{4\pi\epsilon_0 d}} - \phi_{Bt}\right)\right]$, where d , N_c , μ , ϵ_0 , and ϕ_{Bt} are the insulator thickness, density of ionized traps, carrier mobility, dielectric permittivity, and trap barrier height, respectively. The Frenkel Poole conduction is due to emission of trapped electrons into conduction band. The supply of electrons from the traps is through thermal excitation. The barrier reduction is larger than in the case of Schottky emission by a

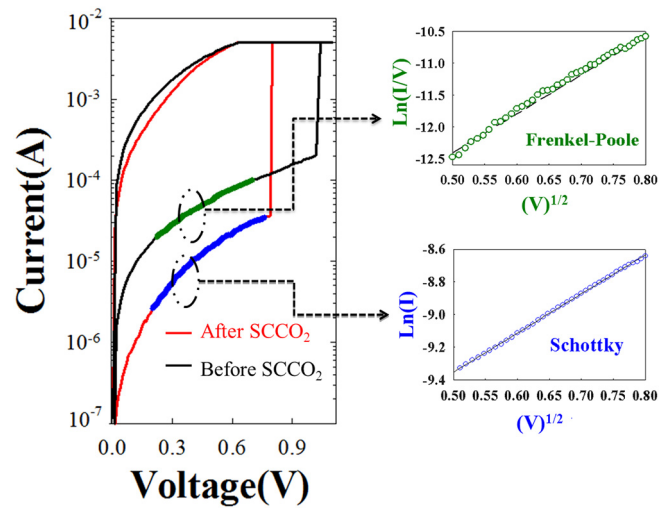


FIG. 2. The current conduction curves in the Sn:SiO_x film before and after SCCO₂ treatment.

factor of 2, which can be obtained as compared with the slope of the plot of $\ln(I)$ versus ($V^{1/2}$) based on the formula of Schottky emission, $I = AA^*T^2 \exp\left[\frac{q}{kT} \left(\sqrt{\frac{qV}{4\pi\epsilon_0 d}} - \phi_B\right)\right]$, where ϕ_B , d , A , and A^* are the Schottky barrier height, film thickness, electrode area, and Richardson's constant for thermionic emission, respectively.¹⁶ The results revealed that the carrier transport of Sn:SiO_x film was dominated by Frenkel Poole conduction due to the trap in the film. After SCCO₂ treatment, the current conduction mechanism will transfer to Schottky emission because of the improvement of dielectric properties. Therefore, we utilized the material spectra analyses to find out the reason of electrical transfer mechanism from Frenkel Poole conduction to Schottky emission. Compared the FTIR spectra of Sn:SiO_x film with and without SCCO₂ treatment (Figure 3), we found that the absorption peak of Sn-O bond at 586 cm⁻¹ was increased after SCCO₂ treatment. The result implies that the density of Sn-O bond was increased in the Sn:SiO_x film after SCCO₂ treatment. In addition, the absorption of Si-O-Si stretch bond at 450 cm⁻¹ was also increased after SCCO₂ treatment, illustrating the

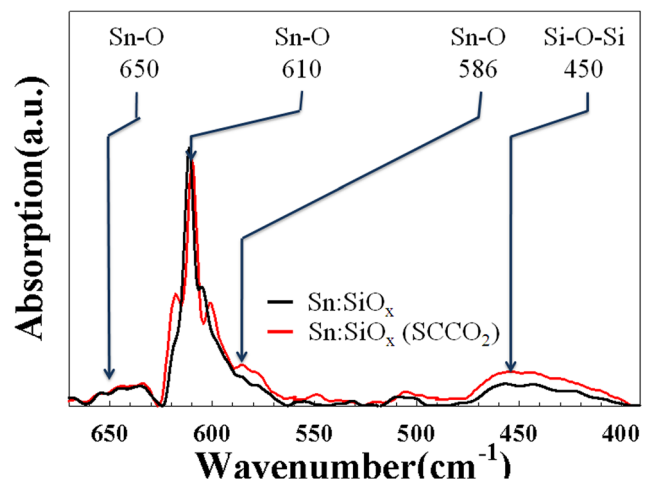


FIG. 3. The comparison of FTIR spectra of Sn:SiO_x film before and after SCCO₂ treatment. Both intensity of Sn-O and Si-O-Si bonds are increased in Sn:SiO_x film after SCCO₂ treatment.

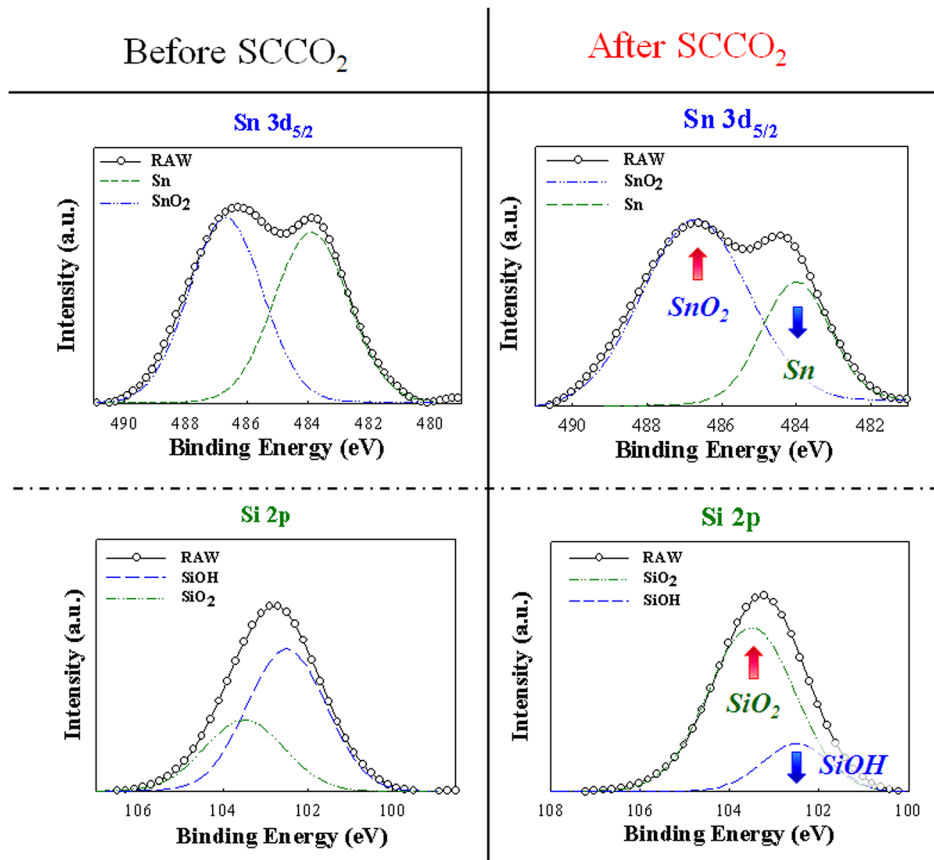


FIG. 4. XPS spectra of Sn $3d_{5/2}$ and Si $2p$ core levels in Sn:SiO_x film before and after SCCO₂ treatment. The mole fraction of metallic tin and Si-OH bonds in Sn:SiO_x film is reduced obviously but that of tin oxide and silicon oxide bonds is increased after SCCO₂ treatment.

content of silicon oxide bonding in the film also increased.¹⁷ According to XPS spectra analyses for Sn $3d_{5/2}$ core level (Figure 4), the mole fraction of Sn-O bond was obviously risen but that of Sn element was decreased in Sn:SiO_x film after SCCO₂ treatment. Besides, the mole fraction of Si-O bond was substantially increased in contrast with that of Si-OH bond after SCCO₂ treatment in terms of the XPS spectra analyses of Si $2p$ core level as shown in Figure 4. Therefore, we infer that the level of oxidation would increase and accompany with dehydration in the post SCCO₂-treated film. These results were consistent with the above-mentioned FTIR analyses.

Based on the material analyses results, we proposed a reaction model to explain the current reduction mechanism of

Sn:SiO_x film with SCCO₂ treatment as shown in Figure 5. As the sample was put into the water-mixed SCCO₂ fluid environment, the H₂O molecule was carried into the dangling bonds of amorphous Sn:SiO_x film by SCCO₂ fluid, which is attributed to the high penetration ability of SCCO₂ fluid. The H₂O molecule was approached to dangling bonds leading to the hydration reaction in the Sn:SiO_x film. Then, monomolecular CO₂ in supercritical fluids induces the dehydration of neighbor hydroxyl groups so as to form Si-O-Si and Sn-O-Si cross-linking bonding in the film. Hence, the trap of Sn:SiO_x film can be passivated by SCCO₂ treatment, which can cause the electrical current conduction in HRS of Sn:SiO_x film transferred from Frenkel Poole conduction to Schottky emission. The phenomena will cause the improvement of dielectric

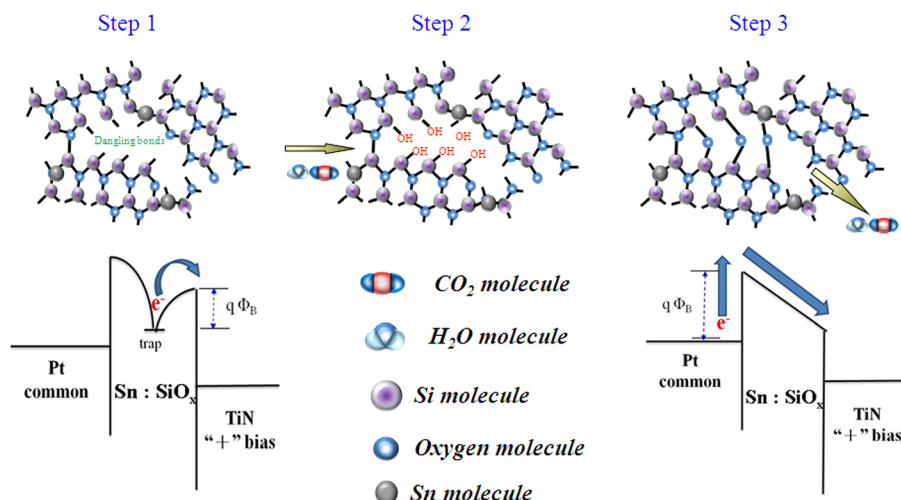


FIG. 5. The schematic diagram of passivation mechanism of SCCO₂ treatment on Sn:SiO_x film. The schematic structure for each step represents the situation of chemical bonding in the amorphous Sn:SiO_x film before and after SCCO₂ treatment.

properties of thin film, which is demonstrated by our previous study.⁸

In summary, the operation current of Sn-doped silicon oxide RRAM device was decreased by supercritical fluid treatment in this study. The water molecular can be brought into the film by supercritical CO₂ fluid, which induce dehydroxyl effect to passivate the dangling bond in the amorphous resistive switching layer. The operation resistance of RRAM can be increased due to the decrease of defect in the layer, which results in low power consumption. Therefore, supercritical fluid treatment can improve the properties of resistive switching layer of RRAM device.

This work was performed at National Science Council Core Facilities Laboratory for Nano-Science and Nano-Technology in Kaohsiung-Pingtung area and supported by the National Science Council of the Republic of China under Contract Nos. NSC 100-2120-M-110-003 and NSC 100-2221-E-110-060.

¹T. C. Chang, F. Y. Jian, S. C. Chen, and Y. T. Tsai, *Mater. Today* **14**(12), 608 (2011).

²F. M. Yang, T. C. Chang, P. T. Liu, P. H. Yeh, Y. C. Yu, J. Y. Lin, S. M. Sze, and J. C. Lou, *Appl. Phys. Lett.* **90**, 132102 (2007).

³C. H. Lin and Y. Kuo, *J. Appl. Phys.* **110**, 024101 (2011).

⁴F. M. Yang, T. C. Chang, P. T. Liu, U. S. Chen, P. H. Yeh, Y. C. Yu, J. Y. Lin, S. M. Sze, and J. C. Lou, *Appl. Phys. Lett.* **90**, 222104 (2007).

⁵Y. T. Li, S. B. Long, M. H. Zhang, Q. Liu, L. B. Shao, S. Zhang, Y. Wang, Q. Y. Zuo, S. Liu, and M. Liu, *IEEE Electron Device Lett.* **31**, 117 (2010).

⁶Y. E. Syu, T. C. Chang, T. M. Tsai, Y. C. Hung, K. C. Chang, M. J. Tsai, M. J. Kao, and S. M. Sze, *IEEE Electron Device Lett.* **32**, 545 (2011).

⁷C. T. Tsai, T. C. Chang, P. T. Liu, P. Y. Yang, Y. C. Kuo, K. T. Kin, P. L. Chang, and F. S. Huang, *Appl. Phys. Lett.* **91**, 012109 (2007).

⁸C. T. Tsai, T. C. Chang, K. T. Kin, P. T. Liu, P. Y. Yang, C. F. Weng, and F. S. Huang, *J. Appl. Phys.* **103**, 074108 (2008).

⁹M. C. Chen, T. C. Chang, S. Y. Huang, K. C. Chang, H. W. Li, S. C. Chen, J. Lu, and Y. Shi, *Appl. Phys. Lett.* **94**, 162111 (2009).

¹⁰C. T. Tsai, P. T. Liu, T. C. Chang, C. W. Wang, P. Y. Yang, and F. S. Yeh, *IEEE Electron Device Lett.* **28**, 584 (2007).

¹¹C. T. Tsai, T. C. Chang, P. T. Liu, Y. L. Cheng, K. T. Kin, and F. S. Huang, *Electrochem. Solid State Lett.* **12**, H35 (2009).

¹²M. C. Chen, T. C. Chang, S. Y. Huang, K. C. Chang, H. C. Huang, S. C. Chen, J. Liu, D. S. Gan, N. J. Ho, T. F. Young, G. W. Jhang, and Y. H. Tai, *Surf. Coat. Technol.* **204**, 1112 (2009).

¹³K. C. Chang, T. M. Tsai, T. C. Chang, Y. E. Syu, H. C. Hung, Y. C. Hung, T. F. Young, D. S. Gan, and N. J. Ho, *Electrochem. Solid-State Lett.* **14**, K47 (2011).

¹⁴K. C. Chang, T. M. Tsai, T. C. Chang, Y. E. Syu, S. L. Chuang, C. H. Li, D. S. Gan, and Simon M. Sze, *Electrochem. Solid-State Lett.* **15**(3), H65–H68 (2012).

¹⁵G. G. Simeoni, T. Bryk, F. A. Gorelli, M. Krisch, G. Ruocco, M. Santoro, and T. Scopigno, *Nat. Phys.* **6**, 503 (2010).

¹⁶R. R. Das, P. Bhattacharya, W. Perez, R. S. Katiyar, and A. S. Bhalla, *Appl. Phys. Lett.* **81**, 880 (2002).

¹⁷K. Nadeem, T. Traussnig, I. Letofsky-Papst, H. Krenn, U. Brossmann, and R. Wuerschum, *J. Alloys Compd.* **493**, 385 (2010).