

Metal-Gate/High- κ /Ge nMOS at Small CET With Higher Mobility Than SiO₂/Si at Wide Range Carrier Densities

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Abstract—High-performance TaN/TiLaO/La₂O₃/SiO₂/(111)-Ge nMOSFETs show high mobility of 432 cm²/V · s at 10¹³ cm⁻² carrier density (N_s), good 1.05 junction ideality factor, and small subthreshold swing of 101 mV/dec, at a small 1.1-nm capacitance-equivalent thickness (CET). This is the first report of higher mobility in the Ge nMOSFET than SiO₂/Si universal mobility at wide medium-high N_s range and small CET of 1.1 nm, which is attributed to using the (111)-Ge substrate, 30-ns laser annealing, SiO₂ interfacial layer, and YbGe_x/n-Ge contact.

Index Terms—Annealing, Ge, high- κ , laser, YbGe, (111).

I. INTRODUCTION

LOW power operation at < 0.5-V drain voltage (V_D) is the most important criterion for advanced IC. To reach this goal, high-mobility new-channel CMOS and small source-drain contact resistance are needed for highly scaled < 10 nm CMOS. Thus, high-mobility InGaAs-nMOS/Ge-pMOS was proposed, but the epitaxial InGaAs on Si is still a challenge due to the lattice mismatch and antiphase boundaries. Alternatively, Ge has both higher bulk electron and hole mobility than Si for new-channel CMOS [1]–[17]. The integration of Ge on Si can be realized by using a Ge-on-insulator (GOI or GeOI) structure [1], where defect-free Ge was realized. This GeOI structure can also suppress the leakage current of small energy band-gap Ge and useful for device-level 3-D IC [5]. The all-Ge CMOS offers major merits of simpler process, lower cost, and potentially higher yield, but the Ge nMOSFET suffers from Fermi-level pinning, poorer source-drain contact, and faster mobility rolloff than universal SiO₂/Si data. The mechanism of fast mobility degradation is still unclear, although high peak mobility was reported for the GeO₂/Ge nMOSFET [14].

It was reported that the electron and hole mobility can be improved in (111) [14] and (110) [5] oriented Ge substrate. Here, we report higher mobility of Ge nMOSFET than SiO₂/Si data over wide medium-high carrier density (N_s) range and a small 1.1-nm capacitance-equivalent thickness (CET). Good

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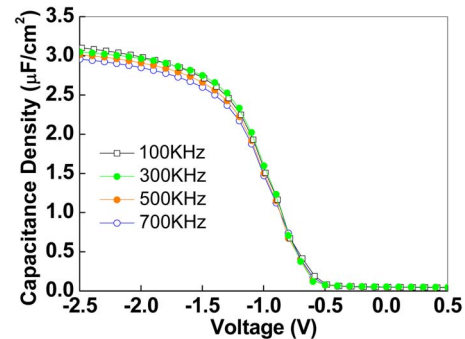


Fig. 1. $C-V$ characteristics of TaN/TiLaO/La₂O₃/SiO₂/(111)-Ge nMOS capacitors after LA.

device characteristics of better ideality factor and subthreshold swing (SS) were achieved, which is attributed to the (111)-Ge [14], 30-ns laser annealing (LA) [7], [16]–[20], SiO₂ interfacial layer to prevent electron wave-function penetration, and YbGe_x contact.

II. EXPERIMENT PROCEDURE

Standard 2-in p-type (111)-oriented Ge wafers with 10-Ω · cm resistivity were used in this study. In contrast to a previous *gate-last* process [16], the Ge nMOSFET was fabricated by a *gate-first* process. After cleaning, the samples were immersed into diluted HF (1 : 10) to remove the native oxide, dipped in de-ionized water, dried and loaded immediately into a deposition chamber. Then, ultrathin 0.7-nm SiO₂, 1-nm high- κ La₂O₃, and 4-nm high- κ TiLaO were deposited by physical vapor deposition [16], followed by O₂ postdeposition anneal (PDA) at 400 °C for 5 min and the first KrF LA. The LaTiO was deposited by mixed La₂O₃ : TiO₂ = 1 : 1, where a high κ of 58 was obtained in TiO₂ from a thick layer [21]. The LA has a short ~30-ns pulsewidth at a 248-nm wavelength. After that, the TaN gate was formed and self-aligned P was implanted at 35 keV and 5 × 10¹⁵ cm⁻², and the second KrF LA was applied for source-drain activation [16]. Finally, 30-nm Yb covered by SiO₂ was deposited to the source-drain and annealed at 500 °C to form the Ge nMOSFET. All the devices were measured at room temperature.

III. RESULT AND DISCUSSION

Fig. 1 shows the $C-V$ characteristics of TaN/TiLaO/La₂O₃/SiO₂/(111)-Ge nMOS capacitors formed by PDA and LA. A CET of 1.1 nm is obtained, which is one of the lowest CET for Ge nMOS. Such small CET is due to the higher κ TiLaO and 30-ns LA, to improve the oxide/Ge with lower

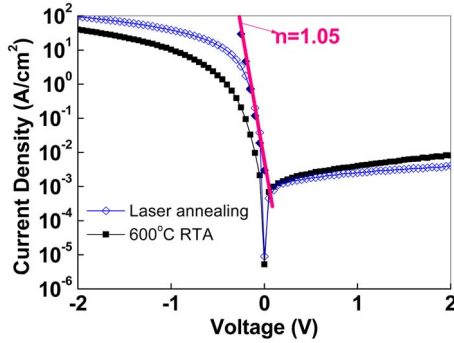


Fig. 2. J - V characteristics of $\text{YbGe}_x/\text{LA } n^+/\text{p-Ge}$ junction.

interface reaction and Ge out-diffusion, from previous cross-sectional transmission electron microscopy and secondary ion mass spectrometry study [12]. The slightly lower capacitance at higher frequency may be related to the gate stack leakage current. The negative flat-band voltage (V_{fb}) is important for low threshold voltage (V_t) nMOSFET, due to the interface dipole of La_2O_3 gate dielectric [22].

It was reported that the LA is an important technology to improve the source-drain junction characteristics for both Ge and Si MOSFETs [7], [16]–[20]. Fig. 2 shows n^+/p junction characteristics of P^+ -implanted p-Ge after $0.25\text{-J}/\text{cm}^2$ LA and YbGe_x contact. Small junction ideality (n) factor of 1.05, a low reverse leakage current, and $> 10^5$ on/off current ratio were obtained, where the n -factor is better than the previous 1.10 value using the same LA condition but Al contact [17]. This is the first report of better contact for the n^+/p Ge junction using low work-function metal [6]. A low specific contact resistivity of $4.4 \times 10^{-7} \Omega \cdot \text{cm}^2$ was obtained by the transfer length method, due to LA and YbGe_x contact.

Fig. 3(a) and (b) shows the I_d - V_g and mobility- N_s data of $\text{TaN}/\text{TiLaO}/\text{La}_2\text{O}_3/\text{SiO}_2/(111)\text{-Ge}$ nMOSFETs. Standard split C - V at 100 kHz was used for mobility extraction [17]. A low V_t of 0.13 V, a small SS of 101 mV/dec, and an interface trap density of $1.3 \times 10^{13} \text{ eV}^{-1} \cdot \text{cm}^{-2}$ were obtained. The low V_t near ideal $4kT/q$ (0.11 V) is due to the negative V_{fb} measured from the C - V characteristics shown in Fig. 1, and the small SS is attributed to higher gate capacitance and relatively good interface. From the mobility- N_s plot, good mobility of $432 \text{ cm}^2/\text{V} \cdot \text{s}$ at 10^{13} cm^{-2} N_s were obtained at 1.1-nm CET. The higher mobility than universal SiO_2/Si data at wide medium N_s is the important factor for low power operation toward $< 0.5\text{-V}$ V_D operation.

For comparison, other reported high mobility data were also plotted in Fig. 3(b) [8], [14]–[16]. It is important to notice that the mobility rolloff with increasing N_s in this letter is slower than the reported high- $\kappa/\text{GeO}_2/\text{Ge}$ nMOS [8], [14], [15]. One possibility of fast mobility degradation in the high- $\kappa/\text{GeO}_2/\text{Ge}$ nMOSFET at higher N_s is due to the interface roughness. The RMS surface roughness of control GeO_2/Ge grown at 400°C is 0.24 nm but becomes much poor to 6.6 nm at 500°C oxidation. In sharp contrast, the RMS roughness of $\text{TiLaO}/\text{La}_2\text{O}_3/\text{SiO}_2/\text{Ge}$ after LA is only 0.17 nm. Such very poor roughness may also degrade the mobility at low field.

Another possibility may be related to the small ΔE_C of the GeO_2/Ge interface [23], [24]. We further simulated the electron wave-function distributions in $\text{TaN}/\text{high-}\kappa/\text{GeO}_2/\text{Ge}$ and $\text{TaN}/\text{high-}\kappa/\text{SiO}_2/\text{Ge}$ nMOS at N_s of 10^{11} and 10^{13} cm^{-2} .

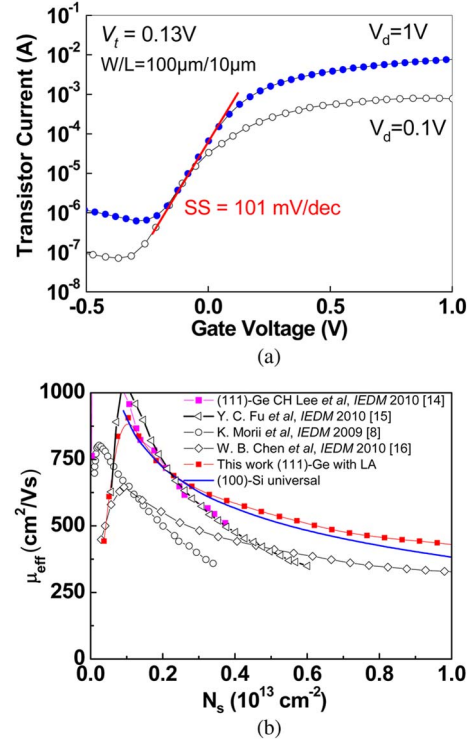


Fig. 3. (a) I_d - V_g and (b) mobility- N_s characteristics of $\text{TaN}/\text{TiLaO}/\text{La}_2\text{O}_3/\text{SiO}_2/(111)\text{-Ge}$ nMOSFETs using LA and YbGe_x contact.

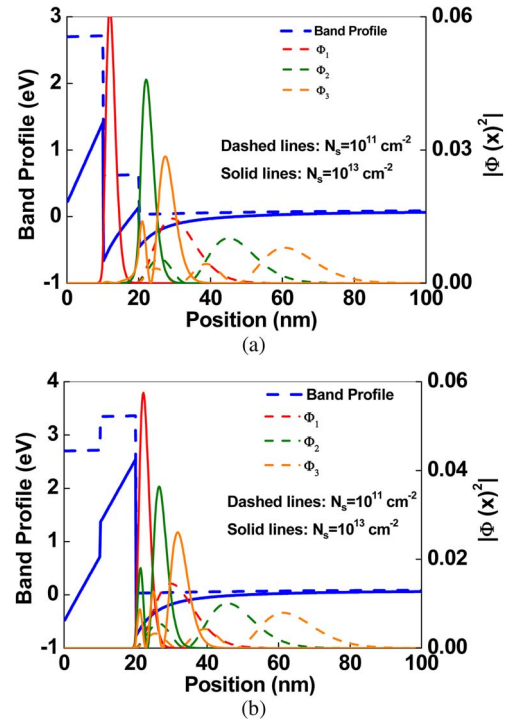


Fig. 4. Electron wave-function distributions in nMOS at inversion. (a) $\text{TaN}/10\text{-nm Al}_2\text{O}_3/10\text{-nm GeO}_2/\text{Ge}$. (b) $\text{TaN}/10\text{-nm Al}_2\text{O}_3/10\text{-nm SiO}_2/\text{Ge}$. The ΔE_C of 0.8 and 3.2 eV were used for GeO_2/Ge [23], [24] and SiO_2/Ge .

As shown in Fig. 4(a), severe penetration of ground-state electron wave function (Φ_1) in GeO_2 is found at higher N_s . In contrast, slight electron wave-function penetration in $\text{TaN}/\text{high-}\kappa/\text{SiO}_2/\text{Ge}$ is observed due to the large ΔE_C [see Fig. 4(b)]. Thus, the very high peak mobility and smooth interface, but having fast mobility rolloff, may be attributed to

the electron wave-function penetration into low-mobility GeO₂: the electron wave functions are in high-mobility Ge at lower N_s but pulled closer to the GeO₂/Ge interface with Φ_1 distribution in low-mobility GeO₂ at higher N_s .

IV. CONCLUSION

Higher mobility over wide N_s range than SiO₂/Si data, low SS of 101 mV/dec, small junction n -factor of 1.05, and small CET of 1.1 nm were reached in the TaN/TiLaO/La₂O₃/SiO₂/(111)-Ge nMOSFET.

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