Elucidating the Physical Property of the InGaN Nanorod Light-Emitting Diode: Large Tunneling Effect

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Abstract—The current–voltage (I-V) characteristics of the InGaN nanorod light-emitting diode (LED) are evaluated using nanoprobes installed on a field-emission scanning electron microscope. The saturated current of the InGaN nanorod LED $(I_O=2\times 10^{-9}~{\rm A})$ is found to be orders of magnitude higher than those obtained by downscaling a conventional InGaN LED to a chip size of $300\times 300~{\mu}{\rm m}^2~(I_{O1}=1\times 10^{-25}~{\rm A};~I_{O2}=1\times 10^{-14}~{\rm A})$. This observation is explained by the fact that the nanorod LED is associated with enhanced tunneling of injected carriers and, therefore, reduction of the defect-assisted leakage current and the diffusion–recombination process that normally occurs in InGaN LEDs.

Index Terms—Light-emitting diode (LED), nanorod, tunneling effect

I. INTRODUCTION

cal components of role in future optoelectronic devices [1], [2]. The possibility of quantum confinement [3], strain relaxation [4], and high crystallographic quality [5] make InGaN nanorods potential contenders for nanolaser [6], [7] and nano-LED [8]-[10], and provide a brand new approach for data storage, optical interconnects, and general lighting. Despite recent progress in nanofabrication techniques, a physical understanding of operating properties of nanodevices is still required to improve their performance and investigate their potential applications. In this paper, we will introduce the unique physical property observed in the nanorod LEDs. We investigate the current–voltage (I-V) characteristics of InGaN nanorod LEDs using nanoprobes installed on a field-emission scanning electron microscope (SEM). The saturation current $(I = 2 \times 10^{-9} \text{ A})$ and ideality factor ($n_{ideal} = 25$) of the nanorod LED were much larger than those of a conventional LED with a chip size of $300 \times 300 \ \mu \text{m}^2$. The implications of the differences between

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the electrical characteristics of nanorod and conventional LEDs are discussed. The effect of enhanced tunneling on forward current is suggested to explain this anomalous electrical behavior in the nanorod LED.

II. EXPERIMENT

An LED wafer was grown on c-axis sapphire substrates by low-pressure metal-organic chemical-vapor deposition. Following the growth of a GaN nucleation layer at 520 °C, a 1.5- μ mthick undoped GaN buffer layer and a 2.5-\(\mu\)m-thick Si-doped n-type $(5 \times 10^{17} \text{ cm}^{-3})$ GaN cladding layer were grown at 1050 °C. The active region consisted of five periods of 2.5 nm $In_{0.23}Ga_{0.77}N/10$ nm GaN multiquantum wells (MQWs, $\lambda =$ 450 nm) with Si doping in GaN barriers to reduce the quantumconfined stark effect. Finally, a 200-nm-thick Mg-doped p-type $(3 \times 10^{17} \text{ cm}^{-3})$ GaN cladding layer with a 50-nm-thick AlGaN electron-blocking layer was grown on the top at 1050 °C. To form nanorod arrays, a 5-nm-thick nickel (Ni) metal film was then deposited on the LED wafer by electron-beam evaporations. The sample underwent rapid thermal annealing at 850 °C for 60 s in ambient nitrogen to yield randomly distributed Ni metal islands with nanodimensions on the LED wafer surface, serving as a hard mask in the subsequent inductively coupled plasma (ICP) process. Then, 3 min of dry etching was conducted to ensure that the MQWs and n-type GaN nanorod arrays were exposed to the air. To form an ohmic contact, Ni nanodots were kept on the top of the nanorod arrays. Similarly, Ti/Al/Ti/Au layers were deposited and annealed at 600 °C to form ohmic contacts with an exposed n-type GaN layer.

The nanorod LED was then subjected to a Zyvex nanomanipulator system to determine the I-V characteristics. A conventional LED counterpart with a typical chip size of 300 × $300 \,\mu\text{m}^2$ was also fabricated for comparison. Fig. 1(a) schematically depicts the nanoprobe setup, which includes nanorod LED arrays, a nanotip, an n-electrode pad (Ti/Al/Ti/Au), and Ni nanodots. Therefore, the nanotip with an internal diameter of 50 nm in the SEM enabled a p-contact to be made directly on the top of the nanorods, while another tip was located at the n-electrode, serving as the second contact, as displayed in Fig. 1(b). This setup was, therefore, able to measure the I-V characteristics of an individual nanorod. Fig. 1(c) presents an SEM image of a nanotip that is touching the top surface of a single nanorod. In this figure, the scale bar represents 100 nm. In this investigation, the mean diameters and the lengths of the nanorods are 80–100 and 350 nm, respectively. The general morphology and exterior

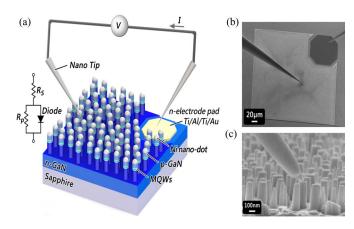


Fig. 1. (a) Probing test of a single nanorod LED performed using nanoprobes installed on the field-emission SEM. (Inset) Effective electrical circuit of the nanorod LED. (b) Electrical connection to a nanorod via two nanotips on the field-emission SEM. (c) Representative SEM image of nanotip in contact with the top of nanorod.

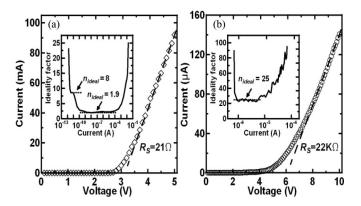


Fig. 2. I-V curves on linear-scale for (a) conventional and (b) single nanorod LEDs. Insets: Ideality factors against injected currents of conventional and single nanorod LEDs.

appearance of the nanorods in the SEM images are strongly determined by the annealing conditions of the Ni film and the ICP etching process [11]. The inset in Fig. 1(a) also presents the effective electrical circuit of an LED, where R_S and R_P represent the series and parallel resistances, respectively. Generally, R_S is determined by excessive contact resistance or by the resistance of the neutral regions, and R_P can be caused mainly by damage to junction regions during the ICP process or by surface imperfections. These parasitic resistances in the nanorod were taken into account in the subsequent calculation.

III. RESULTS AND DISCUSSION

Fig. 2 plots the forward-bias I-V curves on a linear scale for (a) conventional and (b) single nanorod LEDs, respectively. These I-V characteristics were collected and averaged from ten conventional and nanorod LEDs. The R_S values, determined from these figures, are 21 Ω and 22 k Ω for conventional and single nanorod LEDs, respectively.

 R_S in the single nanorod LED is orders of magnitude higher than in the conventional LED primarily because of the small feature size and the high contact resistance of the contact between the nanotip and the nanorod, resulting in a higher turnon voltage for the single nanorod (~6.1 V) than of the conventional LED (\sim 3.2 V). The insets in Fig. 2 display the ideality factors (n_{ideal}) of both LED samples obtained from the measurements using $n_{\text{ideal}} = (q/kT)((\partial \ln I/\partial V)^{-1})$ [12]. Clearly, in the inset in Fig. 2(a), the two distinct ideality factors $n_{\rm ideal} = 8.0 \ (3.7 \times 10^{-12} \ {\rm A} < I < 1 \times 10^{-11} \ {\rm A})$ and $n_{\rm ideal} = 1.9 \ (8.0 \times 10^{-10} \ {\rm A} < I < 5.0 \times 10^{-5} \ {\rm A})$ were determined for the conventional LED. Cao et al. found that in the InGaN-based LED, carrier recombination in the depletion region via dislocation defects, and therefore, leakage currents associated with deep-level traps, causes the ideality factor to exceed 2.0 [13]. In addition, increasing the forward bias causes the defect-assisted leakage current to reach its maximum and then saturate. Consequently, the diffusion-recombination current begins to flow through the depletion region and dominate the LED, yielding an ideality factor of between 1.0 and 2.0. Basically, the conventional LED acts as described. However, the I-V characteristic of the nanorod LED reveals completely different rectification behavior. For a single nanorod LED, a single value of the ideality factor $n_{\text{ideal}} = 25.0 \ (9.0 \times 10^{-7} \ \text{A} < I < 5.5 \times 10^{-6} \ \text{A})$ can be derived, as presented in the inset in Fig. 2(b). Importantly, such a large ideality factor is not likely to be attributable to the defect-assisted leakage current because the dislocation defect is barely detected for the nanorod LED with a diameter less than 100 nm, suggesting that in the nanorod LED, mechanisms of carrier recombination other than the defect-assisted leakage mechanism apply.

To analyze further the I-V properties of the conventional LED, the Shockley equation was modified to take into account the diffusion–recombination and defect-assisted leakage currents, as well as the parasitic effect, which is shown in the diode-circuit model in the inset in Fig. 1(a) [14]. Accordingly, the total injected current of LEDs I is given by

$$I = I_P + I_J = I_P + I_{\text{diffusion}} + I_{\text{leakage}} \tag{1}$$

where I_P represents the current that bypasses the junction region and is given by $I_P = (V - IR_S)/R_P$, I_J is the current that flows across the junction, and comprises the diffusion–recombination current $I_{\rm diffusion}$ and the defect-assisted leakage current $I_{\rm leakage}$. Both $I_{\rm diffusion}$ and $I_{\rm leakage}$ are given by

$$I_{\text{diffusion}} = I_{O1} \left(\exp \left(\frac{e(V - IR_S)}{n_1 kT} \right) - 1 \right)$$
 (2)

$$I_{\text{leakage}} = I_{O2} \left(\exp \left(\frac{e(V - IR_S)}{n_2 kT} \right) - 1 \right)$$
 (3)

where I_{O1} and I_{O2} are saturation currents produced by the diffusion–recombination and the defect-assisted leakage processes, respectively, and n_1 and n_2 are the ideality factors for the diffusion–recombination and the defect-assisted leakage processes, respectively. Substituting (2) and (3) into (1) and considering $I_P = (V - IR_S)/R_P$, enables the I-V curve

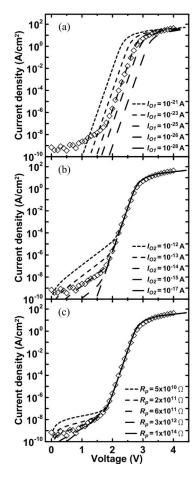


Fig. 3. J-V curve of the conventional InGaN LED on semi-log scale and theoretically fitted using $\ (4)$ with adjusted variables (a) $I_{O\,1}$, (b) $I_{O\,2}$, and (c) R_P .

of an InGaN LED to be rewritten as

$$I = A \cdot J = I_{O1} \left(\exp\left(\frac{e(V - IR_S)}{n_1 kT}\right) - 1 \right)$$

$$+ I_{O2} \left(\exp\left(\frac{e(V - IR_S)}{n_2 kT}\right) - 1 \right) + \frac{(V - IR_S)}{R_P}$$
 (4)

where A and J denote the mesa area and the injected current density of the LED, respectively.

Fig. 3 plots the J-V curve of the conventional InGaN LED on a semilog scale and theoretically fits it using (4) with variables I_{O1} [Fig. 3(a)], I_{O2} [Fig. 3(b)], and R_P [Fig. 3(c)]. Clearly, adjusting the three parameters I_{O1} , I_{O2} , and R_p enables the experimental J-V curve of the conventional InGaN LED to be well fitted. In Fig. 3, $n_1=1.9,\ n_2=8$, and $R_S=21\ \Omega$, obtained experimentally from Fig. 2(a), are used. Notably, according to Fig. 3(b), the diffusion–recombination process becomes less important as I_{O2} increases, indicating that although a relatively large ideality factor of $n_2=8$ accompanies the defect-assisted leakage process, as long as the density of dislocation defects is large enough, then most of the injected carriers will recombine promptly at the deep-level traps that are associated with the dislocation defects, without any diffusion–recombination process. Restated, the saturation current is an important index of the dom-

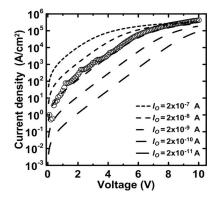


Fig. 4. J-V curve of InGaN nanorod LED on semilog scale, theoretically fit using $n_{\rm ideal}=25$ and $R_S=22~{\rm k}\Omega.$

ination of the recombination of injected carriers in an InGaN LED. Moreover, as displayed in Fig. 3(c), in the region to which a small forward bias (<1 V) is applied, the parallel resistance (R_P) dominates the leakage current. The calculated saturation currents associated with the diffusion–recombination and the defect-assisted leakage processes are $I_{O1} = 1 \times 10^{-25}$ A and $I_{O2} = 1 \times 10^{-14}$ A, respectively. The estimated parallel resistance of the conventional InGaN LED is $R_P = 6 \times 10^{11} \ \Omega$.

Similarly, Fig. 4 plots the J-V curve of the nanorod LED in a semilog scale. Clearly, one major recombination process of injected carriers dominates the J-V curve of the nanorod LED. No diffusion–recombination current (1.0 $< n_{ideal} < 2.0$) is observed in Fig. 4, since the rectification behavior of the nanorod LED exhibits an extremely large ideality factor of $n_{\rm ideal} = 25$. Again, the experimental J-V curve was fitted using (4), and therefore, the diffusion—recombination of injected carriers was neglected. In addition, the experimental J-Vcurve can be fitted directly without taking R_P into account, revealing that no injected carriers are leaked or bypass the junction region in the nanorod LED, which will be illustrated more clearly later in the paper. The calculated saturation current of the nanorod LED is $I_O = 2 \times 10^{-9}$ A, determined using $n_{\rm ideal}=25$ and $R_S=22$ k Ω , and is five orders of magnitude larger than $I_{O2}=1\times 10^{-14}$ A. This result is consistent with our earlier finding that the defect-assisted leakage current did not dominate in the nanorod LED, and that other recombination processes of injected carriers must occur in the nanorod LED.

To further examine the issue of leakage current of our LED samples, Fig. 5 plots the J-V curve on a semi-log scale from -10 to 2 V for (a) conventional and (b) single nanorod LEDs, respectively. In Fig. 5(a), the J-V curve of the conventional LED can be clearly divided into three regions: region I for the applied bias-voltage less than -5.8 V, region II for the applied bias-voltage between -5.8 and 1 V, and region III for the applied bias-voltage larger than 1 V. In region I, the measured reverse current density increases exponentially with the increasing of the reverse voltage, entirely due to the breakdown of the conventional LED. The breakdown voltage of the conventional LED (marked by an arrow) is around -5.8 V. In region II, the J-V curve is primarily dominated by R_P of the conventional LED and can be well explained by the Ohmic law ($V = I \bullet R_P$, red

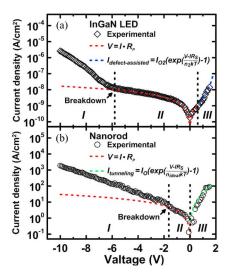


Fig. 5. J-V curves on a semilog scale from -10 to 2 V for (a) conventional and (b) single nanorod LEDs.

dashed-line). Notably, the measured J-V curve of the conventional LED is symmetrical for the applied bias-voltage from -1 to 1 V, indicating R_P still dominates the injected carriers as a small forward bias is applied (<1 V). Restated, R_P is mainly caused by the damage of junction regions and by the surface recombination associated with crystalline imperfection. The calculated R_P and measured leakage current at -5 V of the conventional LED are $6 \times 10^{11} \Omega$ and $1 \times 10^{-8} \text{ A/cm}^2$, respectively. Finally, in region III (>1 V), the carrier recombination via trap levels associated with dislocation defects begins in the LED's depletion region and becomes dominated (blue dashed-line), resulting in the ideality factor of $n_{\text{ideal}} = 8$, as previously discussed in reference to Fig. 2(a).

Similarly, as shown in Fig. 5(b), the measured J-V curve of the nanorod LED can also be divided into three regions. Accordingly, the measured reverse current density increases rapidly as the reverse voltage increases in region I (<-1.7 V). Again, it is because of the breakdown of the nanorod LED. The breakdown voltage of the nanorod LED (marked by an arrow) is around -1.7 V. In addition, as compared to that of the conventional LED, the smaller gradient of the J-V curve of the nanorod LED in region I is mainly attributed to its inherently small value of R_P . For the applied bias-voltage between -1.7and 0 V (region II), the J-V curve of the nanorod LED is dominated by its R_P . The calculated R_P of the nanorod LED in region II is around $1 \times 10^9 \Omega$. As compared to the conventional LED, the surface recombination events in the nanorod LED with large surface-to-volume ratio will in principle lead to a large leakage current, resulting in a much smaller R_P . Furthermore, such a small R_P of the nanorod LED also affects its J-V curve in region I, as one can observe that the breakdown behavior of the nanorod LED looks analogous to the calculated $V = I \bullet R_P$ curve (red dashed-line). In region III (>0 V), the J-V curve of the nanorod LED is not symmetrical to which a reverse bias is applied, implying R_P has no influence on the injected carriers in this region. In contrast, the measured forward current increases exponentially with the

increasing of the forward voltage and can be described well by $I_{\text{tunneling}} = I_o(\exp((e(V - IR_S))/n_{\text{ideal}}kT) - 1)$ (green dashed-line), as previously discussed in reference to Fig. 4. Importantly, it indicates the injected carriers are indeed flowing through the junction region of the nanorod LED, even when only a small forward voltage is supplied. We believe the aforementioned observation is mainly attributed to a significant enhancement of tunneling effect on the injected carriers in the nanorod LED. The tunneling effect, enhanced by the nanoscale features of the nanorod LED, induces the transport of injected carriers through the junction region of the nanorod LED. Moreover, for the nanorod LED, this tunneling of injected carriers is extensive as no parasitic effect or diffusion-recombination process is observed under the measuring conditions. Consequently, the saturated current of the nanorod LED is specifying degree of significance enhanced, leading to an unusually and extremely large ideality factor of $n_{\text{ideal}} = 25$.

IV. CONCLUSION

In conclusion, this study elucidates the I-V characteristics of InGaN single nanorod LEDs. Based on a comparison of the conventional InGaN LED with a chip size of $300 \times 300~\mu\text{m}^2$ and the Shockley equation modified to consider the diffusion–recombination and defect-assisted leakage currents as well as the parasitic effect, this study determines that the effect of enhanced tunneling on the injected carriers is mainly responsible for the experimentally observed large saturated current in the InGaN nanorod LED. Results of this study concerning the electrical property of the nanorod LED significantly contribute to the efforts to manufacture nanoscale photonic and optoelectronic devices.

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