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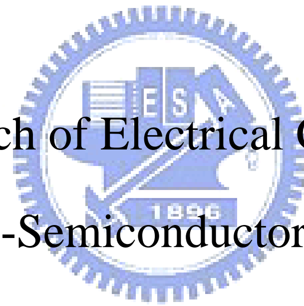
碩 士 論 文

金屬矽化物-高介電常數介電質-半導體場效應電晶體之電性研究



The Research of Electrical Characteristics of High-κ Dielectric -Semiconductor Field-Effect Transistor

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摘要

隨著超大型積體電路技術的不斷發展、元件尺寸的微縮，閘極介電質的厚度必須降低以維持電容值。但漏電流隨著厚度減少而不斷的增加，這會與現代科技中隨處可見的手提裝置理念產生衝突。為了獲得較低的操作電壓，金屬閘極將會取代傳統複晶矽閘極。

在本篇論文中，我們改善金屬矽化物—高介電常數介電質—半導體場效應電晶體的電性。在實驗中我們獲得了好的特性，P(N)型電晶體等效功函數 4.95eV(4.25eV)，以及 1.6nm 的 EOT。此外也量測到低的臨限電壓及好的遷移率。因閘極/高介電常數介面多一層 Si，而使穩定性提高到 1000℃。雖然仍有費米能階窄化的效應，但整個製程能在現有的技術上使用。

The Research of Electrical Characteristics of FUSI Gate-High-κ Dielectric-Semiconductor Field-Effect Transistor.

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Abstract

As the very large scale integration (VLSI) technology continues to be scaled down, the thickness of gate dielectric has to be decreased for maintaining the capacitance value and drive levels. The gate leakage current increases with decreasing thickness, and the phenomenon is counter to the mobile device in the technology node. In order to obtain small threshold voltages, we would replace poly-Si gate by metal gate.

In this thesis, we improve the electrical characters of FUSI Gate-High $-\kappa$ gate dielectric-semiconductor MOSFET. We obtain good performance of proper effective work function of 4.95eV (4.25eV) for p- and n- MOSFET respectively, and about 1.6nm EOT. On the other hand, small threshold voltage and good mobility have also been measured. The

thermal stability is up to 1000℃ due to the inserted Si. Unfortunately, the Fermi-Level pinning effect still occurs. However, the whole process can be used in the factory.

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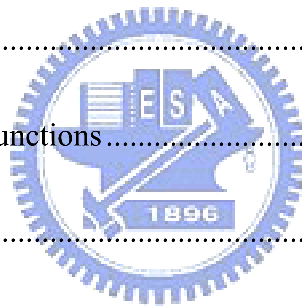

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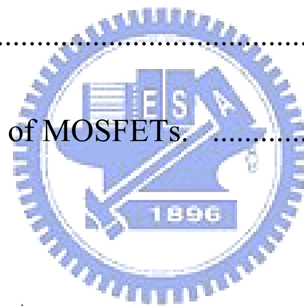

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CHAPATER 1

Introduction

1.1 Motivation to Study High-κ dielectric

1.1.1 Introduction

The complementary metal oxide semiconductor (CMOS) field effect transistor (FET) made from silicon is the most important electronic device. This has arisen because of its low power consumption and because of its performance improvement over forty years according to Moore's Law of scaling. This law notes that the number of devices on an integrated circuit increases exponentially, doubling over a 2–3 year period [1-1]-[1-2]. The minimum feature size in a transistor decreases exponentially each year. The semiconductor Roadmap defines how each design parameter will scale in future years to continue this trend, as shown in figure 1-1. The thickness of the SiO_2 layer presently used as the gate dielectric is now so thin (under 1.4 nm) that the gate leakage current due to direct tunneling of electrons through the SiO_2 becomes too high, exceeding $1Acm^{-2}$ at 1V, so that power dissipation increases to unacceptable values. In addition it becomes increasingly difficult to make and measure accurately such thin films. Finally, the reliability of SiO_2 films against electrical breakdown declines in thin films. These reasons lead to a desire to replace SiO_2 as a gate oxide. Low standby power CMOS requires a leakage current of below 1.5×10^{-2} Acm^{-2} rather than just $1Acm^{-2}$ [1-1]. There have been many difficulties in manufacturing high K oxide layers of

sufficient quality but these have gradually been overcome. To be good substitutes for SiO_2, the high-κ materials must have several advanced features in addition to the high-κ value. They should be chemically stable with Si substrate and the gate electrode and they should be thermally stable at temperatures no less than 500℃. Moreover, they should have good interface properties with the Si substrate so that the structure can have low interface trap density, high channel mobility, low oxide trap density, large bandgap, and large band offset energies [1-4].

1.1.2 Choice of High K oxide

Silicon dioxide is for years due to the following advantages. As a semiconductor, Si has an average performance, but in most aspects SiO_2 is an excellent insulator. SiO_2 can be made from Si by thermal oxidation, but every other semiconductor (Ge, GaAs, GaN, SiC, etc) which must be made by deposition has a poor native oxide. SiO_2 is amorphous, has very few electronic defects and forms an excellent interface with Si. It can be etched and patterned to a nanometer scale. Its only problem is that it is possible to tunnel across it when very thin. Hence, we must lose these advantages of SiO_2 and start to use a new high K oxide [1-8]. The useful gate dielectrics should meet the following fundamental requirements:

1. Thermodynamic stability on silicon with respect to formation of SiO_2 and MSi_x.

2. Amorphous after device integration implies that the dielectrics should remain amorphous after S/D or elevated temperature activation.

3. Low conduction for low leakage and low power consumption. For metal oxides (MO_x), it is well known that bandgap is inversely related to κ value (the aluminum oxide as an exception). Low leakage current implies large band-offset for electrons and holes.

4. High carrier mobility at the dielectric/Si interface. Therefore, the low D_{it} and low bulk charges (low effective fixed charges) are requirements.

5. High breakdown strength and acceptable reliability. The breakdown strength is inversely related to κ value for metal oxides.

1.1.3 *K* Values and Thermal Stability

The oxide's *K* value should be over 12, preferably 25–30. There is a trade off with the band offset condition, which requires a reasonably large band gap to reduce gate leakage currents. Table 1-1 and figure 1-2 show that the *K* of candidate oxides varies inversely with the band gap. In fact, a very large *K* is undesirable in CMOS design because of large fringing fields at the source and drain electrodes [1-5].

The oxide must not react with Si to form either SiO_2 or a silicide according to the unbalanced reactions

$$MO_2 + Si \rightarrow M + SiO_2 \quad (1)$$

$$MO_2 + 2Si \rightarrow MSi + SiO_2 \quad (2)$$

This is because the resulting SiO_2 layer would increase the EOT and negate the effect of using the new oxide. In addition, any silicide formed by (2) is metallic and would short out

the field effect. An interfacial layer of SiO_2 often exists between the Si channel and the high κ oxide layer [1-5~1-8]. There are advantages and drawbacks to this interfacial layer, as long as its presence and thickness can be controlled. The overall EOT of a layer 1 of SiO_2 and a layer 2 of high κ oxide is given by the series capacitance formula (3), which becomes formula (4).

$$1/C=1/C_1+1/C_2 \quad (3)$$

$$EOT=t_{SiO_2}+EOT_{hi\kappa} \quad (4)$$

1.1.4 Crystalline or amorphous oxides

Unlike silicon oxide which is amorphous up to 1100℃ because of its low coordination covalent bonds, High−κ transition metal oxides are generally poor glass former. This is because the metal oxide bonding is normally a high coordination ionic bond with the d-state electrons and the oxide films crystallize easily at low temperatures. For both HfO_2 and ZrO_2, crystallization temperature was expected to be above 900℃. However, the real crystallization temperature is much lower of expected values. This difference may contribute to impurities or some sort of nucleation centers. Thus following post-deposition thermal treatment (PDA) will result in a certain degree of local crystallization. Nevertheless, structural defects in as-deposited amorphous films can be removed through thermal annealing in oxygen-containing ambient. Large amount of shallow oxide traps at the grain boundaries of the crystalline phase will be introduced at the same time. As a result, crystallization will give rise to a large leakage current because of large amount grain boundary traps [1-8].

1.1.5 Band offset

In order to decrease the leakage current, this requires that the potential barrier at each band must be over 1 eV in order to inhibit conduction by the Schottky emission of electrons or holes into the oxide bands, as shown schematically in figure 1-3. However, the reported high-κ materials usually have large leakage current. The large leakage current can be partially attributed to the small conduction band offset energy with respect to the silicon. The small conduction band offset does not only result in large gate direct-tunneling or Fowler–Norheim (FN) current but also give rise to large hot-carrier emission into the gate insulator. This limits the choice of oxide to those with band gaps over 5 eV. Only few materials like Al_2O_3, ZrO_2, HfO_2, Y_2O_3, and La2O3 and various lanthanides and their silicates and aluminates satisfy these requirements. These oxides also have excellent thermal stability. This is because a high heat of formation correlates with a wide band gap in ionic compounds [1-8].

1.2 Motivation to Study Metal Gate

1.2.1 Introduction

Considerable challenges are encountered when bulk CMOS devices are scaled into the sub-100 nm regime for higher integrated circuit (IC) density and performance. The problems of polysilicon (poly-Si) gate depletion, high gate resistance, high gate tunneling leakage current, and boron penetration into the channel region become more severe as the channel length and gate-oxide thickness are aggressively reduced. In present CMOS, the gate electrodes are not real metals but polycrystalline Si doped highly n-type or p-type, respectively, for NMOS and PMOS. Their workfunctions are 4.05 eV and 5.15 eV, respectively, just as those required. The gate metals to be used must be 'band edge metals', with workfunctions equal to the band edge energies of Si, 4.05 and 5.15 eV [1-17].

1.2.2 The Choice of Metal Gates

As the device continuously scaling down, we encountered a lot of difficulties. Such gate electrodes have problems of:

(1) Boron penetration into channels through thin gate dielectric.

(2) High gate resistance.

(3) Polysilicon gate depletion.

(4) High gate tunneling leakage current.

The possible solution is to use the proper metal gate with proper work function. The

parasitic series capacitance due to a depletion in poly crystalline-silicon gates poly gates reduce the gate capacitance and drive current. The metal gate electrode will make poly-depletion free due to the poly depletion will reduce the capacitance and contribute a degradation to EOT in inversion state. The work functions (Φ_m) of metal play an important role for metal-gate/high-κ MOSFET and are shown in Fig. 1-4. Metal gates with workfunctions near the conduction and valence band edges of Si are used in N- and P-MOSFETs respectively. To achieve the desired dual-metal gate workfunctions on high-κ gate dielectrics, candidate metals need to have vacuum workfunctions smaller (larger) than 4.05eV (5.17 eV) for the NMOS (PMOS). For the PMOS gate, inert metals must be used and this makes gate etching particularly challenging. Reactive metals have to be used as the NMOS gate, and this might introduce extrinsic interface states due to defects arising from an interfacial reaction [1-6]. There are more metal candidates for NMOSFET (Ti, Al, and Ta) than for PMOSFET. Among the possible candidates for PMOSFET, Pt and **Ir** are very difficult to etch using plasma processes.

On the other hand, thermal stability of the effective metal electrode and metal diffusion are also important considerations. Recently, lots of metal or metal-nitride materials have been widely researched and successfully intergraded in advanced CMOSFET's, such as TiN, TaN, Pt, Mo and Ir . Tantalum (Ta) has a work-function close to n^+ poly-Si. Tantalum nitride (TaN) is quite stable (to maintain thermal stability up to a 1000℃ RTA) because the activation

energy of metal and nitrogen is relatively low. Tantalum is bonded tightly within nitride and no obvious diffuse was observed in fabricated devices. However, TaN gate on high-κ HfO_2 shows a significant shift of flat band voltage (V_{FB}) toward the mid-gap of Si due to the interface reaction between the TaN and HfO_2 at the high temperature. This is called the "Fermi-level pinning effect." Therefore, the Fermi-level pinning effect needs to be avoided by selecting suitable metal gate and high-κ materials for advanced MOSFETs [1-7]-[1-9].

1.3 The Study of Fermi Level Pinning

1.3.1 Introduction

In order to decrease the threshold voltage with the device scaling down, we have to choose metals with proper workfunctions carefully. The identification of metal-gate materials is very difficult because metal-gate workfunctions are observed to vary with different gate dielectrics and the process parameter. These intrinsic states are predominantly donor-like close to E_v, and mostly acceptor-like near E_c as shown in Fig. 1-5. Charge transfer generally occurs across the interface due to the presence of intrinsic interface states. Charging of these interface states creates a dipole that tends to drive the band lineup toward a position that would give zero dipole charge. Figure 1-5 illustrates the case where the metal Fermi level $\Phi_{F,m}$ is above the charge neutrality level in the dielectric $E_{CNL,d}$, creating a dipole that is charged negatively on the dielectric side. This interface dipole drives the band alignment so that $E_{F,m}$ goes toward $E_{CNL,d}$ and the effective metal work function, $\Phi_{m,eff}$ therefore differs from the vacuum metal work function $\Phi_{m,vac}$. This work function change is proportional to the difference between $E_{CNL,d}$ and $E_{F,m}$, or, equivalently, the difference between $\Phi_{m,vac}$ and $\Phi_{CNL,d}$ [= $(E_{Vac}-E_{CNL,d})/q$]. For some gate materials, there are no defects in the interface, and the workfunctions are determined by intrinsic states. According to reported paper, the workfunctions with Fermi-Level pinning effect are affected by annealing temperature. The Fermi level pinning effect is more obvious at higher temperature. The workfunctions would converge at higher

temperature [1-8]-[1-10].

1.3.2 The Effective Workfunctions

We will discuss about the problems of the workfunctions shifts in the chapter. The workfunctions in vacuum are used as references. In the Schottky limit and without fixed charges, the flat band voltage of a MOS capacitor is given by equation (5). Inverting this equation, an effective workfunction of the gate metal ($\Phi_{m,eff}$) can be derived from the measured flat band voltage of the CV plot of the MOS capacitor, by referencing to the by referencing to the workfunction Φ_s of the Si substrate, 4.05V or 5.15V for a n-type or p-type Si, respectively. We defined a pinning factor as the change of flat band voltage divided by the change in the metal's vacuum workfunction as equation (7). Linearizing this model leads to another definition of effective workfunction, $\Phi_{m,eff}$, as shown in equation (8). In equation (8), $\Phi_{m,vac}$ is the vacuum or true workfunction of metals and $\Phi_{CNL,d}$ is the CNL energy of the oxide, measured from the vacuum level.

$$V_{FB} = \Phi_m - \Phi_s \quad (5)$$

$$\Phi_{m,eff} = V_{FB} + \Phi_s \quad (6)$$

$$S = d\Phi_{FB}/d\Phi_m \quad (7)$$

$$\Phi_{m,eff} = \Phi_{CNL,d} + S(\Phi_{m,vac} - \Phi_{CNL,d}) \quad (8)$$

S is a slope parameter that accounts for dielectric screening and depends on the electronic component of the dielectric constant ε_∞ .The factor decreases as the dielectric permittivity becomes larger. The slope parameter S obeys an empirical relationship given by

equation (9). Materials with a smaller S tend to have obvious Fermi-Level pinning and the effective workfunction would approach $\Phi_{CNL,d}$, and the maximum value for S is unity, which indicates that no pinning of the metal Fermi level occurs [1-8]-[1-10].

$$S = \frac{1}{1 + 0.1(\varepsilon_\infty - 1)^2} \tag{9}$$

1.4 Fully-Silicided Gates

High-performance CMOS technology generally requires two different workfunctions for n-MOS and p-MOS devices. This condition results in complex gate stacks and complex process. Only one kind of gate dielectrics deposited is basically required. One of the approaches uses a single metal layer, which sets workfunction for one type of transistor and an alloy of the same metal formed from an additional layer for the other transistor. An excellent technique, compatible with very large scale integrated VLSI CMOS technology of forming tunable metal gates by full silicidation FUSI of doped polysilicon gates, was recently demonstrated. The ability to form a pileup of dopant at the silicide/dielectric interface and thus to modulate the gate electrode workfunction appears to be strongly related to the silicidation conditions [1-11]. It is widely reported that the silicidation temperature and the ratio of Si and Ni thickness, affect the phase of the FUSI gate. Formation of dopant pileup in front of advancing silicide and ultimately at the silicide/gate dielectric interface is depend on what the diffuse species are. Virtually immobile dopant atoms in silicon at silicidation temperatures pile-up at the silicide front when silicon is the moving specie. Ni-rich silicides such as Ni_2Si form by Ni atoms moving into Si, thus there is no dopant pileup. The pileup formed during NiSi formation implies that Si is moving specie during process. This phenomena is counter to Ni reacted with undoped Si. We would suggest that the presence of immobile dopant may be the key point of the FUSI gates. Several groups have been recently

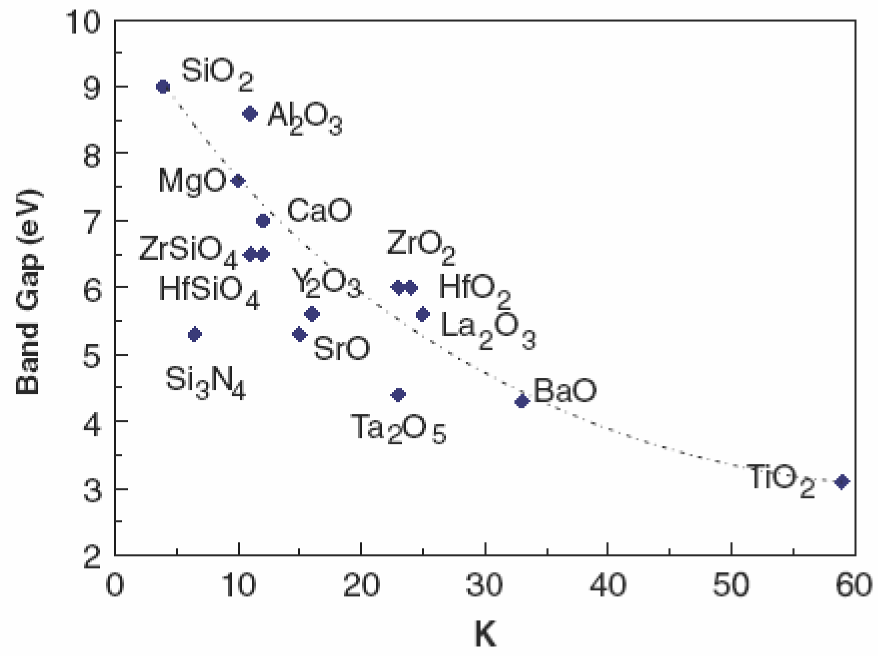
studied FUSI gates on high-*k* dielectrics. Metal gates are proved to decrease severe Fermi level pinning observed on polysilicon/Hf-based dielectric stacks, believed to have been caused by Si-Hf interaction. No pinning was also reported for NiSi-HfAlNO system and for NiSi on HfO_2. However, some studies show that FUSI NiSi does exhibit the pinning. It has been reported for NiSi gates on Hf silicate, NiSi and PtSi on HfO_xN, and for NiSi on HfSiON. So we have to improve the stack formation process, and particularly the type and quality of interface between metal and dielectric to have better performance [1-11]-[1-15]. In Fig. 1-6, we summarize all the challenges of MOSFETs mentioned in Chapter 1.

Materials	SiO_2	Si_3N_4	Al_2O_3	$HfSiO_4$	ZrO_2	HfO_2	La_2O_3
κ	**3.9**	**7**	**9**	**11**	**25**	**25**	**30**

Table.1-1 Static dielectric constant (*K*) of various materials.

Year of Production	2007	2008	2009	2010	2011	2012	2013	2014	2015
MPU/ASIC Metal 1 (M1) ½ Pitch (nm)(contacted)	68	59	52	45	40	36	32	28	25
MPU Physical Gate Length (nm)	25	22	20	18	16	14	13	11	10
L_g: Physical Lgate for High Performance logic (nm) [1]	25	22	20	18	16	14	13	11	10
EOT: Equivalent Oxide Thickness [2]									
Extended planar bulk (Å)	11	9	7.5	6.5	5.5	5			
UTB FD (Å)				7	6	5.5	5	5	5
DG (Å)					8	7	6	6	6
Gate Poly Depletion and Inversion-Layer Equivalent Thickness [3]									
Extended Planar Bulk (Å)	7.4	3.1	2.9	2.8	2.7	2.8			
UTB FD (Å)				4	4	4	4	4	4
DG (Å)					4	4	4	4	4
EOT_{elec}: *Electrical Equivalent Oxide Thickness in inversion [4]*									
Extended Planar Bulk (Å)	18.4	12.1	10.4	9.3	8.2	7.8			
UTB FD (Å)				11	10	9.5	9	9	9
DG (Å)					12	11	10	10	10
$J_{g,limit}$: *Maximum gate leakage current density [5]*									
Extended Planar Bulk (A/cm^2)	8.00E+02	9.09E+02	1.00E+03	1.11E+03	1.25E+03	1.43E+03			
UTB FD (A/cm^2)				1.11E+03	1.25E+03	1.43E+03	1.54E+03	1.82E+03	2.00E+03
DG (A/cm^2)					1.25E+03	1.43E+03	1.54E+03	1.82E+03	2.00E+03
V_{dd}: *Power Supply Voltage (V) [6]*									
Extended Planar Bulk (V)	1.1	1	1	1	0.95	0.9			
UTB FD and DG (V)				1	1	0.9	0.9	0.9	0.8
$V_{t,sat}$: *Saturation Threshold Voltage [7]*									
Extended Planar Bulk (mV)	134	94	94	103	101	112			
UTB FD (mV)				103	99	87	93	99	99
DG (mV)					116	106	103	108	111
$I_{sd,leak}$: *Source/Drain Subthreshold Off-State Leakage Current [8]*									
Extended Planar Bulk (μA/μm)	0.34	0.71	0.70	0.64	0.74	0.88			
UTB FD (μA/μm)				0.33	0.52	0.62	0.58	0.55	0.60
DG (μA/μm)					0.2	0.34	0.37	0.38	0.38
$I_{d,sat}$: *NMOS Drive Current [9]*									
Extended Planar Bulk (μA/μm)	1211	1513	1639	1807	1824	1782			
UTB FD (μA/μm)				1948	2000	1944	2109	2245	2030
DG (μA/μm)					1917	1943	2204	2385	2295
Mobility enhancement factor due to strain [10]	1.8	1.8	1.8	1.8	1.8	1.8	1.8	1.8	1.8
$I_{d,sat}$ *enhancement factor due to strain [11]*									
Extended Planar Bulk	1.09	1.08	1.08	1.08	1.09	1.08			
UTB FD				1.07	1.08	1.08	1.08	1.06	1.05
DG					1.04	1.04	1.04	1.03	1.03

Fig. 1-1 The ITRS Roadmap for semiconductor 2007.



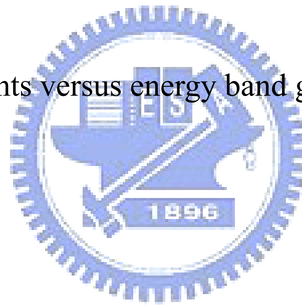


Fig. 1-2 Static dielectric constants versus energy band gap for candidate materials.

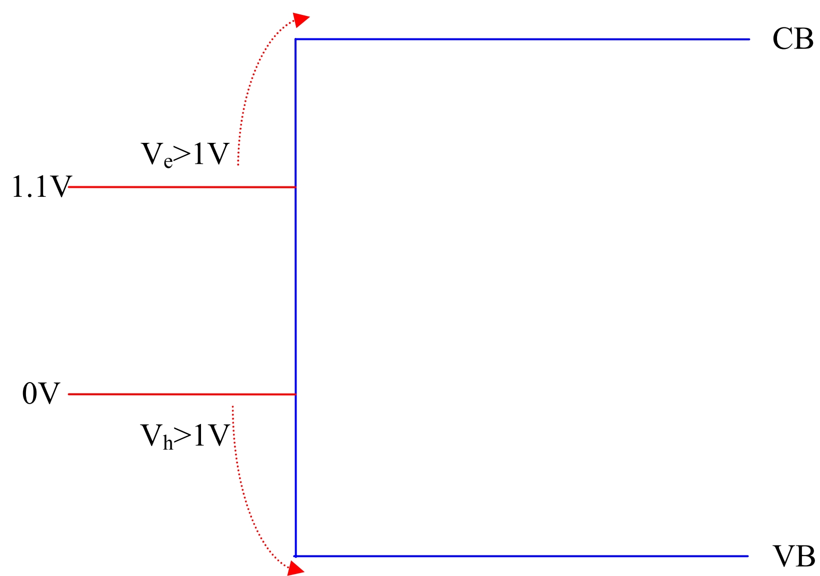


Fig. 1-3 Schematic of band offsets determining carrier injection in oxide band states.

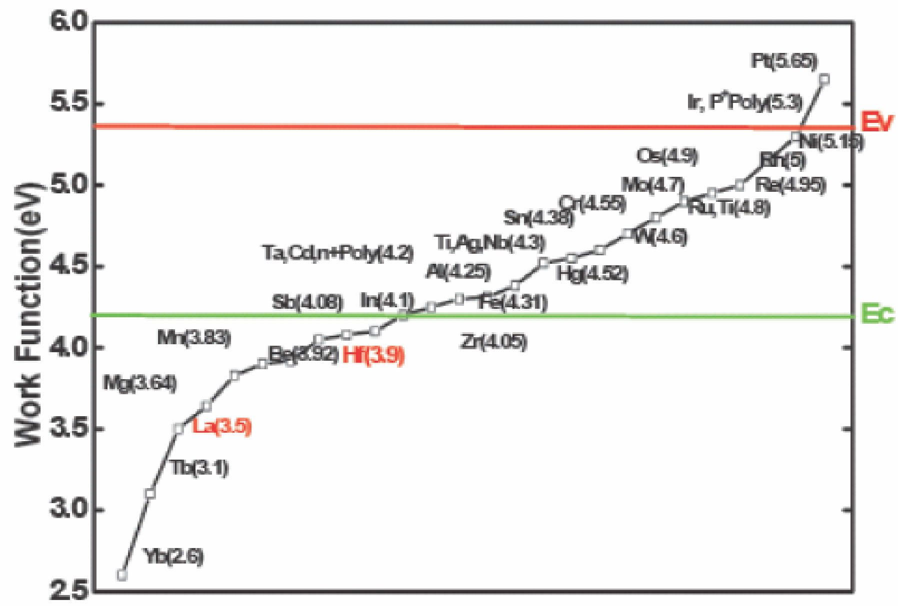


Fig. 1-4 The values of work function for different metal materials.

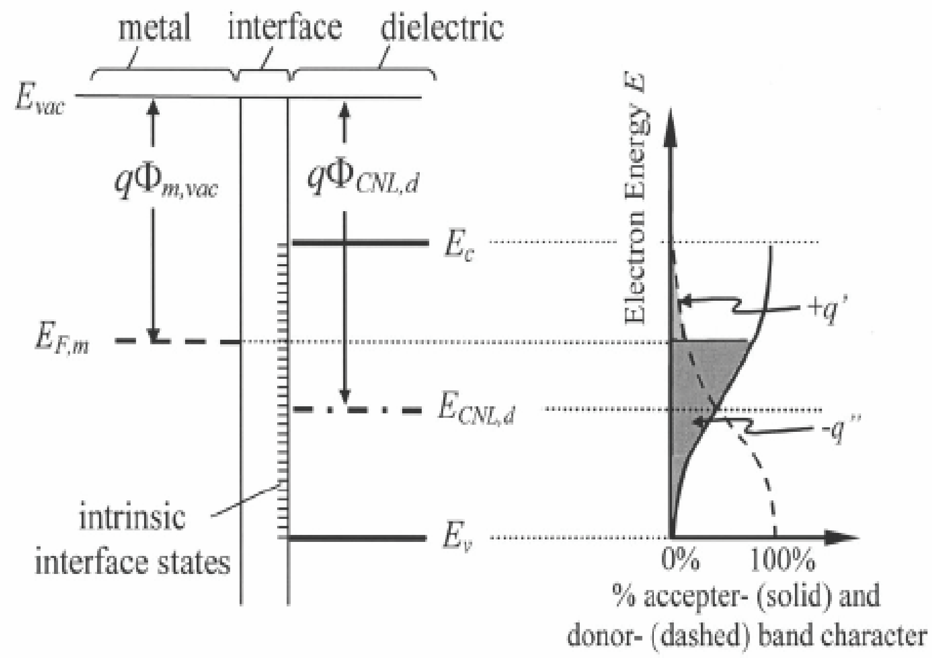
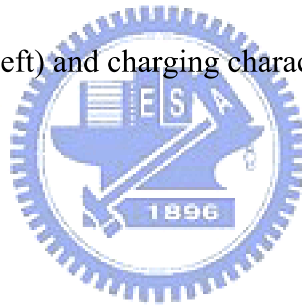
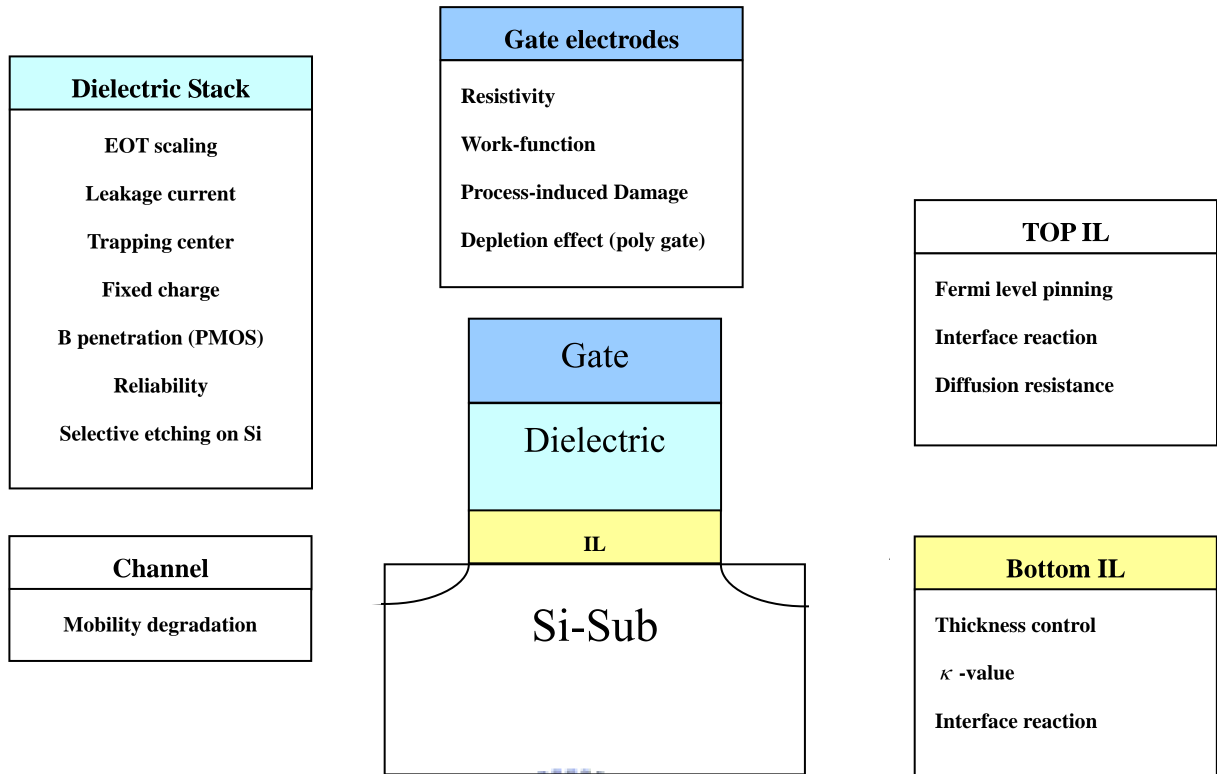


Fig.1-5 Energy band diagram (left) and charging character of interface states (right) for the metal-dielectric interface.





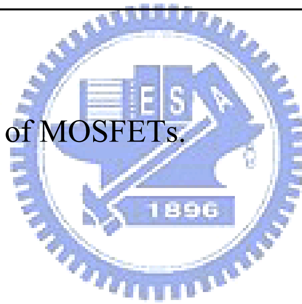


Fig. 1-6 The challenging issues of MOSFETs.

Chapter 2

The Experimental Procedure

2.1 The Fabrication Step

The gate-first Ir_xSi/HfSiON p-MOSFETs and Hf_xSi/HfSiON n-MOSFETS were fabricated on 12-in N-type Si wafers with resistivity of 1–10 Ω・ cm. After RCA cleaning, 4-nm HfSiO dielectric (Hf / (Hf + Si) =50%) was deposited by atomic-layer deposition (ALD). HfSiON gate dielectric was formed by applying NH_3 plasma surface nitridation on HfSiO. For p-MOSFETs, 5–30-nm amorphous Si and 20–30-nm Ir were deposited by physical vapor deposition (PVD) [2-1]. For Ir/Si/HfSiON capacitors, a 1000℃ RTA was applied for 10 s to form Ir_xSi gates. For MOSFETs, additional 400-nm Si was deposited on top of Ir/Si to avoid ion implantation penetrating through the thin Ir/Si. After gate definition, Boron was implanted at 25-KeV energy and 5×10^{15} cm^{-2} dose, and activated at 1000℃ RTA for10 s. Meanwhile, Ir_xSi was also formed during RTA, where the $x = 3$ was determined by X-ray diffraction measurements.

For n-MOSFETS, amorphous Si with various thickness of 50 to 5 nm was deposited on HfSiON as a silicide layer and metal barrier for subsequently deposited 20-nm-thick Hf by physical vapor deposition (PVD) [2-3]. The MOS capacitor was formed by patterning and RTA at 1000℃ for 10 s. Then, additional 150-nm-thick amorphous Si was deposited on Hf/Si/HfSiON to prevent ion implantation penetration, where the n^+ source–drain regions are

formed by using a phosphorus ion implantation at 35 KeV. Then, the 1000 ◦C RTA was applied to activate the implanted dopant and the n-MOSFET was fabricated by this self-aligned gate first process. At such high 1000℃ RTA temperature, the fast silicidation reaching to the Si/HfSiON interface may also reduce the reaction of thin amorphous Si (5 nm) with high- κ dielectric to cause Fermi-level pinning.

2.2 The Measurement of MOSFET

We used ion-mass spectroscopy (SIMS) to measure the Ir distribution profile. The fabricated p-MOSFETs and n-MOSFETs were further characterized by capacitance–voltage (*C*–*V*) and current–voltage (*I*–*V*) measurements. The capacitance–voltage (*C*–*V*) and current–voltage (*I*–*V*) measurements are measured by HP 4156C semiconductor parameter analyzer and HP 4284A precision LCR meter. In order to compare, Al, Ir-gated, and Hf-gated MOS capacitors on HfSiON were also fabricated. To prevent the different oxide charge from causing error in $\Phi_{m,eff}$ extraction, HfSiON was subjected to the same thermal cycle (1000 ℃ RTA for 10 s) before Al gate deposition.

2.3 Process Flow

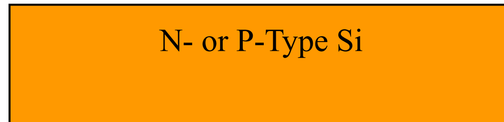


Fig. 2-1 N- or P-Type Si Substrate

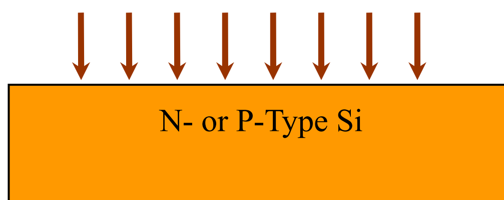


Fig. 2-2 RCA clean

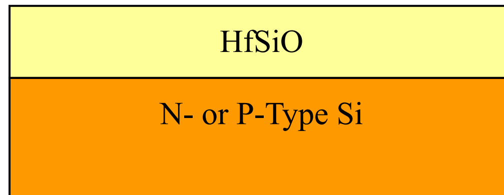


Fig. 2-3 Deposit HfSiO

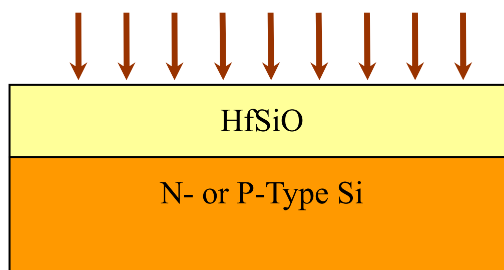


Fig. 2-4 NH_3 Plasma Nitridation

Amorphous-Si

HfSiON

N- or P-Type Si

Fig. 2-5 Deposit Amorphous-Si

Ir or Hf

Amorphous-Si

HfSiON

N- or P-Type Si

Fig. 2-6 Deposit Ir or Hf

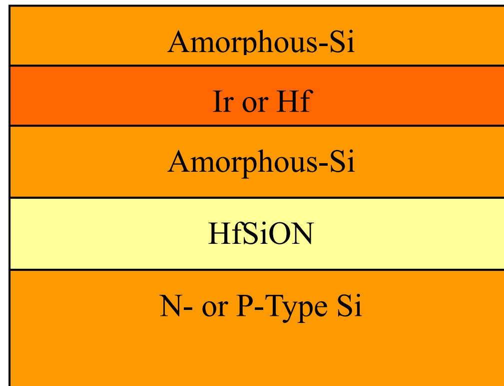


Fig. 2-7 Deposit Si

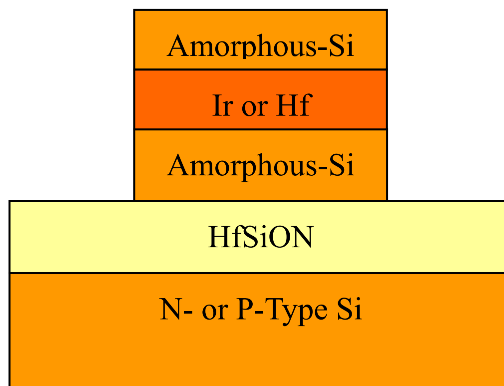


Fig. 2-8 Gate Definition

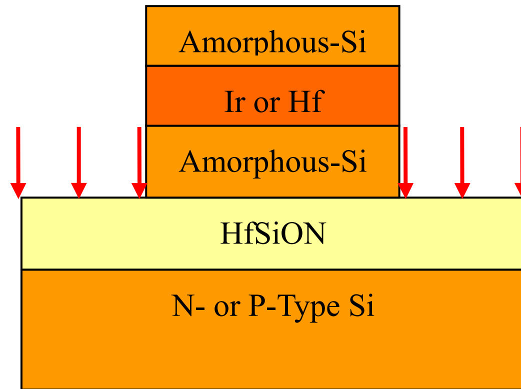


Fig. 2-9 Ion implantation

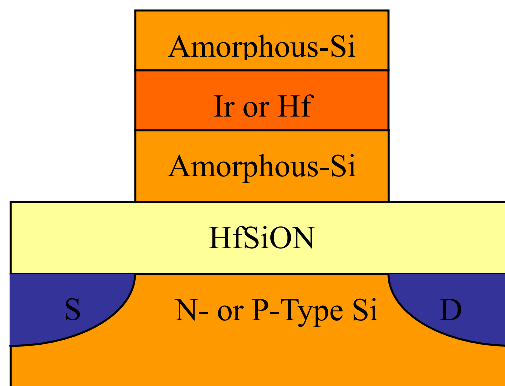


Fig. 2-10 1000℃, 10s, RTA

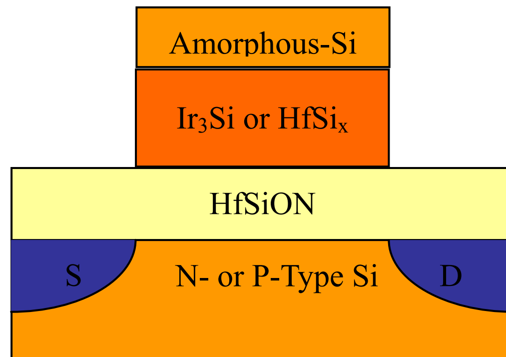


Fig. 2-11 Fabricated MOSEFET

Chapter 3

The Characteristics and Analysis of P-MOSFET

3.1 Introduction

We have measured the *J-V* characteristics of the MOSFETs, and then want to find out their threshold voltages by using equation (10).

$$I_D = \frac{W}{L} C_{OX} \mu_{eff} (V_G - V_{TH} - \frac{V_D}{2}) V_D \quad (10)$$

$$g_D = \frac{\partial I_D}{\partial V_D} \Big|_{V_G = CONSTANT} \quad (11)$$

Measure I_{DS} versus V_{GS} curve with V_S=0V, V_B=0V, and $V_D << V_G - V_{TH}$. Make the I_{DS} versus V_{GS} plot. There is a linear region at the neighboring of V_{GS} at which $g_{m,lin}$ is maximum. The X-axis intersection is $V_{GS} = V_{TH} + \frac{V_D}{2}$, and we could find out the V_{TH}. Then we want to find out the mobility. Using equation (11) finds out the g_d in linear region, and equation (12) gives the effective mobility.

$$\mu_{eff} = \frac{g_d L}{W Q_n} \quad (12)$$

$$Q_n = \int_{-\infty}^{V_G} C_{GC} dV_G \quad (13)$$

3.2 The Effective Metal Workfunctions

We have fabricated three different kinds of gate electrodes. The three materials are Ir_xSi, Ir, and Al. Fig 3.1 and Fig. 3.2 shows the measured *C–V* characteristics of Ir_xSi, Ir, and Al gates on HfSiON MOS devices. We use low-temperature Al-gated HfSiON capacitors as a

reference because pure metal deposited at low temperature has less interface reaction with high-κ dielectrics than high-temperature process. The Al-gated HfSiON capacitors have fewer extrinsic states, and thus the Fermi-level pinning effect is not obvious in the structure [3-4].

$$
\begin{aligned}
V_{fb} &= \Phi_{ms} - Q_f / C_{ox} \\
&= (\Phi_m - \Phi_s) - (Q_f / \varepsilon_0 \kappa_{ox}) t_{ox} \\
&= (\Phi_m - \Phi_s) - (Q_f / \varepsilon_0 \kappa_{sio2}) EOT \qquad (14)
\end{aligned}
$$

The flat band voltage (V_{fb}) is expressed as equation (14), where Φ_m and Φ_s are the work functions for metal gates and Si, respectively. Q_f , C_{ox}, t_{ox}, and equivalent-oxide thickness (*EOT*) are the oxide charge, capacitance, physical thickness, and *EOT* for high-κ dielectrics. Since the three kinds of MOS devices have the same thermal cycle (1000 ◦C RTA for 10 s) before gates formation, we could assume that the fixed charge (Q_f) amount should be the same. In Fig 3-2, the various flat band voltages (V_{fb}) may be due to the different metal work functions. Therefore, the principal effect of V_{fb} shift might be due to the difference of effective workfunction ($\Phi_{m,eff}$). The processes before gate definition are the same, and the MOS capacitors all have EOT values of 1.6nm. The shifts of *C–V* curves with different gate electrodes are attributed to the different work functions ($\Phi_{m,eff}$). Ir/HfSiON after 900 ◦C RTA has a large V_{fb} shift of 1.15 V to Al gates ($\Phi_{m,eff}$ = 4.1eV). It results in the required high $\Phi_{m,eff}$ of 5.25 eV. This work-function value is also close to 5.27 eV for Ir. The pure metal Ir gates showed no obvious pinning effect. This is due to weak bonding strengths of Ir–O or Ir–N that reduce the Fermi-level-pinning-related interface reaction. However, we observed

that Ir/HfSiON capacitors failed after 1000℃ RTA.

3.3 Thermal Stability

In Fig 3-3, Ir/HfSiON is failed after 1000℃ RTA. In order to activate the impurities, the gates have 1000℃ RTA after S/D implantation. To improve thermal stability, additional amorphous Si of 5–30 nm was inserted between Ir and HfSiON and also serve as a metal diffusion blocking layer. After 1000℃ RTA, Ir_xSi gate is formed. Good *C–V* characteristics were measured for Ir_xSi/HfSiON devices after the required 1000℃ RTA, although thermal stability was traded off at the Fermi-level pinning. In Fig 3-1, we obtained a high $\Phi_{m,eff}$ of 4.95 eV for Ir_xSi/HfSiON devices with the inserted 5-nm amorphous Si. Slow depletion for Ir_xSi /HfSiON devices with 30-nm amorphous Si may be due to nonuniform silicidation as examined by TEM, where locally unreacted Si was found to cause voltage drop in gate electrodes. The formation of FUSI gates is evident from the same inversion and accumulation capacitances measured in MOSFETs.

3.4 The J-V Characteristics

After 1000 ◦C RTA, Ir/HfSiON devices had high leakage currents and failed thus as shown in Fig. 3.3. On the other hand, Ir_xSi gates on HfSiON successfully improved thermal stability to 1000 ℃ RTA with low leakage current comparable with p^+ poly-Si gates. 1000 ◦C RTA is required for dopant activation after ion implantation of source and drain. The measured large V_{fb} shift of Ir_xSi is supported by SIMS profile, as shown in Fig. 3-5. Here, Ir

segregation toward amorphous Si formed Ir_xSi on HfSiON surface. Therefore, good thermal stability of 1000 ◦C RTA, a reasonable high $\Phi_{m,eff}$ of 4.95 eV, and a low gate dielectric leakage current can be achieved in Ir_xSi /HfSiON MOS capacitors at the same time. These are the few methods to achieve a high $\Phi_{m,eff}$ in Hf-based oxide p-MOS devices. There is a widely studied tuning method by impurity segregation in FUSI/SiON. However this method can not be applied to high-κ metal oxide due to the stronger interface reaction. In the following, we will study Ir_xSi /HfSiON devices with the thinnest 5-nm amorphous Si which has the best performance in the experiments. Compared with p-MOSFET, the V_{fb} of thicker Si layer is too low. Fig. 3-6 shows the transistor $I_d - V_d$ characteristics as a function of V_g - V_t for 1000 ◦C RTA Ir_xSi/HfSiON p-MOSFETs. The splendid results of $I_d - V_d$ curves of Ir_xSi/HfSiON transistors in Fig. 3-4 show little device performance degradation. Fig. 3-7 shows the $I_d - V_g$ characteristics of Ir_xSi-gated p-MOSFETs with HfSiON as the gate dielectric. In this work, we obtained the low V_t of −0.15 V from the linear $I_d - V_g$ plot, which is consistent with the large $\Phi_{m,eff}$ of 4.95 eV from C–V curves and the Ir accumulation on HfSiON from SIMS. Fig. 3-8 shows the extracted hole mobilities versus gate electric fields from the measured $I_d - V_g$ data of Ir_xSi/HfSiON p-MOSFETs. High hole mobilities of 84 and 53 $cm^2/V \cdot s$ are obtained at peak value and 1 MV/cm effective field for Ir_xSi/HfSiON p-MOSFETs, respectively, which is compatible with the published data in the literature [3-4]. Good hole mobility also indicates low Ir diffusion through HfSiON to

inversion channel, even though excess Ir is necessary to prevent unreacted amorphous Si from causing gate depletion or increased Fermi-level pinning. Therefore, a high $\Phi_{m,eff}$, a small V_t, and good hole mobility are achieved in Ir_xSi/HfSiON p-MOSFETs.

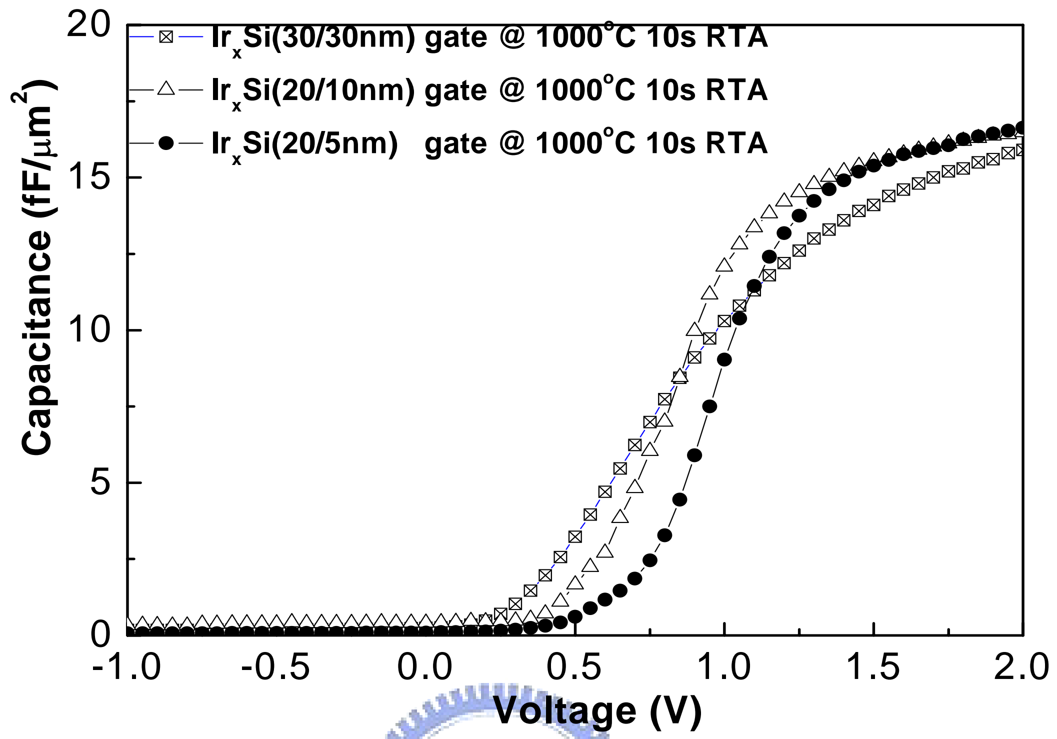


Fig. 3-1 *C–V* curves of HfSiON/n-Si with various Si thickness. The device areas are 100 × 100 μm.

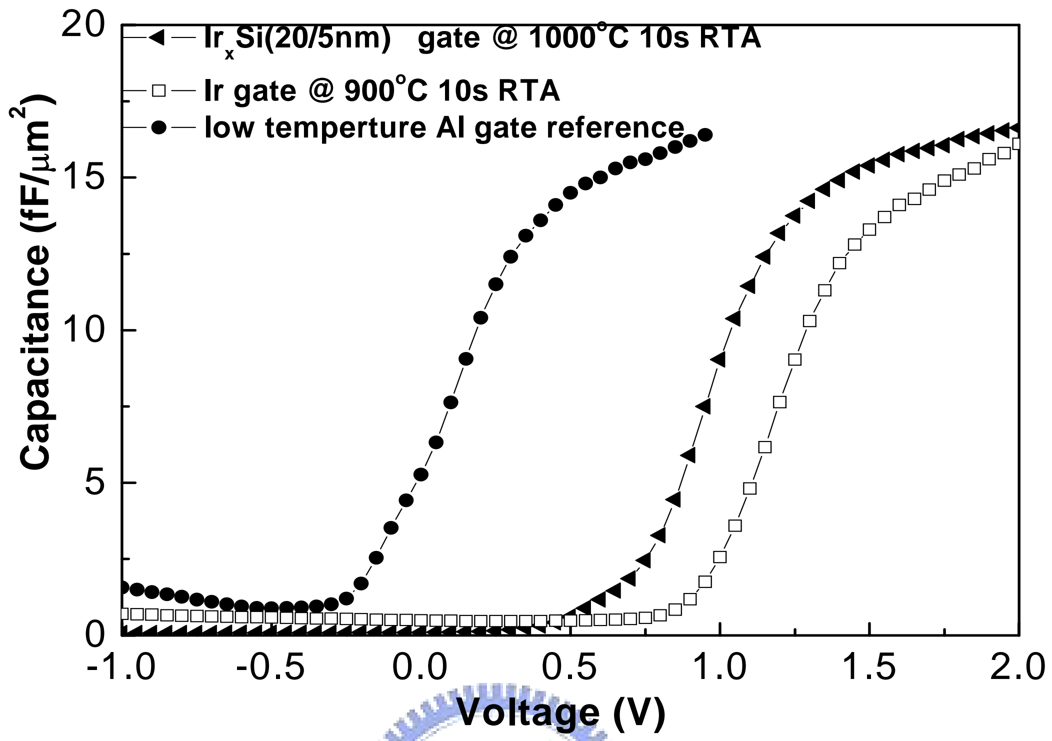
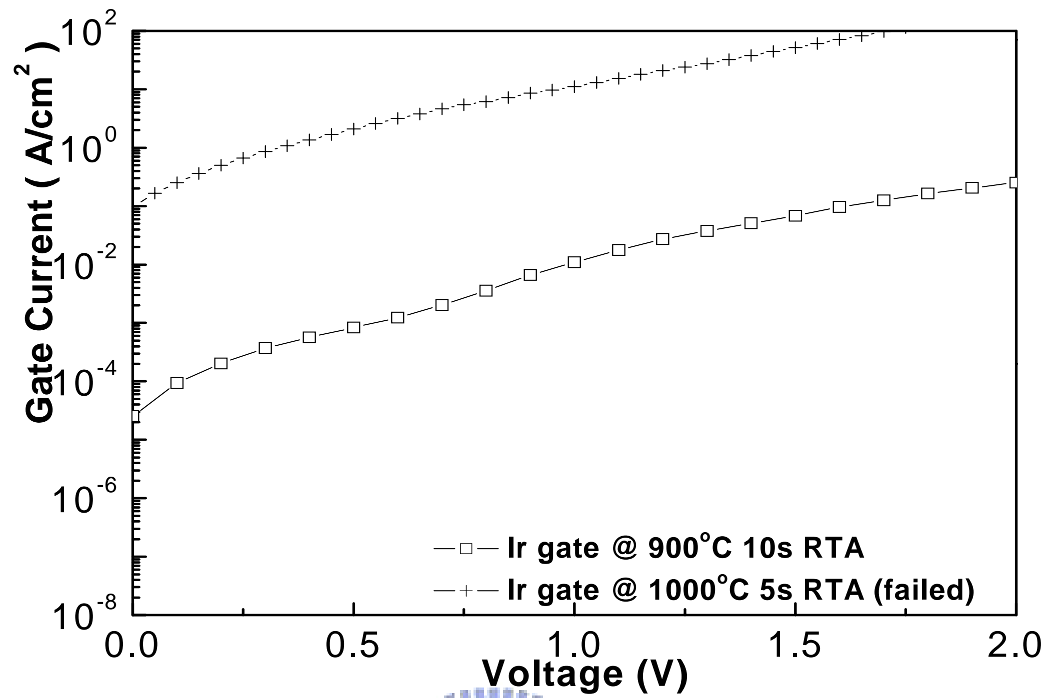


Fig. 3-2 *C–V* curves of Ir/HfSiON, Ir_xSi/HfSiON (20/5nm), and Al/HfSiON.



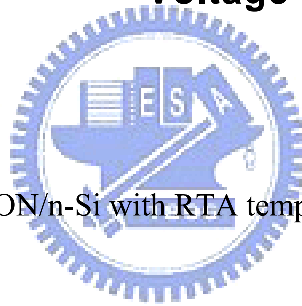


Fig. 3-3 I_g-V_g curves of Ir/HfSiON/n-Si with RTA temperatures of 1000℃ and 900℃.

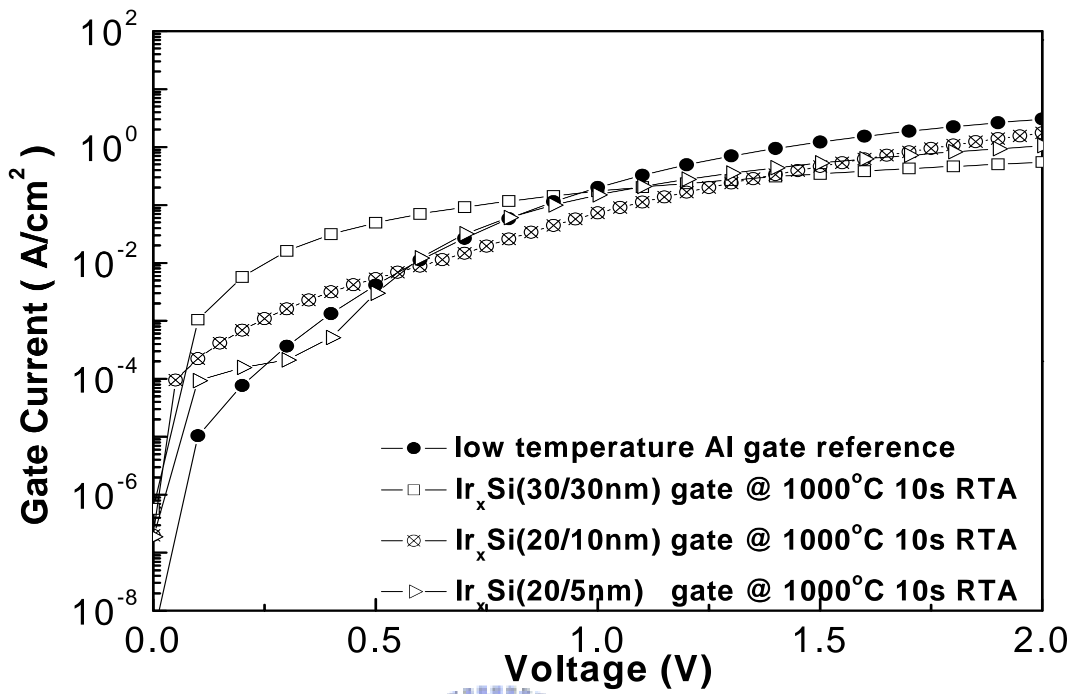


Fig. 3.4 I_g-V_g curves of Ir_xSi/HfSiON and Al/HfSiON.

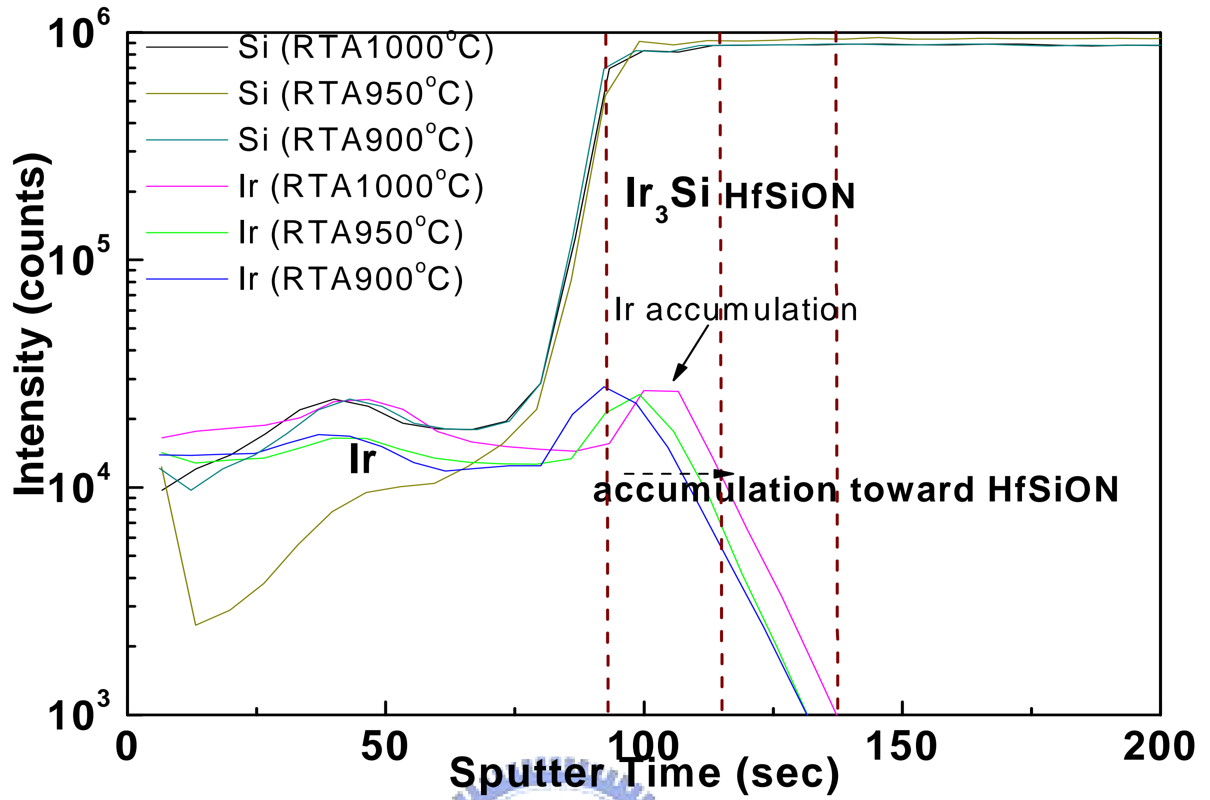


Fig. 3-5 SIMS profile of Ir_3Si gates on HfSiON at different RTA temperatures. The Ir_3Si that accumulated toward HfSiON interface is found to unpin the Fermi level.

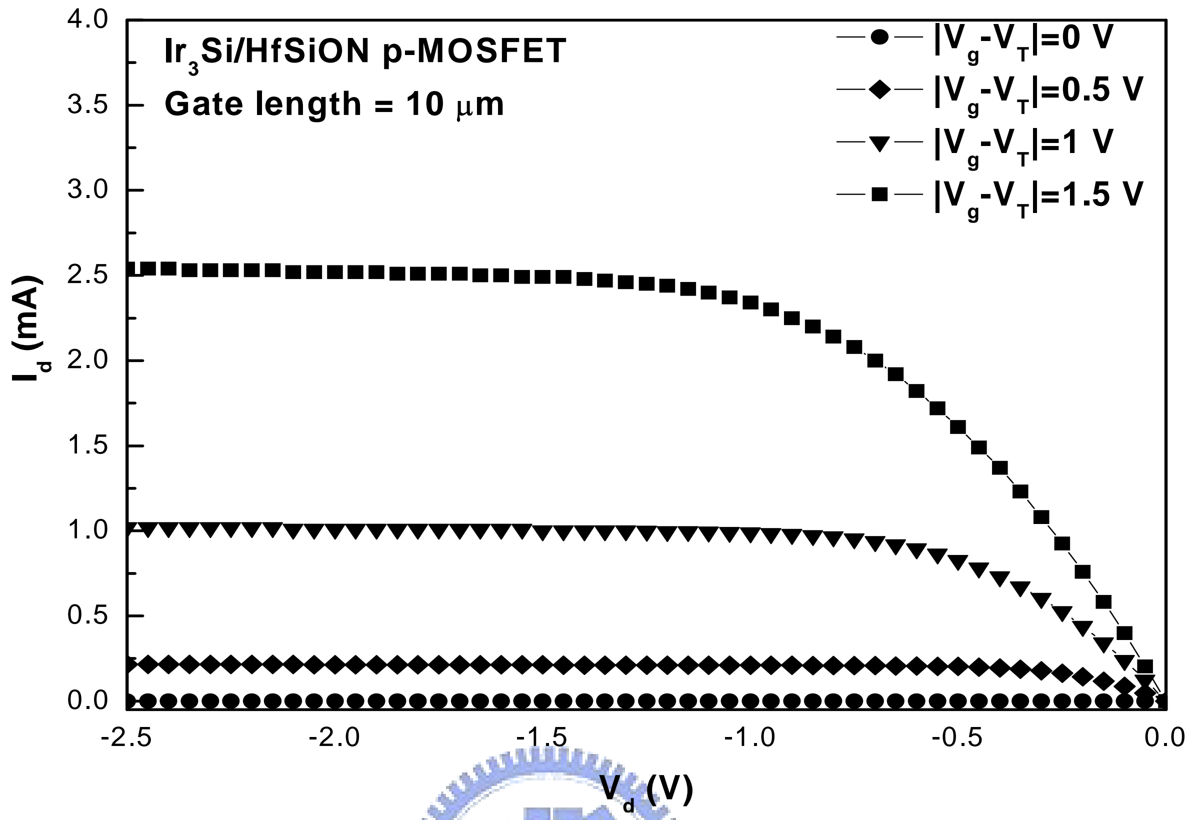


Fig. 3-6 I_d–V_d curves of Ir_3Si/HfSiON p-MOSFETs.

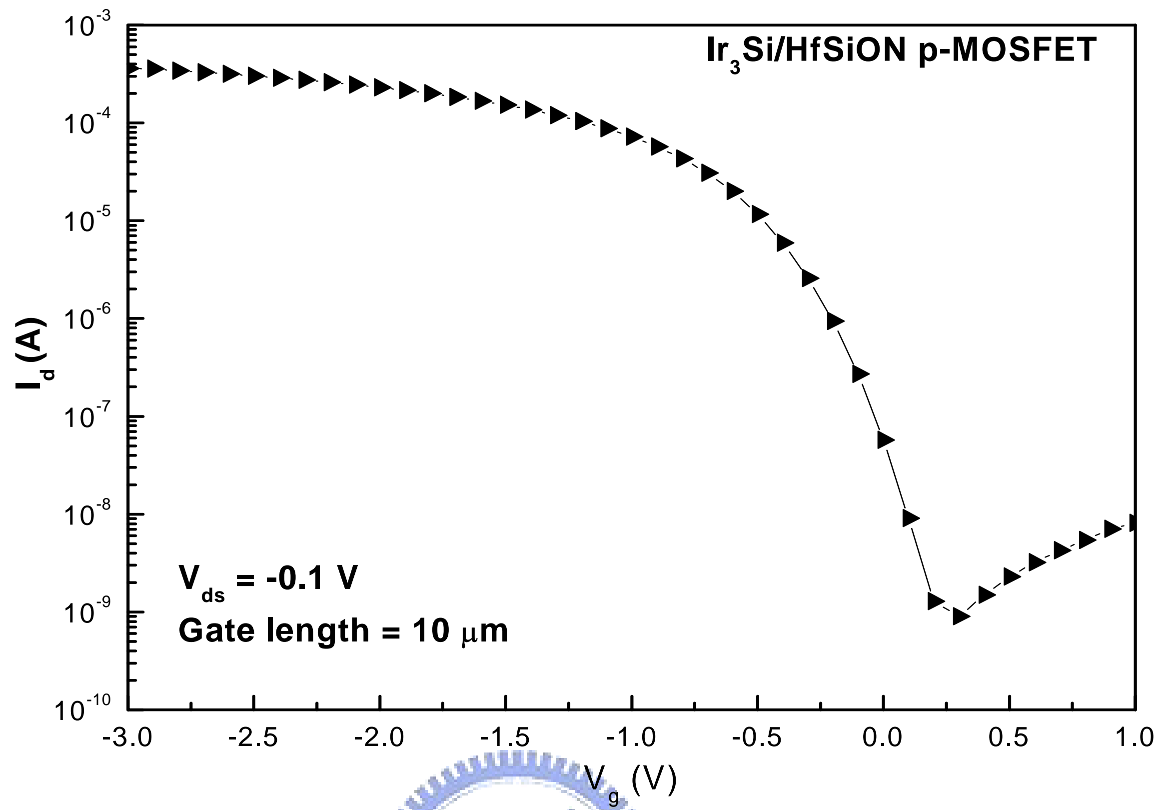


Fig. 3-7 I_d–V_g curves of Ir_3Si/HfSiON p-MOSFETs.

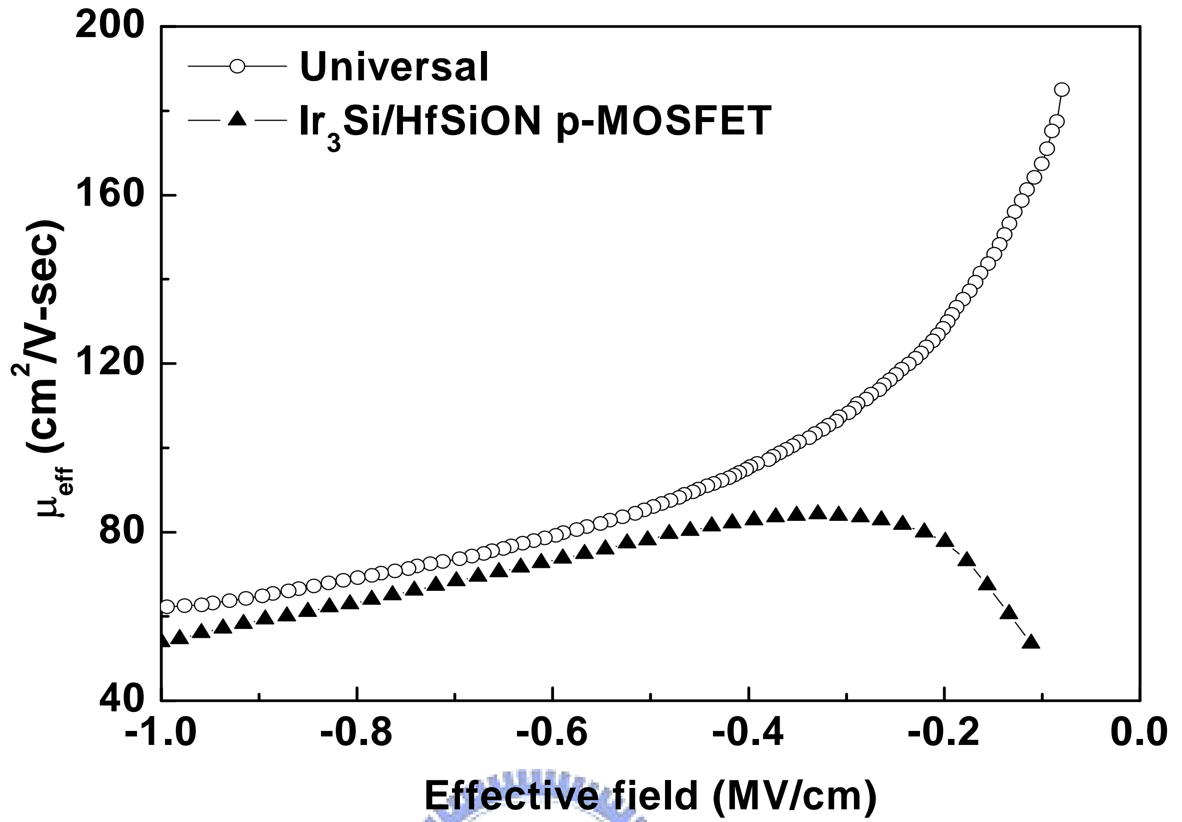


Fig. 3-8 Extracted hole mobilities from I_d–V_g characteristics of Ir_3Si/HfSiON p-MOSFETs.

Chapter4

The Characteristics and Analysis of N-MOSFET

4.1 The Effective Metal Workfunctions and Thermal Stability

Fig. 4-1, Fig. 4-2, and Fig. 4-3 shows the *C–V* and *J–V* characteristics for $HfSi_x$/HfSiON and Al/HfSiON capacitors, where the $HfSi_x$ gate was formed at 1000℃ RTA. The Al-gated capacitor has work-function of 4.1 eV. For various amorphous Si of 50 and 10 nm on HfSiON, the capacitance density decreases as the thickness of amorphous Si increases. This implies that not all amorphous silicon is silicided in $HfSi_x$ gate on HfSiON. Thus a higher flat-band voltage (V_{FB}) due to the Fermi-level pinning on high-κ dielectric occurs. In contrast, the $HfSi_x$ with thin 5-nm amorphous Si has the same capacitance density with Al gate. It is indicated that all amorphous Si is silicided. From the *C–V* shift referenced to the control Al gate, an extracted $\Phi_{m,eff}$ of 4.27 eV is obtained for $HfSi_x$/HfSiON. This result approaches the desired workfunction (Φ_m) of NMOSFET. The low V_{FB} and $\Phi_{m,eff}$ for $HfSi_x$ gate capacitors with 5-nm amorphous Si may be due to the Hf diffusion toward the HfSiON surface through thin amorphous Si that decreases the work function. In addition, low leakage current of 1.9×10^{-5}A/cm at −1 V is measured at an equivalent oxide thickness (EOT) of 1.6 nm. This result shows the good thermal stability of $HfSi_x$ gate on HfSiON dielectric after 1000℃ RTA. Therefore, the experiment obtained reasonable low $\Phi_{m,eff}$ of 4.27 eV and a low gate leakage

current in $HfSi_x$/HfSiON MOS capacitors at the same time [4-5].

4.2 *J-V* Characteristics

Fig. 4-4 shows the transistor $I_D - V_D$ characteristics as a function of $V_g - V_t$ for the 1000 ℃ RTA-annealed $HfSi_x$/HfSiON n-MOSFETs. Fig. 4-5 displays $I_D - V_g$ characteristics of the $HfSi_x$/HfSiON n-MOSFETs. A low V_t of only 0.14 V was measured from the linear $I_D - V_g$ plot, which agrees with the low $\Phi_{m,eff}$ of 4.27 eV from the *C–V* measurements. Fig. 4-6 shows the electron mobility extracted from the measured $I_D - V_g$ curves of the n-MOSFETs. A peak electron mobility of 216cm^2/V·s was obtained for the $HfSi_x$/HfSiON n-MOSFETs.

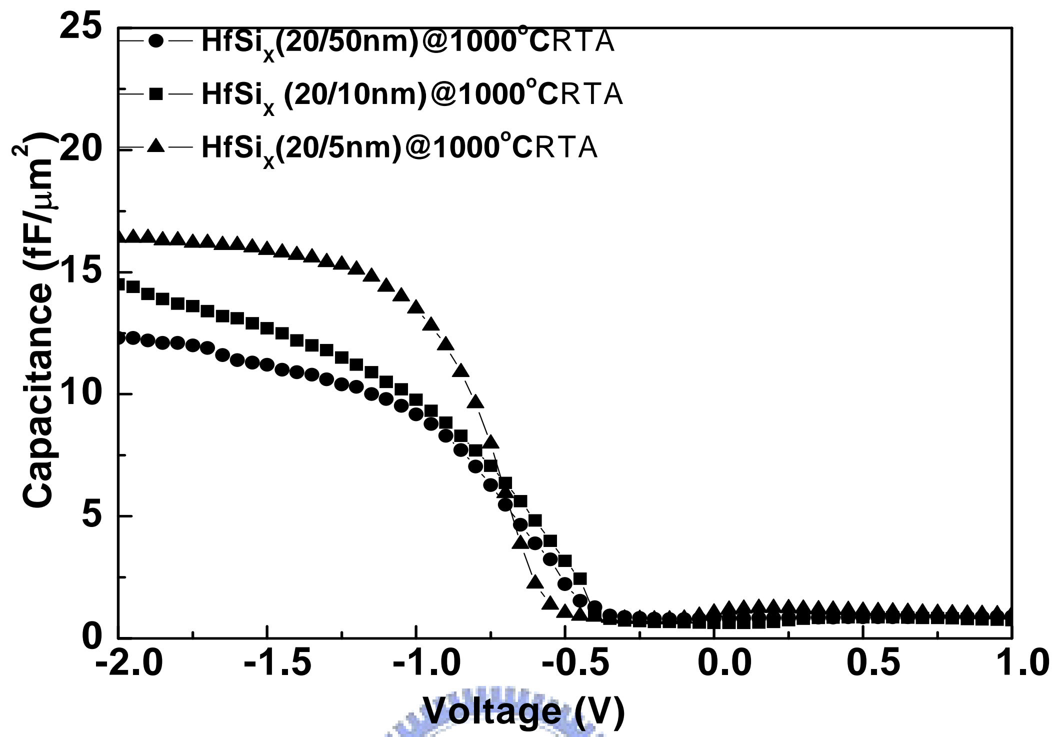


Fig. 4-1 *C–V* characteristics for high-temperature RTA formed $HfSi_x$/HfSiON with various amorphous Si.

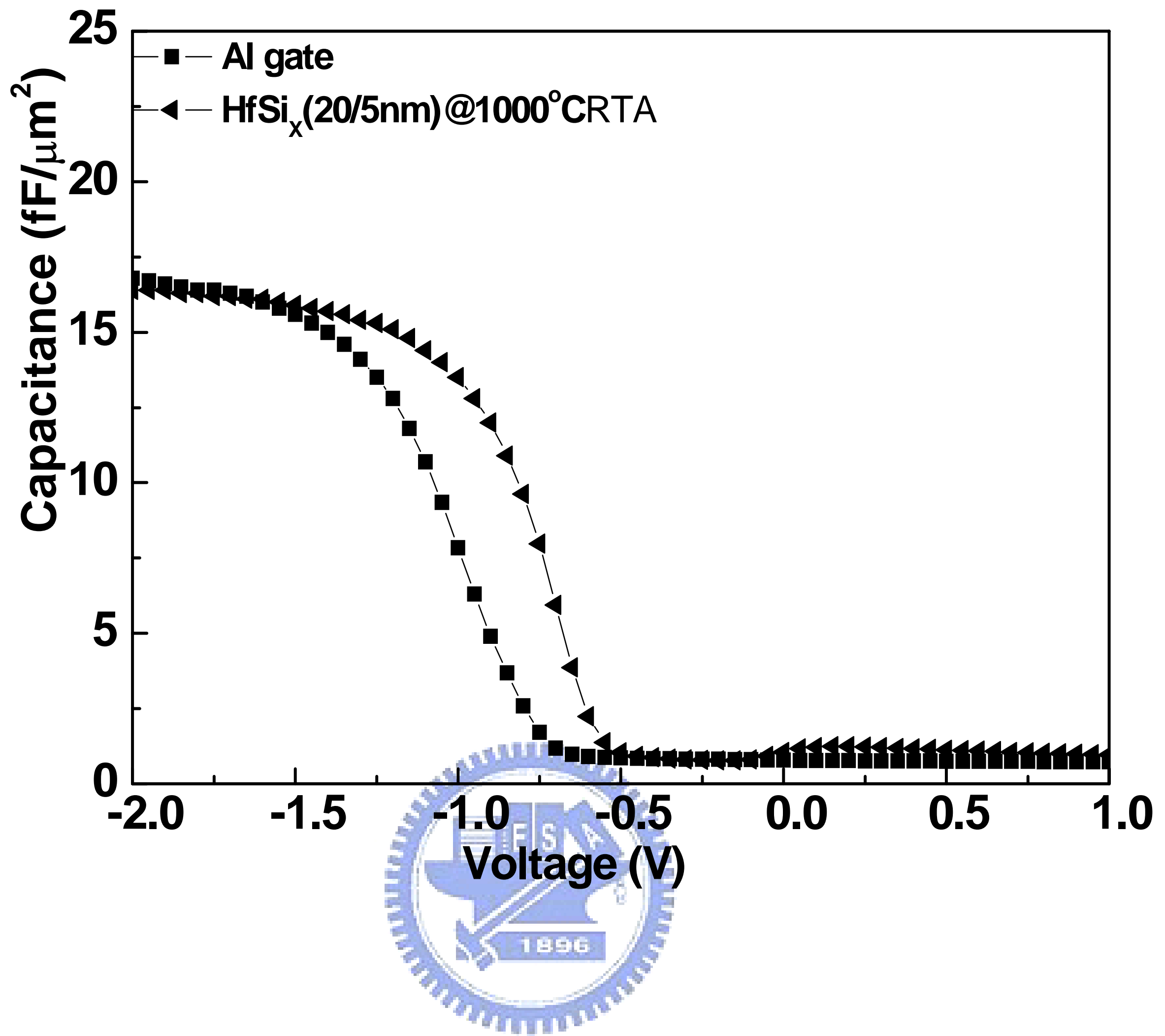


Fig. 4-2 *C–V* characteristics for high-temperature RTA formed $HfSi_x$ (20/5nm) HfSiON and low-temperature Al/HfSiON capacitors.

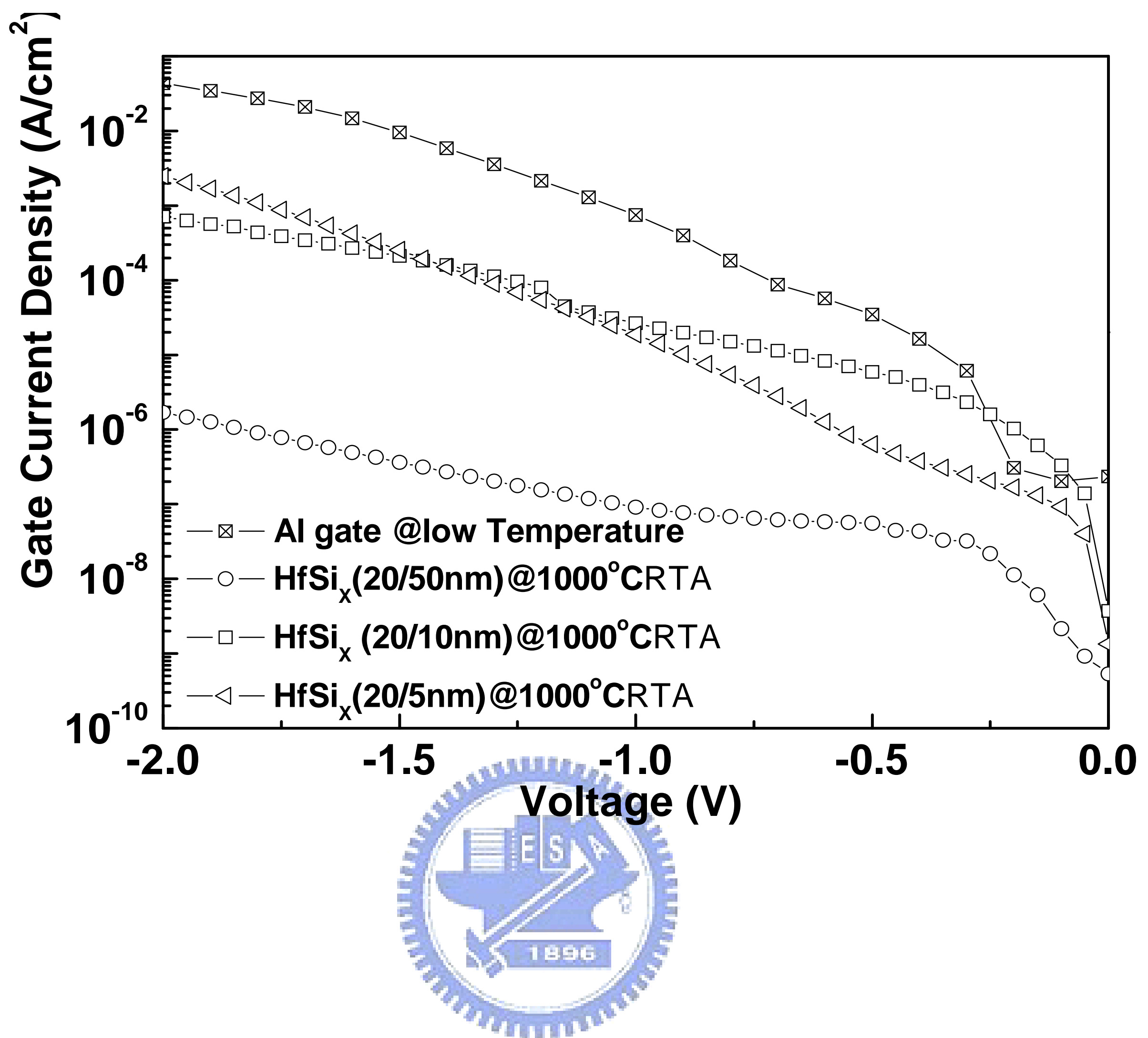


Fig. 4-3 J_g - V_g characteristics for high-temperature RTA formed HfSi$_x$/HfSiON and low-temperature Al/HfSiON capacitors.

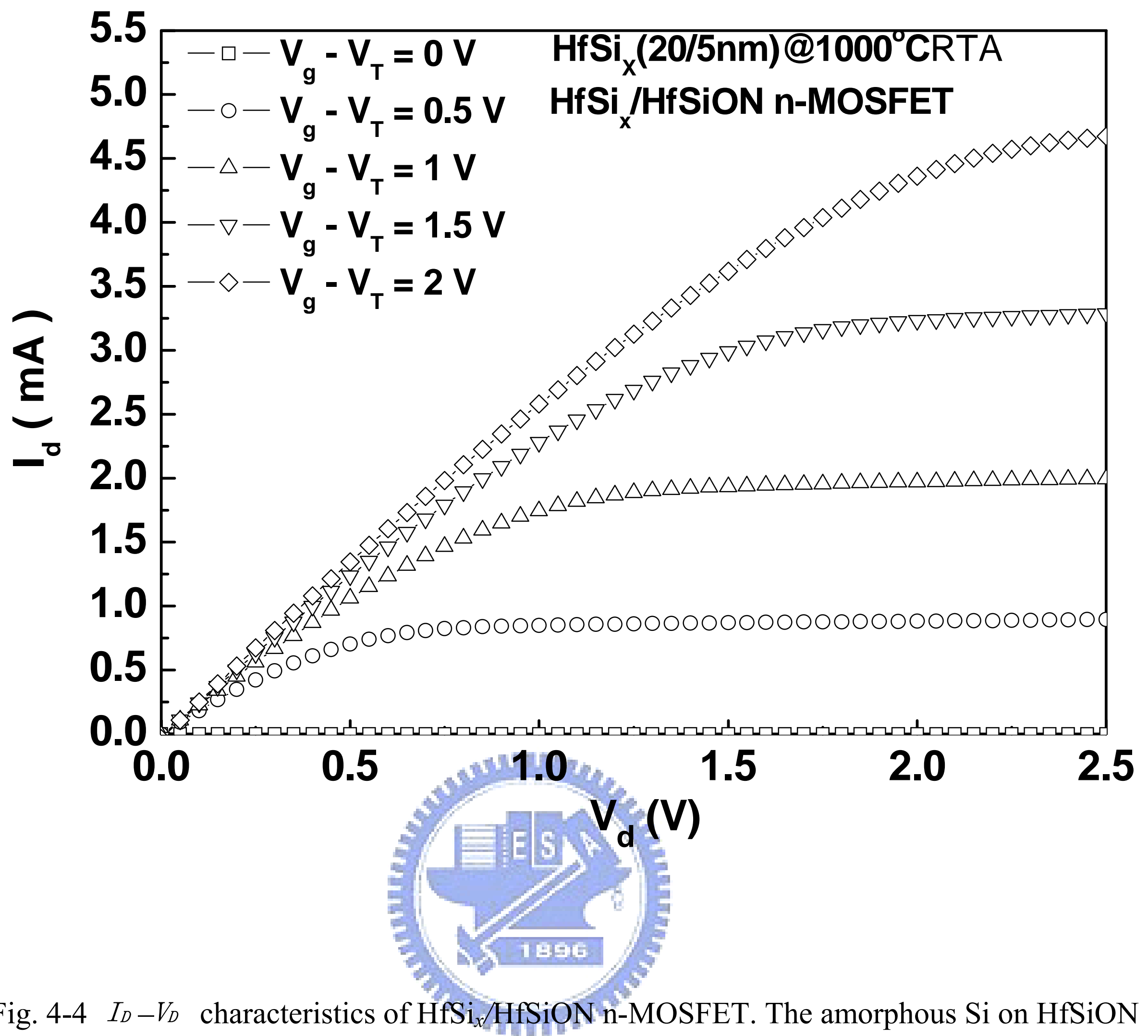


Fig. 4-4 $I_D - V_D$ characteristics of $HfSi_x$/HfSiON n-MOSFET. The amorphous Si on HfSiON was 5 nm and gate length was 10 μm.

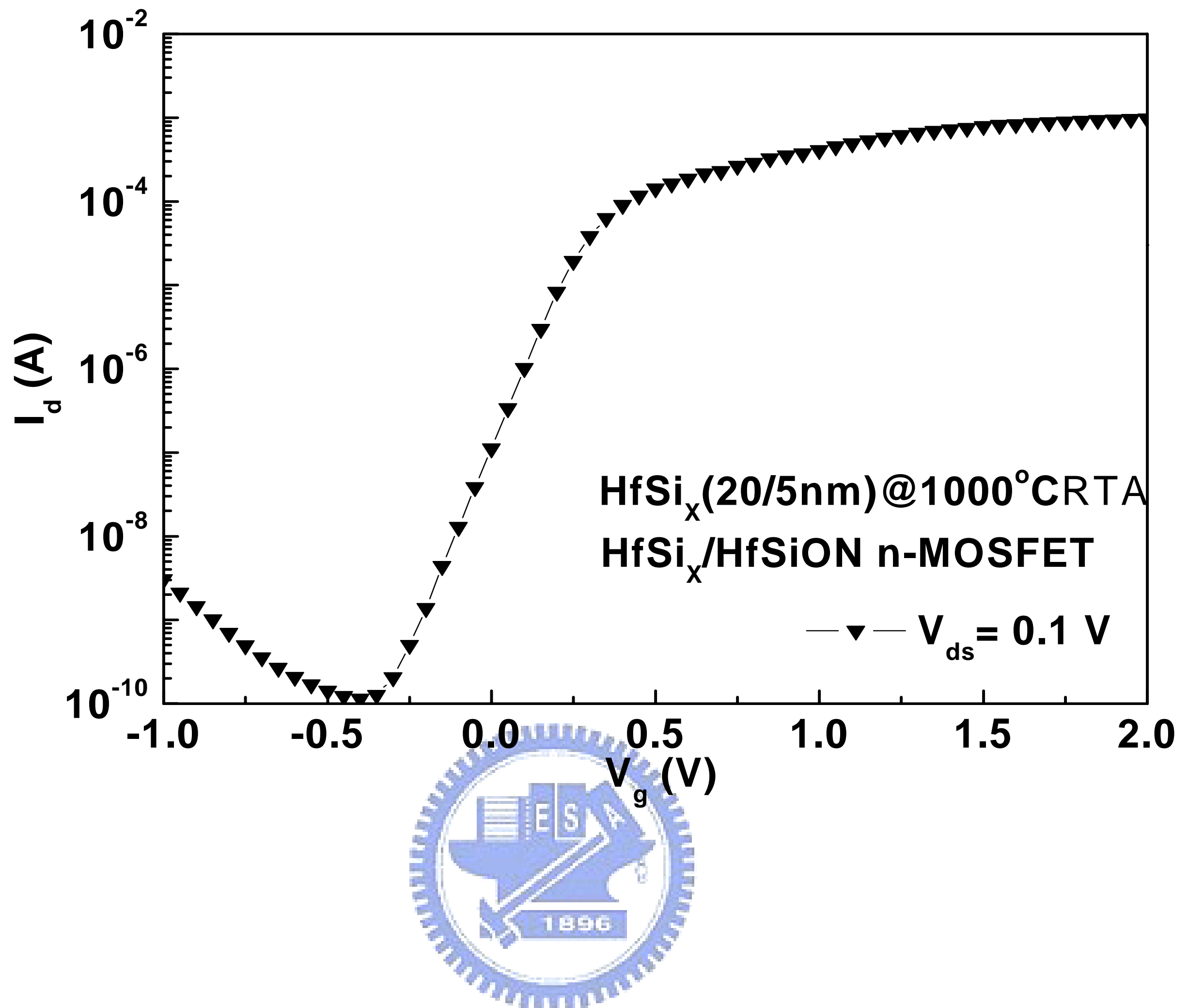


Fig. 4-5 I_d–V_g characteristics of HfSi$_x$/HfSiON n-MOSFET.

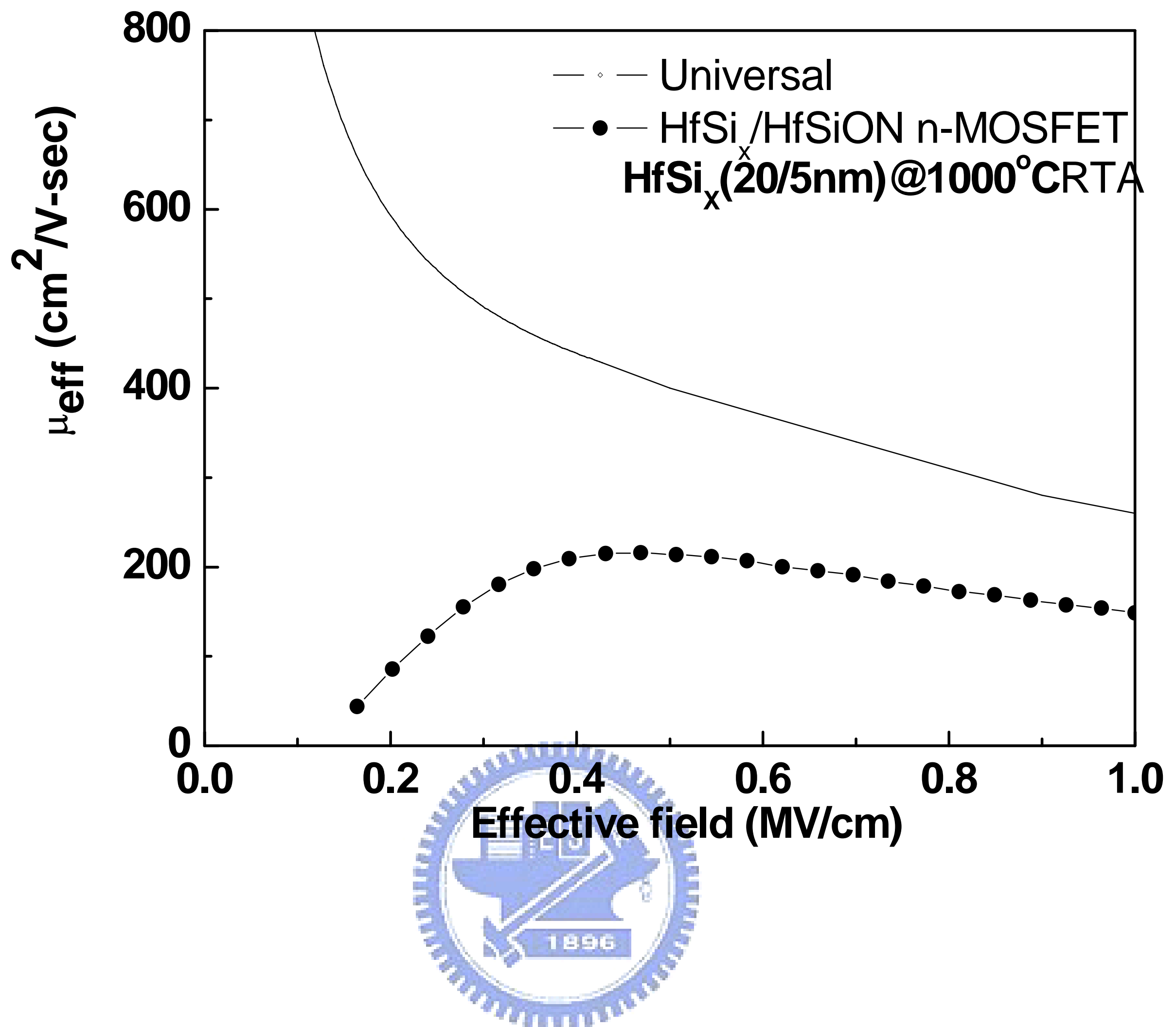


Fig. 4-6 Electron mobility of $HfSi_x$/HfSiON n-MOSFETs.

Chapter5

Conclusion

In the experiment, we have obtained good device performance of $Ir_xSi/HfSiON$ p-MOSFETs with a high $\Phi_{m,eff}$ of 4.95 eV, a small V_t of −0.15 V, a peak hole mobility of 84 $cm^2/V \cdot s$, and 1000℃RTA thermal stability. For NMOSFET, a low $\Phi_{m,eff}$ of 4.27eV, threshold voltage of 0.14V, and a mobility of 216$cm^2/V \cdot s$ are obtained. They are obviously that the processes can be integrated in current technology. On the other hand, we will study hard to decrease EOT by replacing the high-κ gate dielectrics in the future. The threshold voltages of devices are needed to decrease, and avoid the Fermi-Level pinning. However, the research of this work proved that this is an effective way to meet the ITRS roadmap after 2008.

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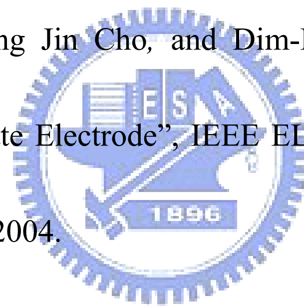

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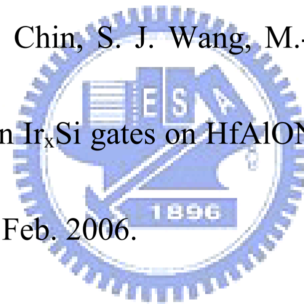

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論文題目：

金屬矽化物-高介電常數介電質-半導體場效應電晶體之電性研究

The Research of Electrical Characteristics of FUSI Gate-High-κ Dielectric-Semiconductor Field-Effect Transistor.