

不同前處理對二氧化鉛閘極介電層在電 特性上的影響

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在先進的互補金氧半技術中，元件尺寸快速的微縮使二氧化矽厚度縮小到不到 1.2 奈米。而極薄的二氧化矽介電層將伴隨著極大的直接穿遂漏電流，而這個直接穿遂漏電流將對元件的功率消耗有嚴重的影響。在閘極二氧化矽介電層薄到 10 奈米以下的情況之下，為了解決這嚴重的直接穿遂漏電流現象，我們將利用高介電係數材料來替換傳統的二氧化矽。我們利用高介電係數材料在相同的等效二氧化矽厚度之下，能擁有較大的實際物理厚度以抵擋直接穿遂漏電流，而且可以維持元件操作所須的閘極電容。

在眾多高介電係數材料之中，二氧化鉛是一種非常有潛力的高介

電係數材料。它有較高的介電係數，足夠高的載子能障（對電子而言為 1.6 電子伏特，對電洞而言為 3.2 電子伏特），以及在製程上能有著良好的穩定性。但是由於材料本身的基本特性，二氧化鈣中的氧氣容易和底下的矽基板產生介電常數較低的物質，這會對元件的微縮能力及電性有很大的影響。所以在實驗中，我們利用氮氣及一氧化二氮高密度電漿來處理矽基板表面，形成一層薄的含氮的界面層，來探討和一般傳統的二氧化矽界面層或直接沉積在矽基板上在電性以及可靠度上的差異，結果顯示經過高密度電漿處理過後的矽表面會有損害，漏電流仍然未達到我們要的標準，必須經由傳統的快速升溫回火來修補這些損害，所以修補後呈現出較小的漏電流、較小的磁滯、較小的頻率分散、較大的崩潰電壓。而就可靠度來說，經由氮氣電漿處理的試片會有最大的崩潰時間、崩潰電荷，以及最好的特性存活時間，所以用氮氣來作為表面處理的氣體是很好的選擇！

我們發現對於由氮化鈦作為電極的電容，在二氧化鈣薄膜裡面主要的捕捉電荷機制是電洞捕捉而不是電子捕捉。而這種行為可以用捕獲面積模型來適當敘述。特別的是，平帶電壓平移是由於陷阱填補而不是陷阱產生。而電洞捕捉為主要機制可以歸因於電洞由基板注入的機率遠大於電子由閘極注入的機率而這是因為氮化鈦電極的功函數造成電洞具有較短的穿透路徑。

我們也發現在二氧化鈣薄膜內的陷阱是「潛在的」，電流隨溫度的變化較小，但是經由一段時間的電壓壓迫，這些陷阱就會被活化而顯現出來，而表現出的電流機制主要以 Frankel-Poole 為主。而我們把這些陷阱再次填滿，電流隨溫度的變化又變小許多！

The Effects of Various Pre-treatments on the Electrical Properties of HfO₂ Gate Dielectrics

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Aggressive device scaling has led to thin (i.e., not more than 1.2 nm) silicon dioxide (SiO₂) gate dielectric in state-of-the-art CMOS technologies. As a result, static leakage power due to direct tunneling through the gate oxide has been increasing at an exponential rate. As technology roadmaps call for sub-10Å gate oxides within the next five years, a variety of alternative high-*k* materials are being investigated as possible replacements for SiO₂. The higher dielectric constant in these materials allows the use of physically thicker films, potentially reducing the tunneling current while maintaining the gate capacitance needed for scaled device operation.

Hafnium oxide (HfO₂) is one of the most potential high-*k* materials. It has a high dielectric constant, high barrier height (1.6eV for electrons, and 3.2eV for holes), and excellent stability. But due to its material properties, the oxygen in HfO₂ easily reacts

with Si substrate and forms a low k interfacial layer. This has significant effects on scaling ability and electrical characteristics. In our experiment, we employ high density plasma (HDP) NH_3 and N_2O plasma surface pre-treatments on Si substrate to form an ultra-thin interfacial layer. Our results show that samples with HDP pre-treatment result in surface damage and the leakage current does not meet device criterion. An oxide RTA is found to be necessary to repair the damage. Samples with plasma pre-treatment and RTA anneal depict lower leakage current, smaller hysteresis and frequency dispersion, and higher breakdown voltage. In addition, NH_3 pre-treatment also results in longer time to breakdown, larger charge to breakdown and better characteristics life time.

We found that the dominant charge trapping mechanism in the high-k gate stack is hole trapping rather than electron trapping. This behavior can be well described by the distributed capture cross section model. In particular, the flatband voltage shift (ΔV_{fb}) is mainly caused by the trap filling instead of the trap creation. The dominant hole trapping can be ascribed to a higher probability for hole tunneling from the substrate, compared to electron tunneling from the gate, due to a shorter tunneling path over the barrier for holes due to the work function of the TiN gate electrode.

We also found that the traps in HfO_2 virgin films are “latent”, so leakage current is essentially independent of temperature. After voltage stress, however, these traps are activated and the dominant current transport mechanism becomes Frenkel-Poole emission. If the existing hole traps are filled by proper voltage stress, the leakage is found to be essentially independent of temperature, similar to that of the virgin sample.

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Contents

Abstract (Chinese)	I
Abstract (English)	III
Acknowledge	V
Contents	VII
Table Captions	IX
Figure Captions	X
Chapter 1 Introduction	1
1-1 General Background	1
1-2 Motivation-Why HfO₂ by MOCVD ?	7
1-3 Organization of the Thesis	8
1-4 Reference	10
Chapter 2 The Effect of Various Surface Pre-deposition Treatments on the Properties of HfO₂ thin films	21
2-1 Introduction	21
2-2 Experimental procedure	22
2-3 Physical and Electrical Characteristics	24
2-3-1 C-V and I-V characteristics	25
2-3-2 Frequency dispersion for HfO₂ gate dielectric	27
2-3-3 Proper sweep range for C-V measurement	29

2-3-4 Hysteresis of C-V measurement	30
2-3-5 High resolution transmission electron microscope	31
2-4 Summary	32
2-5 Reference	33
Chapter 3 Reliability Issues and Charge Trapping Characteristics of	
HfO₂ Gate Dielectrics	55
3-1 Introduction	55
3-2 Experimental procedure	58
3-3 Results and Discussions	59
3-3-1 Reliability Results	59
3-3-2 The characteristics of hole trapping in HfO ₂ /SiO ₂ gate	
dielectrics stack with TiN gate electrode	60
3-3-3 Pre-existing charges and current transfer mechanism	63
3-4 Summary	66
3-5 Reference	68
Chapter 4 Conclusions and Suggestion for Future Work	86
4-1 Conclusions	86
4-2 Future Work	88
Vita	90

TABLE CAPTIONS

Table 1-1 2003 International Technology Roadmap for Semiconductors [1]. The color shade means the solution known and unknown for physical limit.

Table 1-2 Material properties of HfO_2 , ZrO_2 , and Al_2O_3 .

Table 1-3 Comparison of state-of-the-art high-k dielectrics

Table 1-4 Comparison of Deposition Techniques: Sputtering, ALD, MOCVD, and MBE.

Table 2-1 Hysteresis comparison of different annealing temperature with $\text{N}_2\text{O}/\text{NH}_3$ plasma pre-treatment by oxide RTA and spike RTA.

Table 2-2 Hysteresis comparison of different annealing temperature with $\text{N}_2\text{O}/\text{NH}_3$ plasma pre-treatment by oxide RTA and spike RTA.

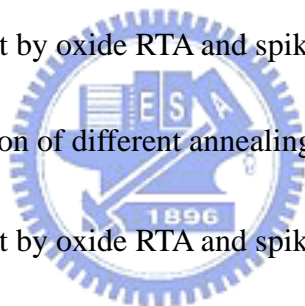


FIGURE CAPTIONS

Chapter 1

Fig. 1-1 The equivalent oxide thickness versus generation nodes for (a) microprocessor, and (b) low power.

Fig. 1-2 Energy gap versus dielectric constant for SiO_2 , Si_3N_4 , ZrO_2 , HfO_2 , ZrSiO_4 , and HfSiO_4

Fig. 1-3 Band alignments of topical high-k dielectrics [33]

Chapter 2

Fig. 2-1 A diagram of typical MOCVD system structure.

Fig. 2-2 The cross-section views of capacitor show LOCOS structure process flow.

Fig. 2-3 The (a) C-V and (b) I-V curves of N_2O pre-treatment with different N_2 RTA temperature.

Fig. 2-4 The (a) C-V and (b) I-V curves of N_2O pre-treatment with different N_2 spike RTA temperature.

Fig. 2-5 The (a) C-V and (b) I-V curves of NH_3 pre-treatment with different N_2 RTA temperature.

Fig. 2-6 The (a) C-V and (b) I-V curves of NH_3 pre-treatment with different N_2 RTA spike temperature.

Fig. 2-7 The (a) C-V and (b) I-V curves of 900°C N_2 annealing with $\text{N}_2\text{O}/\text{NH}_3$ plasma pre-treatment by oxide RTA and spike RTA.

Fig. 2-8 The (a) C-V curve and (b) Zoom in figure of (a) with RTO I.L. Sweep range is

(+1.5V, -1V) form inversion to accumulation.

Fig. 2-9 The (a) C-V curve and (b) Zoom in figure of (a) with RTO I.L. Sweep range is

(+1.5V, -2V) form inversion to accumulation.

Fig. 2-10 The (a) C-V curve and (b) Zoom in figure of (a) with RTO I.L. Sweep range

is (+1.5V, -1V) form accumulation to inversion.

Fig. 2-11 The (a) C-V curve and (b) Zoom in figure of (a) with RTO I.L. Sweep range

is (+1.5V, -1V) form accumulation to inversion.

Fig. 2-12 The C-V curve with different frequency sweep and the equation shows the

definition of frequency dispersion percentage.

Fig. 2-13 The (a) C-V curve and (b) Zoom in figure of (a) with RTO I.L. Sweep range

is (+0.5V, -0.5V) for one cycle sweep (first from inversion to accumulation and then from accumulation to inversion).

Fig. 2-14 The (a) C-V curve and (b) Zoom in figure of (a) with RTO I.L. Inversion

sweep range is increasing form 0.5 V to 2V with a step of 0.5.

Fig. 2-15 The (a) C-V curve and (b) Zoom in figure of (a) with RTO I.L. Accumulation

sweep range is increasing form -0.5 V to -2V with a step of 0.5.

Fig. 2-16 (a) The zoom in plot of the hysteresis of C-V curve with fixed inversion.

voltage and various accumulation. (b) shows the relationship between

flatband shift and accumulation voltage.

Fig. 2-17 The HRTEM image shows the as-deposited sample.

Fig. 2-18 The HRTEM image shows the sample with NH₃ pre-treatment and oxide RTA

N₂ 900°C annealing.

Fig. 2-19 The HRTEM image shows the sample with NH₃ pre-treatment and oxide RTA

N₂ 950°C annealing.

Fig. 2-20 The HRTEM image shows the sample with NH₃ pre-treatment and oxide RTA

N₂ 1000°C annealing.

Chapter 3

Fig. 3-1 (a) The J-V characteristics with different I.L. treatment method (RTO, N₂O,

NH₃) at same post-treatment annealing (900°C).

Fig. 3-1 (b) The Weibull plot shows the gate current densities at V_g = -1V with different I.L. treatment method (RTO, N₂O, NH₃) at same post-treatment annealing (900°C).

Fig. 3-2 (a) The TEM image of the HfO₂ with RTO I.L.

Fig. 3-2 (b) The HRTEM image of the HfO₂ with N₂O pre-treatment and 900°C N₂

RTA annealing.

Fig. 3-2 (c) The HRTEM image of the HfO₂ with NH₃ pre-treatment and 900°C N₂

RTA annealing.

Fig. 3-3 The Weibull plot shows the effective breakdown field with different I.L.

treatment method (RTO, N₂O, NH₃) at same post-treatment annealing (900°C).

Fig. 3-4 (a) The Weibull plot shows the time to breakdown under three constant stresses voltages with N_2O pre-treatment followed by $900^\circ C$ post-nitridation annealing.

Fig. 3-4 (b) The Weibull plot shows the time to breakdown under three constant stresses voltages with NH_3 pre-treatment followed by $900^\circ C$ post-nitridation annealing.

Fig. 3-4 (c) The Weibull plot shows the time to breakdown under three constant stresses voltages with RTO interfacial layer.

Fig. 3-5 The Weibull plot shows the time to breakdown under $-4.6V$ stresses voltage with different I.L. treatment method (RTO, N_2O , NH_3) at same post- treatment same post- treatment annealing ($900^\circ C$).

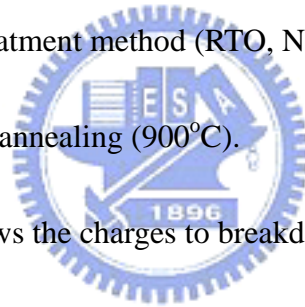


Fig. 3-6 The Weibull plot shows the charges to breakdown under $-4.8V$ stress voltage with different I.L. treatment method (RTO, N_2O , NH_3).

Fig. 3-7 (a) C-V curves and (b) G-V curves measured at 100 kHz with increasing stress time as a parameter. The stress voltage (V_g) was $-3.5 V$. The curve labeled $t = 0 s$ corresponds to the data before stressing.

Fig. 3-7 (c) I-V curves measured with increasing stress time. The stress voltage (V_g) was $-3.5 V$. The curve labeled $t = 0 s$ corresponds to the data before stressing.

Fig. 3-8 Flatband voltage shift v.s. time for RTO I.L. under different constant voltage stress (CVS). Symbols are experimental data, and solid lines are fitting curves.

Fig. 3-9 Energy diagram of $\text{HfO}_2/\text{SiO}_2$ gate stack capacitor with mid-gap TiN metal gate electrode under constant voltage stress ($V_G = -4.2\text{V}$).

Fig. 3-10 Gate current density as a function of injection charge density for various voltages.

Fig. 3-11 Flat band voltage shift v.s. injection charge for RTO I.L. under different Constant voltage stress (CVS).

Fig. 3-12 Injection charge v.s. Time for RTO I.L. under different Constant voltage stress (CVS).

Fig. 3-13 Trapped charge v.s. time for RTO I.L. under different constant voltage (CVS).

Fig. 3-14 Trapped charge v.s. injection charge for RTO I.L. under different constant voltage stress (CVS).

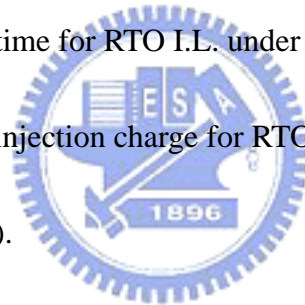


Fig. 3-15 Trapping efficiency v.s. Injection Charge for RTO I.L. under different constant voltage stress (CVS).

Fig. 3-16 (a) Flat band voltage versus stress time under positive constant voltage stress, and (b) Flat band voltage shift versus stress time under dynamic stressing with stress and passivation duration both 1000 sec.

Fig. 3-17 (a) The J-V curves measurement under various temperatures from 25°C to 150°C before stress and (b) is after -4.0V 600s stress.

Fig. 3-18 (a) $\ln(J/E)$ v.s. $1000/T$ under different $\text{HfO}_2/\text{SiO}_x$ stack voltage, and the

symbols are experiment data and the solid line are fit curve. (b) Z v.s. $E^{1/2}$,

Z is the slopes of the curves of (a) and solid curve is fitting line.

Fig. 3-19 Energy band diagram of $\text{HfO}_2/\text{SiO}_x$ stack to illustrate the conduction mechanism of Frankel-Poole emission.

Fig. 3-20 The I-V curves measurement under various temperatures from 25°C to 150°C after -4.0V 2300s stress.

