## 國立交通大學

### 電子工程學系 電子研究所碩士班

### 碩士論文

銦鎵掺雜之非晶態氧化鋅薄膜電晶體在環境與照光偏壓下穩定度 之研究

Study of Environment Effects and Bias Illumination Stability for Amorphous Indium-Gallium-Zinc-Oxide Thin Film Transistors

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## 銦鎵摻雜之非晶態氧化鋅薄膜電晶體在環境與 照光偏壓下穩定度之研究

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近年來,隨著液晶顯示器的尺寸越來越大,用來使液晶旋轉的電晶體所需要 的電子遷移率(mobility)也必須越來越高,但是傳統的非晶矽薄膜電晶體的電子遷 移率太低(<1 cm<sup>2</sup>/V s),因此擁有高電子遷移率(<10 cm<sup>2</sup>/V s)的非晶金屬氧化物 薄膜電晶體對於未來顯示器的應用上非常有潛力,故銦鎵摻雜之非晶態氧化鋅 (α-InGaZnO)薄膜電晶體是我們所要研究的重點。

雖然α-IGZO薄膜電晶體有非高的電子遷移率,但往往會受到環境、照光以 及長時間操作偏壓而對元件產生起始電壓(threshold voltage)的偏移,故我們的實 驗分成兩大部分來探討,第一部分主要是針對α-IGZO薄膜電晶體對氣氛的關係, 我們可以發現到氧氣在含水的環境中或者是較高的溫度下可以更有效率的吸附 在α-IGZO薄膜電晶體上,造成起始電壓向右飄移。第二部主要為測試元件在照 光以及負偏壓下不穩定性的探討,在照長波常以及負偏壓的條件下可以看到電容 有一個提早抬升的趨勢,我們將這個現象歸因於有donor-like 型態的缺陷產生於 介面,但是這個現象在照短波長以及負偏壓下並看不到,這是因為照短波長的光可以產生大量的電洞,這些電洞累積在介面,當在量測的時候這些累積的電洞屏 蔽掉了介面donor-like 型態的缺陷,導致在C-V的量測上沒有提早抬升的趨勢。



# Study of Environment Effects and Bias Illumination Stability for Amorphous Indium-Gallium-Zinc-Oxide Thin Film Transistors

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In recent years, the higher mobility is needed for thin film transistor (TFT) mainly used to twist the liquid crystal for the larger size display. The mobility of traditional amorphous silicon TFT is too low (< 1 cm<sup>2</sup>/V s), while the amorphous metal oxide has mobility higher than 10 cm<sup>2</sup>/V s, which is a very promising material for the future display application. Therefore, the amorphous Indium-Gallium-Zinc-Oxide ( $\alpha$ -InGaZnO) is the main topic that we want to research.

Although the  $\alpha$ -IGZO TFT has very high mobility, it is always suffer from the instability of environment, illumination and long term bias stress which cause the threshold voltage (V<sub>TH</sub>) shift. We can separate our study into two parts. The first part we want to discuss is the relation between the  $\alpha$ -IGZO TFT and oxygen. We found that oxygen absorbs on  $\alpha$ -IGZO is much more efficiently under water-containing or

higher temperature condition, causing a positive threshold voltage shift. The second part is mainly the examination of  $\alpha$ -IGZO TFT instability under the negative bias stress with light. We can find that the stretch-out phenomenon of C-V transfer curve under negative bias stress with long wavelength of light. We contribute this phenomenon to the donor-like traps creation at the interface between gate insulator and channel. However, the C-V transfer curve stretch-out phenomenon disappeared under the negative bias stress with short wavelength of light. It may be due to the huge amount of holes generated through illumination with short wavelength of light. These holes could accumulate at the interface and screen the donor-like traps.



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### **Chapter 1** Introduction

### 1.1 Amorphous Oxide Semiconductor

Amorphous semiconductors have created a new area of electronics devices such as solar cells and active-matrix (AM) flat-panel displays. Single-crystalline semiconductor technology, typified by crystal silicon electronics, is unsuitable for such applications, whereas amorphous or polycrystalline films can be easily formed.

Due to carrier scattering and trapping at grain boundaries in polycrystalline materials (the grain boundary problem), hydrogenated amorphous silicon ( $\alpha$ -Si:H) has been used more widely than polycrystalline silicon (p-Si) for practical large-size applications. However, the critical obstacles to be overcome for  $\alpha$ -Si:H in future applications is low mobility (<1 cm<sup>2</sup>/Vs) and instability against electric stress and photo-illumination.

Amorphous oxide semiconductor (AOS) materials have superior characteristics to conventional semiconductors, and therefore AOS TFT technology has grown very rapidly toward the realization of TFT backplanes for next-generation flat-panel displays. AOS TFTs have several features that are attractive for flat-panel displays and large-area integrated circuits. The advantageous features of AOS TFTs are summarized as follows:

- Low processing temperature: AOS TFTs exhibit satisfactory operation characteristics even if fabricated at room temperature.
- **High electron mobility**: AOSs exhibit high Hall and TFT mobility of > 10 cm<sup>2</sup>/V s by choosing appropriate chemical compositions.
- Low operation voltage: Oxides have electronic structures specific to the ionic

chemical bonds, and therefore they form much fewer defect states in the band gap than conventional covalent semiconductors such as silicon. The low defect density allows small subthreshold swing and low operation voltages.

- Large allowance in the choice of gate insulator: a variety of gate insulators have been examined for AOS TFTs, and good operation characteristics have been demonstrated. This feature would also benefit for two reasons: the unipolarity of AOS, by which only electrons are mobile and thus the valence band offset between the gate insulator and AOS is not critical; and the high ionic chemical bonding nature, which gives rise to fewer defects at the gate insulator/AOS interfaces.
- Simple electrode structure and low off current: Silicon-based field-effect transistors require a p-n junction for the source and drain electrodes to suppress inversion operation and the consequent increase in off current. Since AOS TFTs have not exhibited inversion p-channel operation, simple metallic contacts may be used for source and drain electrodes without increasing the off current. [1.1]

### 1.2 Origin of High Electron Mobility for Amorphous Oxides Semiconductor

It is believed that the properties of amorphous semiconductors are considerably degraded compared with their corresponding crystalline phases, which is actually the case for silicon because intrinsic crystalline silicon (c-Si) exhibits an electron mobility of 1500 cm<sup>2</sup>/V s, which deteriorates to less than 2 cm<sup>2</sup>/V s in  $\alpha$ -Si:H. On the other hand, AOSs exhibit large electron mobilities of greater than 10 cm<sup>2</sup>/V s even in amorphous structures. It benefits from the strong ionicity of oxides. In silicon, the conduction band minimum (CBM) and valence band maximum (VBM) are made of

anti-bonding (sp3  $\sigma^*$ ) and bonding (sp3  $\sigma$ ) states of Si sp3 hybridized orbitals, and its band gap is formed by the energy splitting of the  $\sigma^*-\sigma$  levels as shown in Figure 1-2-1(a). By contrast, oxides have strong ionicity and charge transfer occurs from metal to oxygen atoms as shown in Figure 1-2-1(b), and the electronic structure is stabilized by the Madelung potential formed by these ions, raising the electronic levels in cations and lowering the levels in anions. Consequently, the CBM is primarily formed by the unoccupied s orbitals and the VBM of cations by fully occupied O 2p orbitals, as illustrated in Figure 1-2-1(c).

Conventional amorphous semiconductors such as  $\alpha$ -Si:H exhibit much deteriorated carrier transport properties compared with the corresponding crystalline materials. This is because the chemical bonds in the covalent semiconductors are made of sp3 or p orbitals with strong spatial directivity. By contrast, the CBMs of non-transition metal oxides are made of the spherically extended s orbitals of metal cations, and their overlaps with neighboring metal s orbitals are not altered appreciably by the disordered amorphous structure; therefore, the electronic levels of the CBM are insensitive to local structural randomness, and electron transport is not affected significantly as shown in Figure 1-2-2.

#### 1.3 The Promising Material of Amorphous Oxide Semiconductor

As the size and the resolution of digital television (TV) increase, switching devices with higher electron mobility are required because the charging time for each pixel decreases in addition to the increased RC delay in the signal lines. It is estimated that electron mobility over 3 cm<sup>2</sup> / V s is required for driving the ultrahigh-definition(7680 × 4320) TV at 120-Hz frame rate. The required mobility will be even higher for displays with higher resolution, a faster frame rate and a larger

panel size as shown in Figure 1-3-1. Recently, three-dimensional (3D) displays have appeared on the market with panel sizes of ~55 inches and frame rates of 240 Hz. However, higher frame rates of 480 Hz are required to improve the picture quality because a 3D display must project two or more picture frames alternately for the left and right eyes.

On the other hand, there are other applications of TFTs, such as "system-on-glass" and "system-on-panel" technologies, which are used in the mass-produced, small-size flat-panel displays in mobile phones. The circuits are currently constructed from poly-Si TFTs because these applications require high TFT mobility for high-speed driver/peripheral circuits, and complementary metal-oxide semiconductor logic circuits are used to achieve ultra-low power consumption. It has also been reported that amorphous amorphous Indium-Gallium-Zinc-Oxide ( $\alpha$ -IGZO) can be used as resistance random access memory (ReRAM). [1.2] AOS TFTs are also expected to be developed for memory, pixel driver and other peripheral circuits in the same flat panel once a larger Hall mobility of AOS exceeds 50 cm<sup>2</sup>/V s and other suitable property are realized. [1.1, 1.3]

### 1.4 Why Use $\alpha$ -IGZO?

The most essential feature of a semiconductor is that the carrier concentration should be controllable over several orders of magnitude. Oxide TFTs have been fabricated using ZnO, but the material known as a good transparent conducting oxide, contains a high density of mobile electrons (typically  $>> 10^{18}$  cm<sup>-3</sup>) even in nominally undoped states. This causes a serious problem for TFTs by making it difficult to control the channel conductance and the threshold voltage. This feature often results in instability of TFT characteristics.

The pure ZnO and In<sub>2</sub>O<sub>3</sub> form crystalline films even when deposited at room temperature. This is widely observed for many metals and oxides, for which it is difficult to obtain amorphous phases as shown in Figure 1-4-1. On the other hand, the binary oxide compounds, such as Zn-In-O and Zn-Ga-O, form amorphous phases if the mixing ratio of the oxides exceeds a certain threshold value. In general, mixing of two or more cations having different ionic charges and sizes is effective for enhancing the formation of an amorphous phase and suppressing crystallization; this is the reason why AOSs are basically multi-component systems, and this fact implies that we can find a rich variety of AOS materials.

Indium, gallium, zinc and tin are the major consistents of good transparent conducting oxides, and their unoccupied s orbitals form the electron transport paths in AOSs; therefore, having a majority of these ions is the fundamental requirement to obtain AOSs with large electron mobilities. To understand the role of the other ions, it is useful to compare systems such as  $\alpha$ -1ZO and  $\alpha$ -1GZO. As seen in Figure 1-4-2, a-IZO has higher electron mobilities than  $\alpha$ -1GZO, but is much more difficult to control at the low electron concentrations required for TFTs. Ga-O bonds are much stronger than In-O and Zn-O bonds, meaning that the incorporation of gallium suppresses the formation of oxygen deficiencies and the consequent generation of mobile electrons. On the other hand, the incorporation of high gallium content deteriorates the electron mobility. Therefore, the addition of an appropriate amount of a stabilizer ion that forms a strong chemical bond with the oxygen ion is important to obtain stable AOS materials and TFTs. [1.1], [1.3]

Although there are many pros for  $\alpha$ -IGZO, it still needs to solute some other issue for  $\alpha$ -IGZO. The instability of metal oxide TFTs remains as an obstacle to overcome for practical applications to electronic devices. The device stability is sensitive to bias/current stress, temperature, light illumination, and water vapor exposure.

The unit pixel circuit of an active matrix organic light-emitting diode (AM-OLED) is basically comprised of two transistors, the switching and driving transistors. The current flowing through the OLED is controlled by a drive thin-film transistor (TFT). During the programming cycle, the gate of the drive TFT is charged to a programming voltage through a switch TFT to control the luminance of the pixel during the rest of the frame time. In the case of the switching transistor, the variation of the mobility and threshold voltage is less important. However, the brightness of each pixel is highly dependent on the drain current of the driving transistor. For example, a threshold voltage (V<sub>TH</sub>) shift of the driving transistor causes a change in the resulting luminance of OLED diodes. Thus, the V<sub>TH</sub> shift in  $\alpha$ -IGZO which is ZnO-based transistors should be minimized.

6



Figure 1-2-1 Band gap formation mechanisms in (a) covalent and (b,c) ionic semiconductors. [1.1]



H. Hosono et al., J. Non-Crystalline Solids 203, 334 (1996).

**Figure 1-2-2** Schematic orbital drawings for the carrier transport paths (conduction band bottoms) in crystalline and amorphous semiconductors.





**Figure 1-4-2** Electron transport properties of IGZO thin films. [1.1]

### **Chapter 2** Motivation

#### **2.1 Illumination instability**

 $\alpha$ -IGZO TFT is usually used as driving the current which is through the AM-OLED. However, for such device in the flat panel is almost always exposed to back light. This would cause a negative threshold voltage (V<sub>TH</sub>) shift of  $\alpha$ -IGZO TFT to affect the brightness of AM-OLED. Hence, the instability of illumination is the quit important issue to consider.

There are several models to explain the instability of illumination. Most general thinking is that the  $\alpha$ -IGZO suffer from the light illumination would generate electron-hole pair. Because the energy band gap of  $\alpha$ -IGZO is about 3.0eV [2.1] and the wavelength for visible light is about 400~700 nm (energy is about 1.77~3.1eV), once the energy of light larger than the bandgap of  $\alpha$ -IGZO, the electrons in the valence band would have chance to excite to conduction band and contribute to the free electrons. This will lead to the increase in conductivity and the decrease in threshold voltage (V<sub>TH</sub>).

Takechi et al [2.2] attribute the instability to the structure change during the illumination and the possible explanation for the structure change is the generation of oxygen vacancies, which provide free electrons and interstitial oxygen atoms as show in Figure 2-1-1. The significant threshold voltage ( $V_{TH}$ ) decrease is owing to those free electrons generated mainly by this way. Moreover, when switching off the light, it could find that slow recovery of threshold voltage ( $V_{TH}$ ). It results from the reversible structure change which involves the recombination of oxygen vacancies, free electrons and oxygen atoms. Through illumination, it offers sufficient energy for oxygen atom to diffuse into an interstitial site. However, after switching off light, the

interstitial oxygen atoms require relatively high energies to diffuse back to original site.

Lee et al [2.3] - [2.4] also indicate that the illumination of  $\alpha$ -IGZO would cause the metastable donor state with long decay time. Due to  $\alpha$ -IGZO contains the weakly bounded oxygen ions with small formation energy of an oxygen deficiency as shown in Figure 2-1-2. Ghaffarzadeh et al [2.5] show that the recovery time is dependent on the temperature when switching off light. With the higher temperature the shorter recovery time is needed which means the drain current decay rapidly due to the quick negative threshold voltage (V<sub>TH</sub>) shift as the time getting longer at higher temperature. It implies that structure relaxes or the metastable donor state vanishes sooner.

Another report [2.6] – [2.8] thought the decrease in threshold voltage (V<sub>TH</sub>) is due to the neutral oxygen vacancy (V<sub>0</sub>) of original  $\alpha$ -IGZO which creates a defect localized state in the band gap and doses not contribute to the conduction band. Under illumination, light could cause photoionization of neutral oxygen vacancies  $(V_0 \rightarrow V_0^+ + e^- \text{ or } V_0 \rightarrow V_0^{2+} + 2e^-)$ . The  $V_0^+$  and  $V_0^{2+}$  states have higher energy level in the bandgap so electrons could contribute to the conduction band, hence led to the decrease in threshold voltage (V<sub>TH</sub>). Besides, the slow recovery of threshold voltage (V<sub>TH</sub>) is thought to be that it should take several minute that the ionization oxygen vacancy ( $V_0^+ \cdot V_0^{2+}$ ) can decay to zero.

Park [2,9] thinks that the oxygen vacancy ( $V_0$ ) generated by UV light before negative bias stress operation would act as hole trap center in the bulk or at the semiconductor/gate insulator interface. These trap center could trap hole to become charged ( $V_0^{+2}$ ) when operating in the negative bias stress and diffuse toward the semiconductor/gate insulator boundary. This would provide positive charge buildup consequently and cause more severe negative threshold voltage ( $V_{TH}$ ) shift than the device without UV light illumination. Although these are so many models to explain the illumination instability of  $\alpha$ -IGZO TFT, there is still no clear way to solve this natural problem.





Figure 2-1-2 A possible atomicconfiguration model for metastable donor model. The large red sphere shows a weakly bonded oxygen ion, and the lightred sphere shows a metastable site. [2.3]

#### 2.2 Electrical Instability

The electrical instability for  $\alpha$ -IGZO TFT is a main issue in the application of flat-panel displays (FPDs). It is commonly used as switching TFT or drive TFT on FPDs for active-matrix liquid crystal displays (AM-LCDs) or organic light emitting diodes (AM-OLED). If the instability occurs in the switching TFT, that will affect the twist of liquid crystal and cause the problem of color rendering. On the other hand, if the instability occurs in the drive TFT, the current flow through the AM-OLED device will change. That would lead to variations in the respective pixel brightness. So the bias stress test for long-term reliability is an important factor in the TFT application.

Several reports has shown the electrical instability for  $\alpha$ -Si [2.10] – [2.14], ploy-Si [2.15] and amorphous oxide semiconductor [2.16] – [2.17] as active layer. The reason to explain instabilities for  $\alpha$ -IGZO TFT is the carrier trapping. Suresh [2.18] showed that the I-V transfer characteristic of  $\alpha$ -IGZO TFT, Figure 2-2-1, under a positive bias stress could cause a positive threshold voltage shift. This phenomenon results from electrons being trapped at the channel/dielectric interface or injected into the gate dielectric which screen the applied gate voltage, thus a larger positive gate voltage is required for the device to turn on, causing positive threshold voltage (V<sub>TH</sub>) shift. The electrons trapping model could also explain the hysteresis loop for the C-V back-to-back measurement (VG =–10 to 10, –20 to 20, and –30 to 30 V) which shows in Fig 2-2-2. It indicates the successively flat band voltage (V<sub>FB</sub>) shift due to higher positive voltage for each hysteresis loop in the return sweep.

On the other hand, unlike  $\alpha$ -Si TFT [2.12], the threshold voltage of  $\alpha$ -IGZO TFT under negative bias stress or the flat band voltage of forward C-V sweep is essentially no change. This means no positive charges (holes) can be trap at the channel/dielectric interface or get injected into the gate dielectric rather than positive charge cannot trap

at the channel/dielectric interface or get injected into the gate dielectric. There are two reasons explain why no hole in  $\alpha$ -IGZO. Nomura et al [2.19] found that high-density electron traps in the subgap (>10<sup>20</sup> cm<sup>-3</sup>) where exist above valence band maximum with the energy range of ~1.5 eV which is detected hard X-ray photoemission spectroscopy (HX-PES). Hence the Fermi level (E<sub>F</sub>) will pin at this energy level because the high-density electron traps are more than two orders of magnitude greater than a hole density (~10<sup>18</sup> cm<sup>-3</sup>) induced by a gate voltage in a typical TFT structure, i.e. those huge amount of electrons in the high-density traps cannot be remove by negative bias. On the other hand, in spite of the high-density electron traps could be remove completely, hole is localized due to the disordered atomic configuration in  $\alpha$ -IGZO break the coherent hybridization of O 2p orbitals which mainly form valence band [2.1].

The better describe the time evolution of threshold voltage which caused by the carrier trap at the channel/dielectric interface or get injected into the gate dielectric is the stretched-exponential model [2.13].

$$\Delta V_{th} = \Delta V_0 \left\{ 1 - exp \left[ -\left(\frac{t_{stress}}{\tau}\right)^{\beta} \right] \right\}$$
(2.2.1)

$$\Delta V_0 = V_{G(stress)} - V_{th(initial)} \tag{2.2.2}$$

$$\tau = \tau_0 \exp\left(\frac{E_\tau}{kT_{stress}}\right) \tag{2.2.3}$$

In the above equations, the  $\Delta V_{th}$  is the effective voltage drop across the gate insulator (or also called effective stress voltage);  $V_{th(initial)}$  is the initial threshold voltage;  $V_{G(stress)}$  is the stress voltage which is applied by gate;  $\tau$  represents the characteristic trapping time of carriers where the thermal activation energy is given by  $E_a = E_{\tau}\beta$ , with  $\beta$  being the stretched-exponential exponent;  $E_{\tau}$  is the average effective energy barrier that carriers in the channel need to overcome before they can enter the insulator, with  $\tau_0$  being the thermal prefactor for emission over the barrier. Fitting threshold voltage shift with stress time from equation (2.2.1) could get  $\tau$  and  $\beta$ , extract  $E_{\tau}$  by applying equation (2.2.3) to the  $\tau$  versus  $1/kT_{stress}$  plots. The  $E_{\tau}$ is extracted from  $\alpha$ -IGZO TFT by Fung. [2.20] for positive bias temperature stress (0.78 eV) is smaller than the value of negative bias temperature stress (2.16 eV). Fung. suggests that the electrons are experienced a lower energy barrier than holes do during the charge injection process near the channel/dielectric interface. However, (1) the electron transport in the conduction band is composed of s orbital which is different from the hole that transport in the valence band which is sp3 hybridized orbital. Carrier migration under amorphous condition in the s orbital is more easily than sp3 hybridized orbital due to the orientation. (2) According to previous study [2.4], the conduction band offset is usually larger than valence band offset. So the  $E_{\tau}$  means not only the carrier experienced energy barrier near the channel/dielectric interface but also the barrier which is carrier transport in orbital. Gelatos [2.12] extracted the  $E_{\tau}$  for positive bias stress is 0.98 eV and for negative stress is 1.03 eV form  $\alpha\text{-Si:H}$ TFT. The energy barrier for both electrons and holes needed to overcome to trap are quit equal. Hence the main dominant for carrier trapping mechanism seems the energy barrier which is carrier transport rather than the energy barrier of band offset.

Another possible mechanism for electrical instability is the shallow trap creation [2.21] (the subgap density of states in the energy region close to conduction band) under constant current stress by simulation method which showed in Figure 2-2-3. The formation of the extra shallow traps is considered to be related to the breaking of the weak chemical bonds and subsequent formation of defects such as vacancy defects and low valence state cations.



Figure 2-2-1 Comparison of the effect of a positive and a negative gate bias stress



Figure 2-2-2 Hysteresis sweeps -10 to 10, -20 to 20, and -30 to 30 V were made consecutively. [2.18]



Figure 2-2-3 Subgap DOSs extracted from the device simulations. [2.21]



#### 2.3 Environment Instability

Bottom-gate structured is common used for TFT because this kind of structure is easily formed owing to the deposition of channel layer and single mask step to form the source and drain electrodes. Several reports [2.22] – [2.26] showed the ZnO-based material is very sensitive to environment condition. The  $\alpha$ -IGZO is one of the ZnO-based material, once the back-channel surface is exposed to the atmosphere, the  $\alpha$ -IGZO TFT characteristics can be affect by the adsorption, desorption and diffusion of atmospheric gases causing instability.

The major gases in the air which affect the  $\alpha$ -IGZO TFT instability are oxygen. Some reports [2.27] – [2.28] indicated the oxygen  $\alpha$ -IGZO could be absorbed on the  $\alpha$ -IGZO TFT channel as shown in Figure 2.3.1 and lead to threshold voltage increase. The reason is that oxygen capture an electron form the conduction band of  $\alpha$ -IGZO and the resulting oxygen species can exist in O<sub>2</sub>. At the same time, because the electron is captured by oxygen form the  $\alpha$ -IGZO, the depletion region formed owing to the charge transfer and the positive space charge which appear in the  $\alpha$ -IGZO bulk, this would cause an upward band bending as shown in Figure 2-3-2. In such condition just like applied a negative bias in the back channel, so the larger threshold is needed to turn on the  $\alpha$ -IGZO TFT (like MOSFET body effect).

$$O_2(gas) + e^- \to O_2^-(solid) \tag{2.3.1}$$

The oxygen capture an electron form the  $\alpha$ -IGZO TFT channel can be described by the equation the equation (2.3.1). The equilibrium constant K is given by  $[O_2^-]_{solid}/P_{O_2}[n]$ , where  $[O_2^-]_{solid}$ ,  $P_{O_2}$ , and [n] are the adsorbed oxygen concentration on the  $\alpha$ -IGZO surface, the partial oxygen pressure, and the electron density in the channel, respectively. As increase in  $P_{O_2}$ , the concentration of  $[O_2^-]_{solid}$  increase because the equilibrium constant K should be invariant at a fixed temperature. Likewise, when gate is applied positive voltage, the accumulated electron density [n] in the channel region increase and the concentration of the adsorption  $[O_2^-]_{solid}$  is also increase. Both results in a positive threshold voltage shift.

On the other hand, much research for the effect of water  $(H_2O)$  adsorption on the  $\alpha$ -IGZO TFT has been studied. Park et al [2.29] reported that H<sub>2</sub>O can act as electron donor as shown in Figure 2-3-3 as well as acceptorlike trap when the water absorbed on the  $\alpha$ -IGZO TFT channel. Once H<sub>2</sub>O donates a negative charge to the  $\alpha$ -IGZO, forming a back channel layer with a high electron concentration, could lead to a more negative threshold voltage shift. Also, water could also form acceptorlike trap in the a-IGZO TFT bulk. This effect is proved by the different bulk thickness of a-IGZO TFT sample because the water molecular could diffuse into the  $\alpha$ -IGZO. For thicker  $\alpha$ -IGZO, the water molecular cannot diffuse into too deep region of  $\alpha$ -IGZO film. Although many acceptorlike traps are created, the created traps neither respond to the applied gate voltage as a trapping center, nor contribute to the gate swing change because the Fermi level near the back channel region is pinned when the gate voltage is applied. This also cause the electron could not be trapped in the acceptorlike defect. On the other hand, for the thinner  $\alpha$ -IGZO films, the degradation of the subthreshold swing was enhanced due to the created trap could full response to the applied gate voltage as shown in Figure 2.3.4.

Furthermore, oxygen and water are not only naturally absorbed/desorption by physical or chemical bonding but also related to the electrical field. Jeong et al [2.28] indicated that the oxygen and water could adsorbed/desorbed on the surface of the  $\alpha$ -IGZO film under the influence of an electric field electrical. The electric field induced adsorption of oxygen molecules from the ambient atmosphere under the application of positive gate voltage stress as shown in Figure 2.3.4 (a). However,

under the positive gate bias stress could lead to the water molecules desorption form the  $\alpha$ -IGZO film. It is an opposite trend to oxygen as shown in Figure 2.3.4 (b).

Recently, the zinc–tin oxide (ZTO) TFT which is also one of the ZnO-based materials showed negative threshold voltage ( $V_{TH}$ ) shift when exposed to various organic solvents. Such result can be explained by terms of polarity or electronegativity of the adsorbed solvent molecules. When solvent molecules are adsorbed on the back channel surface, extra electron carriers can be accumulated near the channel surface possibly due to electronegativity of the solvent molecules. If the adsorbed solvent molecule has higher polarity (higher dielectric constant) and thus higher electronegativity, then the number of accumulated carriers near the surface will be larger compared to solvent molecule with lower polarity. It is possibly due to more downward band bending near the back channel surface thus higher negative threshold voltage ( $V_{TH}$ ) shift caused by adsorption of high-polar solvent molecules [2.30]. Base on the same reason,  $\alpha$ -IGZO TFT may suffer such the same result under the same condition, the back channel effect which exposure to atmosphere is an essential problem.

In order to solve the  $\alpha$ -IGZO TFT instability problem, it is usually deposition another passivation layer such as SiO<sub>2</sub> [2.31], SiN<sub>X</sub> [2.32], Al<sub>2</sub>O<sub>3</sub> [2.33], TiO<sub>X</sub> [2.34] or organic material to protect the device back channel from the effect of environment.


Figure 2.3.1 The Schematic diagram of oxygen molecular adsorbed on the  $\alpha$ -IGZO TFT.



**Figure 2.3.2** The electron in the conduction band are cpatured by oxygen, the space charge are appaer causing an upward band bending.



**Figure 2.3.3** The Schematic diagram of water molecular donor elecrons to α-IGZO TFT. [2.29]



**Figure 2.3.4** The Schematic band diagram of water been as trap for (a) thinner (b) thicker.



Figure 2-3-5 Schematic showing the electric-field-induced (a) adsorption of oxygen molecules and (b) desorption of water molecules under positive gate bias stress. [2.28]



## **Chapter 3** Fabrication and Characterization

### 3.1 Fabrication Process of $\alpha$ -IGZO TFTs

Bottom gate coplanar  $\alpha$ -IGZO TFTs were produced on glass substrate in this work. The plasma enhanced chemical vapor deposition (PECVD) -derived SiO<sub>x</sub> (300 nm) film as gate insulator was grown at 370 °C, over the patterned Ti/Al/Ti (50/200/50 nm) trilayer gate electrodes. The Ti/Al/Ti (50/200/50 nm) source/drain electrodes were formed by sputtering and then patterned into the dimension of channel. A 30 nm thick  $\alpha$ -IGZO film was deposited by dc magnetron sputtering system at room temperature, using a target of In:Ga:Zn =1:1:1 in atomic ratio, a plasma discharge power of 300 W, and in an ambiance of gas mixture ratio of O<sub>2</sub> / Ar = 6.7% with a working pressure of 5 mTorr. After that the device was annealed in an oven at 330 °C for 2 hour as shown in Figure 3-1-1.

The device has the property that threshold voltage ( $V_{TH}$ ) is about -0.2V, field-effect mobility ( $\mu_{FE}$ ) is about 18.9 cm<sup>2</sup>/V s, subthreshold swing (S.S.) is about 0.2 V/decay, and the on/off ratio is over 10<sup>6</sup>.



### Bottom gate structure

**Figure 3-1-1** The device structure of  $\alpha$  -IGZO TFT.

#### **3.2 Methods of Device Parameter Extraction**

In this section, we will introduce the methods of typical parameter extraction such as the **threshold voltage** ( $V_{TH}$ ), **subthreshold swing** (S.S.), **field-effect mobility** ( $\mu_{FE}$ ) from the device characteristics.

Several methods are used to determinate the **threshold voltage**,  $V_{TH}$ , which is the most important parameter of the semiconductor devices. The method to determinate the threshold voltage in this thesis is the *constant drain current method*, the voltage at a specific normalized drain current (NI<sub>D</sub>) is taken as the threshold voltage. This technique is adopted in most studies of TFTs. It can give a threshold voltage close to that obtained by the complex linear extrapolation method. Typically, the specific normalized current NI<sub>D</sub> =  $I_D/(W/L)$  is defined at 10nA for V<sub>D</sub> operated in linear region and 100nA for V<sub>D</sub> operated in saturation region, to extract the threshold voltage of TFTs in most papers.

The **subthreshold swing** (*S.S*) is a significant parameter to describe the control ability of gate bias toward drain current and the efficiency of the switch turning on and off. It is defined as the amount of gate voltage required to increase/decrease drain current by one order of magnitude. It should be independent of drain voltage and gate voltage. However, in reality, the subthreshold swing might increase with drain voltage due to the short-channel effects such as charge sharing, avalanche multiplication, and punch through-like effects. It is also related to the gate voltage due to some undesirable factors such as serial resistance and interface state. In this experiment, the sub-threshold swing is defined as 10pA~10nA of drain current range about three orders of magnitude corresponds to the gate voltage range, and then take the inverse of the slope.

The field-effect mobility  $(\mu_{FE})$  is determined from the transconductance  $g_m$  at

low drain voltage (linear region). The transfer characteristics of  $\alpha$ -IGZO TFTs are similar to those of conventional MOSFETs, ignoring any other non-ideal effect and assuming the electric field in the channel is uniform, so the first order I-V relationship in the bulk Si MOSFETs can be applied to the  $\alpha$ -IGZO TFTs, which can be expressed as

$$I_{D} = \mu_{FE} C_{ox} \frac{W}{L} [(V_{G} - V_{T}) V_{D} - \frac{1}{2} V_{D}^{2}]$$
(3.2.1)

where  $C_{ox}$  is the gate oxide capacitance per unit area

*W* is channel width

*L* is channel length

 $V_T$  is the threshold voltage.

If  $V_D$  is much smaller than  $(V_G - V_T)$  (i.e.,  $V_D \ll V_G - V_T$ ) and  $V_G > V_T$ , the drain current STERNE STE can be approximated as:

$$I_D = \mu_{FE} C_{ox} \frac{W}{L} (V_G - V_T) V_D$$
(3.2.2)  
The transconductance is defined as

$$g_m = \frac{\partial I_D}{\partial V_G} \Big|_{V_D = const.} = \frac{WC_{ox}\mu_{FE}}{L}V_D$$
(3.2.3)

Therefore, the field-effect mobility can be obtained by

$$\mu_{FE} = \frac{L}{C_{ox}WV_D} g_{\rm m} \tag{3.2.4}$$

The mobility value was taken from Equation(3.2.4) with maximum  $\mu_{\text{FE}}$ .

### Chapter 4 Result & Discuss

#### 4.1 Water Induce Oxygen Adsorption

 $\alpha$ -IGZO is one of the ZnO based material which is very sensitive to environment condition especially oxygen, water and other organic material as mentioned in chapter 2. We already know the oxygen tend to absorb on the  $\alpha$ -IGZO. Moreover, there are water (H<sub>2</sub>O), nitrogen, and oxygen in the air, so the ambient effect for  $\alpha$ -IGZO TFT is we would like to discuss in this section.

In order to discuss different ambient effect, we use the unpassivation  $\alpha$ -IGZO device which means the back channel is exposure to outside. First, we put the sample into the vacuum chamber and then illuminate the  $\alpha$  -IGZO TFT at higher temperature to let the oxygen which is already exist on the  $\alpha$  -IGZO desorption. Second, nitrogen, oxygen, oxygen through DI water (water-containing oxygen) was fed into the chamber until the chamber pressure reached 0 (vacuum), 1, 10, 100, 150, 610, 760 torr at 303K as shown in Figure 4-1-1. The electrical characterization was carried out using Agilent 4156C Semiconductor Parameter Analyzer in a vacuum chamber system. Figure 4-1-2 to Figure 4-1-4 is the I-V transfer curve (V<sub>D</sub>=10V) for nitrogen, oxygen, and water-containing oxygen respectively and Figure 4-1-5 is the threshold voltage shift which is extracted by the method which is already mentioned in the chapter 3 for each ambient condition. From the experiment result, we could figure out that as the partial pressure of nitrogen increased, the threshold voltage nearly unchanged. Because nitrogen is inert gas, it does not react with  $\alpha$ -IGZO easily which led to threshold voltage  $(V_{TH})$  nearly unchanged. As to oxygen, the larger positive threshold voltage (V<sub>TH</sub>) shift occurred when the partial pressure of oxygen increased, owing to oxygen could capture an electron from the conduction band of  $\alpha$ -IGZO and the positive space charge appeared (depletion region). The positive charge cause an upward band bending which means  $\alpha$ -IGZO TFT need larger gate voltage to turn on the transistor followed by the equation (2.3.1). In addition, when water is added to certain oxygen gases, the oxygen may be more easily adsorbed in the water-containing environment. Hence, we proposed a model to explain this effect. According to chapter 2 mentioned, water has two role for  $\alpha$ -IGZO TFT. One role is water could donor an electron to  $\alpha$ -IGZO TFT. As we know, the equilibrium constant unchanged under the same temperature. As the electron increased, the equation (2.3.1) would occur easily which cause more oxygen absorbed on  $\alpha$ -IGZO. The other role is water could act as defect for  $\alpha$ -IGZO, so it could trap electron from  $\alpha$ -IGZO to let the total conductance decreased. Base on the two role of water, the total net threshold voltage (V<sub>TH</sub>) shift for water-containing oxygen environment is larger than only oxygen.





Figure 4-1-1 The measurement system for different ambient condition.



Figure 4-1-2 The I-V transfer curve under different partial pressure of nitrogen.



Figure 4-1-3 The I-V transfer curve under different partial pressure of oxygen.





Figure 4-1-4 The I-V transfer curve under different partial pressure of water-containing oxygen.



Figure 4-1-5 The threshold voltage shift for different ambient.



#### 4.2 Oxygen Adsorption Would Passivate Trap

 $\alpha$ -IGZO is one of the ZnO based material which is very sensitive to environment condition especially oxygen, water and other organic material as mentioned in chapter 2. However the 21% of ambient atmosphere is oxygen, once the  $\alpha$ -IGZO thin film transistor (TFT) back channel expose to ambient atmosphere that will suffer huge instability for the transfer curve. Therefore we like to discuss the  $\alpha$ -IGZO TFT about the oxygen absorption degree at different temperature and the effect of subgap trap.

The device we use is without any passivation layer on  $\alpha$ -IGZO TFT which means the back channel is exposed to atmosphere, Hence, we could discuss the relationship between the  $\alpha$ -IGZO TFT and oxygen. The measurement procedure of each temperature as follow: First, we illuminate  $\alpha$ -IGZO TFT at higher temperature in the vacuum chamber and the pump is always pumping at the same time to clean some part of the oxygen which is already absorbed on the  $\alpha$ -IGZO TFT. The electrical characterization was carried out using Agilent 4156C Semiconductor Parameter Analyzer in a vacuum chamber system. Then, oxygen gas was fed into the chamber until the chamber pressure reached 1, 10, 100 and 760 torr as shown in Figure 4-2-1.

Figures 4-2-2 to Figure 4-2-5 show the V<sub>G</sub>-I<sub>D</sub> transfer curve of  $\alpha$ -IGZO TFT at 273 K, 303 K, 333 K and 363 K respectively under the 0 (vacuum), 1, 10, 100, 760 torr oxygen ambient. Note that, the positive threshold voltage (V<sub>TH</sub>) shift which was extracted by the method in chapter 3 is not obvious at the lower temperature while the threshold voltage (V<sub>TH</sub>) shift is apparent in the higher temperature as shown in Figure 4-2-6. It means that the oxygen of the atmosphere is easier absorbed on the channel of  $\alpha$ -IGZO TFT at higher temperature. This would cause positive threshold voltage shift (V<sub>TH</sub>) of  $\alpha$ -IGZO TFT. Because oxygen could capture an electron from  $\alpha$ -IGZO and the positive space charge would appear lead to the upward band bending.

In addition, the subthreshold gate swing from Figures 4-2-2 to Figure 4-2-5 for each transfer  $V_G$ -I<sub>D</sub> curve was extracted by the method which is already mentioned in the chapter 3 as shown in Figure 4-2-7. Note that, the subthreshold swing does not show any trend with the partial pressure of oxygen increased at each temperature and variation among the range about 0.07 V/decade. The subthreshold swing can be associated with the density of deep bulk state ( $N_{BS}$ ) and interface state at the interface between gate insulator and semiconductor layers ( $N_{SS}$ ) by the following formula (4.2.1) :

$$S.S. = \frac{2.3k_BT}{qlog(e)} \left[ 1 + \frac{qd_{ins}}{\varepsilon_{ins}} \left( \sqrt{\varepsilon_s N_{BS} + qN_{SS}} \right) \right]$$
(4.2.1)

where  $\mathbf{K}_{\mathbf{B}}$  denotes the Boltzmann constant;  $\mathbf{T}$  the temperature;  $\mathbf{q}$  the elementary electric charge;  $\boldsymbol{\varepsilon}_{ins}$  and  $\boldsymbol{\varepsilon}_{s}$  are permittivity in insulator and semiconductor, respectively; and  $d_{ins}$  is the effective thickness of the insulator; [4.1] – [4.3]. The almost unchanged subthreshold swing means the density of state in the oxygen absorbed of  $\alpha$ -IGZO is the same. However the formula (4.2.1) is an approximation method to estimate the average amount of density of state in  $\alpha$ -IGZO subgap. The value of the density of state would be dominated by the maximum value among the subgap, this value doesn't mean that all the density of state are the same in the  $\alpha$ -IGZO subgap energy level. Hence we use another method to extract the density of state of  $\alpha$ -IGZO under the condition which is oxygen absorbed on  $\alpha$ -IGZO at different temperature.

Reference shows the C-V measurement and numerical calculation method in which Kimura et al clamed to estimate the density of state in the subgap of  $\alpha$ -IGZO [4.4]. The C-V transfer curve is carried out using Agilent 4284A which the gate of the  $\alpha$ -IGZO TFT is connected to Capacitance Measurement High (CMH) and the source and drain is connected to Capacitance Measurement Low (CML). Figure 4-2-8 shows

the measured results of the C-V characteristic of  $\alpha$ -IGZO TFT. Here the characteristics at applied frequency of 40 kHz which is the minimum value that we could get the capacitance value. It is observed that capacitance become small as the gate voltage bias below flat band voltage (V<sub>FB</sub>) because the electrons are depleted in the  $\alpha$ -IGZO film which means only the source and drain overlap capacitance detected. On the contrary, owing to the electrons are accumulated at the channel of the  $\alpha$ -IGZO film, the capacitance become larger (overlap capacitance and channel capacitance) as gate voltage over flat band voltage

The trap densities were extracted using the following algorithm. First, the surface potential ( $\Phi_S$ ) is calculated from the C-V characteristic. By applying Q = CV to the gate insulator, differentiating it by gate voltage, and transforming it, equations (4.2.2) and (4.2.3) are acquired. Moreover, by integrating equation (4.2.3) of gate voltage, equation (4.2.4) is acquired. Here, C<sub>1</sub> is the geometrical capacitance of the gate insulator of channel region, C<sub>g</sub> is the measurement capacitance in the C-V characteristic subtracts the source and drain overlap capacitance which means the minimum value of the C-V characteristic and V<sub>FB</sub> is the flatband voltage, which corresponds to gate voltage where C<sub>g</sub> begins to increase as shown red dash line in Figure 4-2-8.

$$\mathbf{Q} = C_i (V_G - \phi_s) \tag{4.2.2}$$

$$\frac{\partial \phi_s}{\partial V_G} = 1 - \frac{\frac{\partial Q}{\partial V_G}}{C_i} = 1 - \frac{C_g}{C_i}$$
(4.2.3)

$$\phi_s = \int_{V_{FB}}^{V_G} (1 - \frac{c_g}{c_i}) \, dV_G \tag{4.2.4}$$

Surface potential is a function of gate voltage. Figure 4-2-9 shows the calculated result of the surface potential for  $\alpha$ -IGZO TFT form C-V measurement. It is found that surface potential is saturated at about 1 eV for high V<sub>G</sub>-V<sub>FB</sub> which indicates that the electrons are accumulated at the surface of the  $\alpha$ -IGZO film and that conduction

band edge approach the Fermi level as shown in Figure 4-2-10. Owing to the electrons are increased rapidly when the gate voltage larger than the threshold voltage, all the incensement of the gate bias above threshold voltage crossed over the gate insulator. Hence, the surface potential is no longer bending.

Next, we want to calculate the charge density in the channel layer by applying boundary condition and Poisson equation. Equation (4.2.5) is the boundary condition which means that the electric displacement is continuous between the vertical directions of gate insulator and semiconductor interface. Here,  $(\partial \phi / \partial x)_s$  is the surface potential gradient in the channel layer,  $\epsilon_i$  is the dielectric constant of the gate insulator,  $\epsilon_s$  is the dielectric constant of channel layer, and  $t_i$  is the thickness of the gate insulator.

$$\left(\frac{\partial \emptyset}{\partial x}\right)_{S} = \left(\frac{\epsilon_{i}}{\epsilon_{s}}\right)\left(\frac{V_{G} - \emptyset_{S}}{t_{i}}\right) \tag{4.2.5}$$

Equation (4.2.6) is the Poisson equation which means that the charge could determine the potential at each position alone the  $\alpha$ -IGZO bulk direction. We would assume the charge is nearly the same in any position in the  $\alpha$ -IGZO alone the  $\alpha$ -IGZO bulk direction and the thickness of the  $\alpha$ -IGZO is small so the integrate of equation (4.2.6) of each position is equation (4.2.7). Here, the d is the thickness of the  $\alpha$ -IGZO. Note that, the gradient of potential in the  $\alpha$ -IGZO at any position is the same so we could calculate from equation (4.2.5) and (4.2.7) to get the equation (4.2.8) which means the average charge density among the  $\alpha$ -IGZO for each gate bias voltage (i.e. each surface band bending).

$$\frac{\partial^2 \phi}{\partial x^2} = -\frac{\rho}{\epsilon_s} \tag{4.2.6}$$

$$\frac{\partial \phi}{\partial x} = -\frac{\rho}{\epsilon_s} d \tag{4.2.7}$$

$$\rho = -\varepsilon_i \frac{(VG - \phi S)}{d \cdot t_i} \tag{4.2.8}$$

Figure 4-2-11 shows the calculation result of the channel charge as a function of gate bias voltage. Note that the charge density  $\rho$  is saturate about 10<sup>-2</sup> coulomb/cm<sup>3</sup> which means the number of electron is 10<sup>17</sup> #/cm<sup>3</sup> when the gate voltage is 2V, this value consist with previous study result. On the contrary, there are tail state near the conduction band is about 10<sup>17</sup>~10<sup>18</sup> cm<sup>-3</sup> and deep state 10<sup>20</sup> cm<sup>-3</sup> which is 2.3 eV below the conduction band [4.5]. However the accumulation electron is only 10<sup>17</sup> #/cm<sup>3</sup> which means the relatively holes by gate bias is approximate this amount. When applied positive gate voltage Fermi level could be pinned near the conduction band owing to not too much electron could not fill this amount of density of state. On the other hand, when applied negative gate voltage Fermi level could be pinned at 2.3 eV below the conduction band due to the deep occupied state which is already full of electrons and it is not easy to exclude them.

The charge density  $\rho$  contain localized trapped electron, free electrons... and so on. The localized trapped electron only related to the density of state according to the equation (4.2.9). Here, the N is density of state of  $\alpha$ -IGZO subgap. So if differentiating the charge density  $\rho$  by the potential among the subgap of  $\alpha$ -IGZO, the density of state will be obtained as shown in Figure 4-2-12. Such C-V characteristic measurement range is among the subgap of  $\alpha$ -IGZO where the interval of the red dash line as shown in Figure 4-2-13 so the density of state which we calculate is belonged to this interval.

$$\rho = -q \int_0^{\emptyset} N \, d\emptyset \tag{4.2.9}$$

Figure 4-2-14 to Figure 4-2-17 show the C-V characteristic of  $\alpha$ -IGZO TFT at 273 K, 303 K, 333 K and 363 K respectively under the 0 (vacuum), 1, 10, 100, 760 torr oxygen partial pressure and Figure 4-2-18 to Figure 4-2-21 show the result of density of state based on the Figure 4-2-14 to Figure 4-2-17 C-V characteristic

measurement by the method which is mention above. Note that, the density of state nearly unchanged when the oxygen absorbed on the  $\alpha$ -IGZO under 273K and 303K while the density of state decrease when the oxygen absorbed on the  $\alpha$ -IGZO under 333K and 363K. This means at higher temperature not only oxygen absorption easier but also oxygen could passivation the trap among the  $\alpha$ -IGZO lead to the density of state which we calculated decreased.

To further confirm the oxygen absorbed on the  $\alpha$ -IGZO could compensate trap of  $\alpha$ -IGZO. We take two conditions into consideration. One is the  $\alpha$ -IGZO TFT at 363 K under vacuum chamber; the other is also at 363 K under vacuum chamber but after the oxygen absorbed on the  $\alpha$ -IGZO. The reason for the chamber temperature at 363 K is that according to the previous experiment the oxygen absorbed efficiently ona-IGZO at 363 K. Moreover, both conditions which measured in the vacuum chamber are for the purpose to exclude the field induced oxygen absorption on the  $\alpha$ -IGZO to affect our measurement result. The capacitance of  $\alpha$ -IGZO TFT was subject to back-to-back hysteresis loop sweeps  $V_G = -5$  to 4, -5 to 8, -5 to 12, -5 to 16 and -5 to 20 V which we fix the negative side at -5 V and vary positive side to consider the electron trapping effect. Here, we do not consider the negative bias side because it is not easy for  $\alpha$ -IGZO TFT to inverse hole. From Figure 4-2-22, we observe that the flat band voltage (V<sub>FB</sub>) shift successively to more positive voltage for each hysteresis loop during the return sweep. On the other hand, the device which the oxygen is already absorbed has lower flat band voltage (V<sub>FB</sub>) shift than the device which is without oxygen absorbed. The flat band voltage  $(V_{FB})$  shift result from three possible cases that electrons which are from  $\alpha$ -IGZO TFT channel could be trapped: (1) in the bulk or back channel of  $\alpha$ -IGZO TFT. (2) at the channel and dielectric interface. (3) injected into the dielectric. Due to the same device, the possible reason that could cause flat band voltage shift must not be electrons trapped at the channel

and dielectric interface or injected into the dielectric. Hence, the only one possible reason for the oxygen absorbed device which has lower flat band voltage ( $V_{FB}$ ) shift under back to back sweep is the trap of  $\alpha$ -IGZO decreased. As we sweep on the reverse loop, the Fermi level is from near conduction band to away conduction band. This would cause some electrons trapped in the  $\alpha$ -IGZO trap lead to the higher flat band voltage ( $V_{FB}$ ) as show in Figure 4-2-23. Base on this model, the lower flat band voltage shift means the lower trap of the oxygen absorbed  $\alpha$ -IGZO. Therefore, we could conclude that the oxygen absorbed on the  $\alpha$ -IGZO could compensate trap of  $\alpha$ -IGZO.



# **Measurement chamber**



Figure 4-2-1 The schematic of oxygen molecular fed into the measurement.



Figure 4-2-2 The I-V transfer curve for different partial pressure of the oxygen molecular which absorbed on  $\alpha$  -IGZO at 273K.



Figure 4-2-3 The I-V transfer curve for different partial pressure of the oxygen molecular which absorbed on  $\alpha$ -IGZO at 303K.



Figure 4-2-4 The I-V transfer curve for different partial pressure of the oxygen molecular which absorbed on  $\alpha$  -IGZO at 333K.



**Figure 4-2-5** The I-V transfer curve for different partial pressure of the oxygen molecular which absorbed on  $\alpha$  -IGZO at 363K.



Figure 4-2-6 The threshold voltage  $(V_{TH})$  shift under different oxygen partial pressure at different temperature.



Figure 4-2-7 The subthreshold swing (S.S.) shift under different oxygen partial



Figure 4-2-8 The C-V transfer curve of α-IGZO TFT.



Figure 4-2-9 The surface potential as a function of gate voltage.



**Figure 4-2-10** (a) The flat band condition (b) The maximum surface potential as gate voltage larger than threshold voltage ( $V_{TH}$ ) of  $\alpha$  -IGZO.



Figure 4-2-11 The maximum surface potential as a gate voltage larger than threshold



Figure 4-2-12 The density of state of of  $\alpha$ -IGZO TFT.



Figure 4-2-13 The C-V detects range among two red dash lines. [4.8]



Figure 4-2-14 The C-V transfer curve for different partial pressure of the oxygen molecular which absorbed on  $\alpha$  -IGZO at 273K.



Figure 4-2-15 The C-V transfer curve for different partial pressure of the oxygen molecular which absorbed on  $\alpha$ -IGZO at 303K.

C



Figure 4-2-16 The C-V transfer curve for different partial pressure of the oxygen molecular which absorbed on  $\alpha$ -IGZO at 333K.



Figure 4-2-17 The C-V transfer curve for different partial pressure of the oxygen molecular which absorbed on  $\alpha$ -IGZO at 363K.



Figure 4-2-18 The density of state for different partial pressure of the oxygen molecular which absorbed on  $\alpha$  -IGZO at 273K.



Figure 4-2-19 The density of state for different partial pressure of the oxygen molecular which absorbed on  $\alpha$ -IGZO at 303K.



Figure 4-2-20 The density of state for different partial pressure of the oxygen molecular which absorbed on  $\alpha$ -IGZO at 333K.



Figure 4-2-21 The density of state for different partial pressure of the oxygen molecular which absorbed on  $\alpha$ -IGZO at 363K.



Figure 4-2-22 The flat band voltage shift for back to back sweep under vacuum and vacuum which is oxygen already absorbed.



Figure 4-2-23 The schematic band diagram of flat band voltage ( $V_{FB}$ ) shift because of



#### **4.3** Negative Illumination Bias Stress Instability (NBIS)

Thin film transistors (TFTs), served as driving devices and switching devices for the active matrix liquid crystal display (AM-LCD) and the active matrix organic light emitting diodes (AM-OLED), have been universal applied. Since most of the proposed uses of the TFT will expose the TFT to a backlight or ambient light during operation, the black matrix (BM) integrated in the pixels is usually utilized, which can obstruct the formation of the photo-leakage current. The aperture ratio is the ratio between the transparent area, excluding a pixel's wiring area and transistor area (usually hidden by black matrix), and the whole pixel area. As the open area ratio gets higher, the light penetrates more efficiently, which results in the brighter images for the display, with the same magnitude of back-light source. Several technologies of the wiring and element design improved the aperture ratio by reducing the black matrix area, thus increasing the projector's luminosity. Hence, if the integrated black matrix can be fully removed from the pixel of flat panel display, the highest aperture ratio might be achieved. The photo-leakage current of the  $\alpha$ -IGZO TFTs obtained from our experimental results exhibited no significant variation, while a remarkable negative shift of threshold voltage is observed. The reason of this finding has been explained by several models as mentioned in Chapter 2. However, the  $\alpha$  -IGZO TFTs, used as switching devices, usually working at negative gate bias side for most of the time to ensure that the transistor is under the "off-state". For application, if the device did not display at fully off-state condition, it would possibly cause the image persistence or bright dot. Hence, the instability of negative bias under illumination for the  $\alpha$  -IGZO TFTs is a emergent issue we need to discuss in this report [4.6].

The three light sources with different wavelength, which are produced and classified by the halogen lamp passing through the color filters, are used to illuminate

on  $\alpha$ -IGZO TFTs. The wavelengths are 653 nm (with a bandgap of 1.9 eV), 556 nm (2.23 eV) and 500 nm (2.48 eV), respectively, which is measured by USB2000. Besides, the above light intensity received by the  $\alpha$ -IGZO surface are controlled at the same magnitude.

Biases of between -30 and 20 V were applied to the gate contact and 10 V were applied to the drain contact for various periods up to 2000 sec and all the electrical measurements of current-voltage (I-V) were carried out in vacuum chamber at 303 K using an Agilent 4156C Semiconductor Parameter Analyzer. The capacitance-voltage (C-V) characteristics of  $\alpha$ -IGZO TFTs is measured by Agilent 4284A, where the source and drain electrodes connect to Capacitance Measurement Low (CML) and the gate electrode connect to Capacitance Measurement High (CMH). The C-V characteristic is acquired at the frequency of 200 kHz by the gate voltage sweeping from -20 V to 5 V and -25V to 0V.

Figure 4-3-1 to Figure 4-3-6 show the Vg-Ip and C-V electrical characteristics of  $\alpha$  -IGZO TFTs, illuminated by the above three light sources with different wavelength and measured at 0, 100, 300, 600, 1000 and 2000 sec to record the electrical characteristics. The threshold voltage exhibited almost the same values as the illumination was performed by the light sources with the wavelengths of 653 nm and 556 nm, as seen in Figure 4-3-1 and Figure 4-3-3, while a large negative threshold voltage shift was observed from Figure 4-3-5 as the  $\alpha$ -IGZO TFT was illuminated by a light source with wavelength of 500 nm. As a wide band gap semiconductor for  $\alpha$  -IGZO (3 eV) [4.7], corresponding to the light wavelength of 413 nm, it is transparent to most of the visible wavelengths (400~700 nm), usually called the transparent semiconductor. Thus, the lights with wavelengths longer than 413 nm should not be absorbed by  $\alpha$  -IGZO film and not affect the  $\alpha$  -IGZO TFT characteristics theoretically, i.e. our experimental light with wavelength of 500 nm

cannot penetrate through  $\alpha$  -IGZO film and influence the characteristics of  $\alpha$  -IGZO TFTs. Nevertheless, the abnormal phenomenon was observed in our experimental results, in which the energy of light was absorbed by electrons, which were excited to the conduction band minimum, resulting in a negative threshold voltage shift.

Previous study [4.8] indicated that there are high density of state (about  $10^{20}$  cm<sup>-3</sup>) in the  $\alpha$ -IGZO bandgap, which is 2.3 eV below the conduction band minimum, and shallow defects of  $10^{17}$  cm<sup>-3</sup> often related to electrical characteristic, which is about 0.2 eV below the conduction band minimum as shown in Figure 4-3-7. The origin of the high density of state near the valance band maximum is that the oxygen vacancy state (V<sub>0</sub>) keeps a large space, which traps two electrons and forms a deep fully-occupied state. On the contrary, the shallow defect states are due to the oxygen deficiency, which does not form a large space, and therefore electrons cannot be trapped at V<sub>0</sub> site, which results in the doping of free electrons to conduction band minimum.

In order to investigate these abnormal findings about the light absorption of  $\alpha$  -IGZO film, we proposed simple band diagrams to explain the reasons. As seen in Figure 4-3-8 (a), as the light with wavelength of 653 nm is introduced, the electrons existing in the deep level cannot be excited to the conduction band minimum due to the lack of energy. Figure 4-3-8 (b) showed that the light energy with wavelength of 556 nm can excite the electrons through the successive two steps: one excitation occurs from deep levels to tail states; the other excitation appears from tail states to conduction bands. Due to the limitation of density of states for tail states is about 10<sup>17</sup> cm<sup>-3</sup>, the largest number of electron excited from deep levels to tail states is about 10<sup>17</sup> cm<sup>-3</sup>, which is not large enough to change the  $\alpha$ -IGZO conductivity and cause negative threshold voltage shift. Because the electron which is generated by light is slow and the amount could not be able to change the conductivity. As a result of the

light with wavelength of 500 nm is able to excite electrons from the deep level to conduction band minimum, as seen in Figure 4-3-8 (c), it would cause a huge negative threshold voltage shift. Accordingly, the light with wavelength of 500 nm can induce larger amount of electrons and holes in  $\alpha$  -IGZO than the others.

For the sake of electrical operation, electrical reliability of  $\alpha$  -IGZO TFT is also required to investigate, in which the illumination with the three lights of different wavelength (653, 556 and 500 nm) and gate bias stress of (-30 V + V<sub>TH</sub>) are applied simultaneously. Measurement of the electrical characteristic for V<sub>G</sub>-I<sub>D</sub> and C-V was carried out for a predetermined time for 0 s, 100 s, 300 s, 600 s, 1000 s, and 2000 s, as shown in Figure 4-3-9 to Figure 4-3-14. Figure 4-3-9 shows the invariant threshold voltage as the  $\alpha$ -IGZO TFT was experienced a negative bias illumination with wavelength of 653 nm for 2000 s, while the C-V curve was stretched-out seriously below the flat band voltage with the increasing stress time, as shown in Figure 4-3-10.

The characteristics variation of  $\alpha$ -IGZO TFT experienced a negative bias illumination with wavelength of 556 nm is larger than the variation with wavelength of 653 nm. Similarly, a negative threshold voltage shift is observed in Figure 4-3-11, while the C-V stretched-out phenomenon is not obvious in the bias stress under light illumination of 556 nm wavelength as shown in Figure 4-3-12. Furthermore, as the  $\alpha$ -IGZO TFT is illuminated by the light with wavelength of 500 nm and experienced a negative gate bias, the threshold voltage shift remains the same direction and is larger than the  $\alpha$ -IGZO TFT illuminated by the light with wavelength of 500 nm only, as seen in Figure 4-3-13. Interestingly, the C-V stretched-out phenomenon disappeared, shown in Figure 4-3-14, as the device underwent a negative bias with illumination of 500 nm wavelength.

Oh at al. [4.9] claimed that the origin of the C-V curve stretch-out and negative threshold voltage shift under negative bias illumination stress is due to the light,

which could generate the oxygen vacancy (V<sub>0</sub>) and induce the ionized oxygen vacancy (V<sub>0</sub><sup>+2</sup>). The V<sub>0</sub><sup>+2</sup> state, located at the interface by negative bias, acts as the positive fixed charge and therefore can screen the gate bias effectively. Since the photo-excited holes are "trapped" in the stable V<sub>0</sub><sup>+2</sup> states during the negative bias stress with visible light illumination, we considered that the photo-excited electrons stimulated from the V<sub>0</sub> state to the conduction band can be survived due to the lack of the direct recombination between the excess holes and photo-excited electrons. The above two mechanisms result in a larger negative shift in the transfer curves of  $\alpha$ -IGZO TFT. Furthermore, the another role of V<sub>0</sub><sup>+2</sup> states is the donor-like states, which is above the high density of states and 2.3 eV below the conduction band minimum. Consequently, the V<sub>0</sub><sup>+2</sup> states are positive without the occupying electrons at the ionized oxygen vacancy states would result in the stretch-out of C-V curve.

Such model is not appropriate to explain our experimental data, since the serious stretch-out of C-V curve dose not accompany a negative threshold voltage shift under negative bias illumination stress with light wavelength of 653 nm. However, the C-V stretch-out phenomenon gradually disappears and the negative threshold voltage shift is obvious as the negative bias illumination stress with light wavelengths of 556 and 500 nm is applied.

We also used the ISE-TCAD to simulate the C-V transfer curve of amorphous silicon ( $\alpha$ -Si) TFT. In this simulation, the donor-like interface defects of  $10^{12}$  cm<sup>-2</sup> below the intrinsic Fermi energy (E<sub>i</sub>) and presenting a uniform distribution within 0.2 eV (show in the inset of Figure 4-3-15) for  $\alpha$ -Si near the drain side is assumed. The Figure 4-3-15 shows that the stretch-out phenomenon of C-V curve appeared as the assumption of donor-like interface defects is introduced. Accordingly, such stretch-out phenomenon of C-V curve induced by negative bias illumination stress

can be ascribed to the donor-like defects generated by the gate bias stress. As seen in Figure 4-3-10, the two dash lines defined the C-V stretch-out region between A and B points, which is corresponding to a donor-like defects in the deep energy band gap of  $\alpha$  -IGZO, presented in Figure 4-3-16 (a). In the methodology of the C-V measurement, the C-V transfer curve we measured is the total capacitance of source/drain overlap and channel for  $\alpha$ -IGZO TFT. When the applied gate voltage is larger than the flat band voltage (V<sub>FB</sub>) of  $\alpha$ -IGZO TFT, the electrons accumulate gradually in the channel area. However, only the source/drain overlap capacitance is obtained with the gate voltage below the flat band voltage (V\_{FB}) of  $\alpha$  -IGZO TFT. Hence, the additional capacitance value appearing in the C-V stretch-out region can be related to the donor-like interface defects. We would assume the donor-like defects are among the whole channel but only the position where the donor-like defects near source and drain side could contribute to the C-V transfer curve stretch-out because of carrier from electrode have chance to react with those defects. On the other hand, due to the C-V transfer curve stretch-out phenomenon in the gate voltage below flat band voltage region, the carrier density in the channel is not high enough to react with defect that contribute to C-V transfer curve stretch-out as show in Figure 4-3-16 (b).

However, the stretched-out of C-V curve, which accompanied a larger negative shift of threshold voltage in the I-V curve, disappeared gradually when the negative bias illumination stress with the shorter light wavelengths (556 and 500 nm) were utilized. The different results between the lights with long wavelength (653 nm) and the shorter wavelengths (556 and 500 nm) is ascribed to the photon energy with wavelength of 653 nm cannot excite sufficient electron-hole pairs. These light-generated holes would attack the interface between the gate insulator and  $\alpha$ -IGZO bulk to form the donor-like interface traps with the application of the negative bias illumination stress. However, as the light with wavelength of 653 nm is applied,
there have no sufficient holes to be accumulated around the donor-like interface traps. Hence, the C-V transfer curve shows the stretch-out phenomenon without threshold voltage shift, as shown in Figure 4-3-17 (a). On the contrary, negative bias illumination stress with wavelength of 500 nm could generate large amount electron-hole pairs, which have been already explained in the experiment of illumination only for  $\alpha$  -IGZO TFT as shown in Figure 4-3-8. These large numbers of holes generated by the negative bias illumination stress could generate interface donor-like defects, and then these holes could also accumulate at the dielectric/channel interface causing a huge negative threshold voltage (V<sub>TH</sub>) shift as shown in Figure 4-3-17 (b). Besides, the donor-like trap is positive charged because of it above the Fermi energy level when device suffer negative gate bias stress. As we measure C-V transfer curve from -25V to 0V, the donor-like trap turn from above the Fermi level to below Fermi level. Hence, the donor-like trap tends to capture an electron to become neutral, this change cause an additional interface capacitance as shown in Figure 4-3-18 (a). On the contrary, we could not see the C-V transfer curve stretch-out in Figure 4-3-14, because donor-like trap is screened by holes accumulated at dielectric/channel interface. The donor-like trap could not capture an electron to become neutral, so no additional capacitance detected as shown in Figure 4-3-18 (b).

Figure 4-3-19 shows the recovery in the dark environment after negative gate bias stress with 500 nm wavelength illumination. We could found that the C-V transfer curve show stretch-out phenomenon form gradually as the recovery time getting longer. Because some part of holes which accumulated at dielectric/channel interface by negative gate bias stress with 500 nm wavelength illumination recombine with electron in the dark environment. The donor-like trap no longer screen by those holes, it could become neutral as below Fermi level and attribute to additional capacitance. Figure 4-3-20 which shows the recovery in the dark environment after

negative gate bias stress with 653 nm wavelength illumination. Because no sufficient holes accumulate at dielectric/channel interface the donor-like trap would not be screen. The C-V stretch-out remains the same in the recovery period.

Here, we proposed another model to explain the stretch-out phenomenon of C-V transfer curve disappear under negative bias stress with 500 nm wavelength of light due to the hole trapping in the gate insulator. Once the hole trapping in the gate insulator, the electrical field of insulator will enhance. Thus, the electric field in the channel will decrease which lead to the Fermi level cannot "go through" donor-like defects as shown in Figure 4-3-21. Hence, we could not measure the C-V transfer curve stretch-out. Both models are possible occur simultaneously and not contradictory.

#### 

The light wavelength (556 nm) is between the two other lights; both the C-V stretch-out and negative threshold voltage shift can be observed as the negative bias illumination stress with light wave length of 556 nm was introduced. It indicates that only part of the generated-hole accumulated at the interface causing partial donor-like trap be screened or some holes trapped in the gate insulator. As the experimental results Oh et al. have reported.



Figure 4-3-1 The I-V transfer curve under 653 nm wavelength of light illumination.





Figure 4-3-2 The C-V transfer curve under 653 nm wavelength of light illumination.



Figure 4-3-3 The I-V transfer curve under 556 nm wavelength of light illumination.





Figure 4-3-4 The C-V transfer curve under 556 nm wavelength of light illumination.



Figure 4-3-5 The I-V transfer curve under 500 nm wavelength of light illumination.





Figure 4-3-6 The I-V transfer curve under 500 nm wavelength of light illumination.



**Figure 4-3-7** The subgap density of state distribution for  $\alpha$  -IGZO. [4.8]



Figure 4-3-8 The schematic of electron in the deep trap under three different wavelength illumination condition.



Figure 4-3-9 The I-V transfer curve under 653 nm wavelength of light illumination





Figure 4-3-10 The C-V transfer curve under 653 nm wavelength of light illumination and  $V_{G}-V_{TH} = -30$ V.



Figure 4-3-11 The I-V transfer curve under 556 nm wavelength of light illumination



Figure 4-3-12 The C-V transfer curve under 556 nm wavelength of light illumination

and 
$$V_{G}-V_{TH} = -30V$$
.



Figure 4-3-13 The I-V transfer curve under 500 nm wavelength of light illumination



Figure 4-3-14 The IC-V transfer curve under 500 nm wavelength of light illumination

and 
$$V_{G}-V_{TH} = -30V$$
.



Figure 4-3-15 The ISE-TCAD C-V simulation for  $\alpha$ -Si TFT with donor-like state



Figure 4-3-16 (a)The schematic of donor-like trap generate region within the bandgap of a-IGZO under NBIS. (b) Only the donor-like trap near source and drain could contribute to C-V transfer curve stretch-out.



Figure 4-3-17 The schematic is (a) hole could generate donor-like interface trap (b) too much holes accumulate at the interface.



Figure 4-3-18 (a) the donor-like trap could become neutral as below Fermi level. (b) the donor-like trap could not become neutral as below Fermi level because the accumulated holes screen it.



**Figure 4-3-19** The C-V stretch-out appears in the recovery under dark environment after negative gate bias stress with 500nm wavelength of light.





**Figure 4-3-20** The C-V stretch-out appears in the recovery under dark environment after negative gate bias stress with 653nm wavelength of light.



# Chapter 5 Conclusion

## 5.1 Oxygen Absorption Effect

Oxygen adsorption is more efficiency under the water-containing circumstance, due to the electron donation from water to  $\alpha$ -IGZO, which enhanced the oxygen adsorption. On the other hand, the adsorbing capability of oxygen on the back-channel of  $\alpha$ -IGZO TFT were studied at various ambient temperatures. With the oxygen partial pressure increasing, positive threshold voltage shifts of  $\alpha$ -IGZO TFTs were observed, especially at the higher ambient temperatures. We estimate the density of states in the subgap of IGZO by C-V measurement and numerical calculation method, and we analyze the impact of oxygen adsorption on the electrical characteristics. As the ambient temperatures are at 0 and  $30^{\circ}$ C, the estimated density of states are nearly unchanged, while the density of states decrease with the adsorbed temperatures at 60 and 90°C. It implies that higher ambient temperatures can assist the oxygen to be adsorbed by  $\alpha$ -IGZO easily and can further passivate part of the traps at IGZO surface, leading to a decrease in the density of states. In addition, hysteresis measurements were conducted to ascertain the oxygen absorbed on the  $\alpha$ -IGZO, which can compensate part of the traps of  $\alpha$ -IGZO surface.

### 5.2 Negative Illumination Bias Stress Effect

Negative bias stress with 653 nm wavelength of light for  $\alpha$ -IGZO TFT could generate deep donor-like interface traps, which would result in the C-V transfer curve stretch-out. However, these donor-like traps disappeared under negative bias stress with 500 nm wavelength of light. We consider that these donor-like traps may be screened by holes accumulating at the dielectric/channel interface through illumination by short wavelength of light and results in a huge negative threshold voltage shift without C-V transfer curve stretch-out.



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### Chapter 4 Result & Discuss

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# Resume

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論文題目:

銦鎵摻雜之非晶態氧化鋅薄膜電晶體在環境與照光偏壓下穩定

度之研究

Study of Environment Effects and Bias Illumination Stability for Amorphous Indium-Gallium-Zinc-Oxide Thin Film Transistors