

Abstract

The as-quenched microstructure of the alloy A ($\text{Cu}_{2.9}\text{Mn}_{0.1}\text{Al}$) was D0_3 phase containing plate-like γ_1' martensite. When the manganese content was added to the 8 at%, the as-quenched microstructures of alloys B ($\text{Cu}_{2.8}\text{Mn}_{0.2}\text{Al}$) and C ($\text{Cu}_{2.7}\text{Mn}_{0.3}\text{Al}$) were changed to ($\text{D0}_3 + \text{L-J}$) phases. However, when manganese content was added over 8 wt. pct. further, the as-quenched microstructure of the Alloy D ($\text{Cu}_{2.6}\text{Mn}_{0.4}\text{Al}$) was a mixture of ($\text{D0}_3 + \text{L2}_1 + \text{L-J}$) phases. Thus, the martensite start (M_s) temperature was decreased with increasing the manganese content. On the contrary, the amount of the L-J particles was increased with increasing the manganese content.

Both of the $a/2\langle 100 \rangle$ and the $a/4\langle 111 \rangle$ APBs could be observed by the $\text{A2} \rightarrow \text{B2} \rightarrow \text{D0}_3$ continuous ordering transition during quenching in the both alloy B and alloy C. The $a/4\langle 111 \rangle$ APBs have never been found by other workers in the Cu-Mn-Al

alloys before. Besides, no evidence of the $a/4\langle 111 \rangle$ APBs could be observed in the as-quenched microstructure of the alloy D. It clearly shows that the $a/4\langle 111 \rangle$ APBs was increased with increasing the manganese content.

