# 國立交通大學

電子工程學系 電子研究所碩士班

## 碩士論文

矽鍺薄膜在不同氧化條件下之電性研究

A Study of Electrical Properties of SiGe Film with Various Oxidation Conditions

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- 指導教授: 張國明 博士
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中華民國 九十五年 九 月

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近年來,隨著元件尺寸縮小,增加載子遷移率已成為提升元件效能的趨勢之 一,而電子及電洞遷移率皆高的矽鍺合金在此情況下也自然而然地被廣泛研究。 藉由對矽鍺合金氧化,我們可以輕易提高鍺在矽鍺合金中的比例,進而造成更高 的電子及電洞遷移率。在本實驗中,我們對矽鍺薄膜做了多種不同條件的氧化製 程,並且製作成為傳統的金氧半場效電晶體(MOSFET),藉由電性的量測,我們 研究不同氧化條件對於矽鍺-金氧半場效電晶體的電特性影響。從實驗中我們發 現,氧化溫度越高、時間越長、氧氣流量越大的元件,表現出較佳的電流特性; 而在氧化速率的實驗中,氧化較慢的元件則有較低的漏電流。在文中,我們嘗試 找出造成這些現象的可能原因,並且探討出最佳的氧化條件。

# A Study of Electrical Properties of SiGe Film with Various Oxidation Conditions

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As channel length of metal-oxide-semiconductor field-effect transistors (MOSFETs) is deeply scaled down, enhancing the carrier mobility in the channel is desired for improving the performance of complementary MOS (CMOS) circuit. SiGe is a promising channel material because of its high electron and hole mobility. Higher Ge content in SiGe film makes higher mobility of both electrons and holes, and the oxidation process of SiGe is an easy way to achieve higher Ge content. In this study, various oxidation conditions were performed of SiGe film, and a conventional p-MOSFET was fabricated on it. I-V characteristics were measured to investigate the influence on electrical properties of the various oxidation conditions. It is found that the devices of higher oxidation temperature, longer oxidation time, and higher oxygen flow show better electrical performance. It is also found that devices under lower oxidation rate has lower leakage current and then induce higher On/Off ratio. Discussions of the results were made and the optimum condition of oxidation was therefore suggested.

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#### Chapter 1

#### Introduction

#### 1.1 An Overview of the Applications of High Mobility SiGe Alloy

There are considerable interests in enhanced p-type metal-oxide-semiconductor field effect transistors (MOSFETs) for complementary logic applications, because of the inferior performance of conventional p-type MOS devices owing to the lower hole mobility. As Ge has the highest hole mobility of all semiconductors (1900 cm<sup>2</sup>/Vs), it attracts much attention to Ge or SiGe based p-type MOSFETs. Compared with Si, the hole mobility of Ge is about 5 times higher [1-2].

Unfortunately, growing thermal oxide on Ge usually results in a high density of interface traps [3]. Additionally, defect-free Ge cannot be grown directly on Si substrate due to the 4.2% lattice mismatch between the two materials [4-5].

High Ge content SiGe channels have been grown on Si substrates, using a graded SiGe buffer layer. This technique successfully utilized to achieve higher mobility in p-type MOSFET [6].

#### **1.2 Ge Condensation Process in SiGe Film**

Conventionally, relaxed  $Si_{1-x}Ge_x$  film with high Ge content has been obtained by growing compositionally graded  $Si_{1-x}Ge_x$  much thicker than the critical thickness in order to introduce dislocations to relax strain in  $Si_{1-x}Ge_x$  layer [7]. Although the density of dislocations in the relaxed SiGe layer by this approach has been greatly reduced over time, the control of dislocations is still challenging and requires optimization for dislocation suppression. In addition, the thick  $Si_{1-x}Ge_x$  films need for strain relaxation through dislocations (on the order of a few microns, with a typical grading rate of 10% germanium increment per micron) poses a serious bottleneck for throughput [8].

An approach to achieve high Ge content in relaxed  $Si_{1-x}Ge_x$  layer has been reported that takes the advantage of the selective removal of Si atoms from SiGe film by  $Si_{1-x}Ge_x$  thermal oxidation [9]. When the supply of Si atoms diffused from the SiGe to the oxidation interface meets the consumption of Si atoms during oxidation, only Si atoms would be oxidized because Si oxidation is preferred to Ge oxidation [10], which leads to an increased Ge content in the  $Si_{1-x}Ge_x$  layer as the film is thinned down.

#### 1.3 Mechanism of SiGe Oxidation

The oxidation of SiGe films has been studied to a great extent [11-12]. This is mainly because of the potential technological applications of the SiGe [13], but also because it was demonstrated that small amounts of Ge in Si or SiGe layer greatly enhance the rate of oxidation [14]. In all cases reported so far, oxidation of SiGe alloys resulted in a pure SiO<sub>2</sub> layer, and a pileup of Ge at the oxide/SiGe interface [15-16]. Because of the large difference between the heat of formation of SiO<sub>2</sub> (-204 kcal/mol) and GeO<sub>2</sub> (-119 kcal/mol) [17], Si is more reactive than Ge while oxidation. As a result, the surface oxide is nearly pure SiO<sub>2</sub>, and Ge piles up underneath [19]. Some other features of SiGe oxidation are as follows: (1) During oxidation, Ge is completely rejected from the oxide. (2) There is no loss of Ge after oxidation. (3) The oxidation rate of SiGe in dry oxygen is mostly the same as that of pure Si. For wet oxidation, the rate of SiGe in the linear regime is 2 to 3 times higher than the rate of pure Si; but it is almost the same in the parabolic regime [19].

Taking Ge concentration into consideration, it is found that GeO<sub>2</sub> might form above pure SiO<sub>2</sub> of those SiGe films with higher Ge content [20]. Below 50 at.% Ge, oxidation remains similar to what has been described above. For alloys containing above 50 at.% Ge, a markedly different behavior is found: A two-layer oxide is formed, consisting of a mixed (Si, Ge)  $O_2$  layer near the surface, and a pure SiO<sub>2</sub> layer underneath [19]. According to the theory of binary alloy oxidation [21], at Si concentration lower than that at which Si becomes selectively oxidized (demonstrated to be 50 at.%), both Si and Ge will be simultaneously oxidized at the initial oxidation stage. As the oxidation proceeds, Ge gradually becomes depleted in the alloy phase in a zone next to the oxide/SiGe interface due to its higher growth rate [22], and the oxygen concentration is gradually getting lower. Eventually, oxide thickness will reach a critical value, which is proportional to the Ge concentration in SiGe film, such that the oxygen and Ge concentration at the oxide/SiGe interface are too low for Ge to be oxidized. At this stage a steady-state condition has been reached at which selective oxidation of Si succeeds and Ge piles up at the oxide/SiGe interface again. So that it is shown that Ge at the oxide/SiGe is in elemental form, while Ge at the oxide surface is in an intermediate oxidized form [23].

#### **1.4 Motivation**

SiGe oxidation is widely used in either SiGe/Si hetro-junction devices, poly-SiGe gated MOSs, or SiGe channel devices. However, the optimal oxidation condition is not known yet. In this study, we adopt different oxidation temperature, oxidation time, oxygen flow, and oxidation rate to investigate the influence on the electrical characteristics of SiGe-based p-MOSFETs.

I-V characteristics were measured by HP4156A. The optimization of oxidation

condition was also described to improve the electrical performance of the devices.

#### **1.5 Thesis Organization**

In chapter 1, a brief overview of SiGe and its oxidation mechanism were introduced.

In chapter 2, the fabrication process flow of SiGe-based p-MOSFET and the experimental recipes will be described.

In chapter 3, the electrical characteristics of the devices through different recipes will be shown, and discussions of the results of different recipes will be made.

In chapter 4, a conclusion is addressed.

Finally, in chapter 5, future works as well as suggestion for further research will

be given.

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#### **Chapter 2**

#### Experiment

#### 2.1 The Process Flow of SiGe-based P-MOSFET

P type (Boron doped) Si substrate (100) was used in this study. After initial RCA cleaning, 3000Å oxide was thermally grown at 980°C by horizontal furnace through wet oxidation, served as isolation layer. After one more time RCA cleaning, above is a 150Å thick amorphous-Si deposited by LPCVD at 550°C, as a buffer layer between SiO<sub>2</sub> and SiGe film. SiGe (Ge content at 7% and 11%) film was then deposited onto amorphous-Si layer about 800Å by UHV-CVD. The cross sectional view is shown in figure 2-1.

By means of photolithography, the active region could be defined after photo-resist has been removed. After S-D and channel patterning (top view shown in figure 2-2), TCP poly etcher was employed for SiGe etching by Cl<sub>2</sub> and HBr Later, the main part of this study proceeds by oxidizing the remaining SiGe part through different recipes. Different oxidation parameters like temperature, oxidation time, and oxygen flow were applied in order to find out the optimum oxidation condition of SiGe channel. Moreover, pre-deposited-oxide before oxidation, oxidation rate controlling were also performed trying to make better electrical characteristics performance of SiGe-based p-MOSFETs possible. All experimental factors are listed in table 2-1(A) to table 2-1(E).

Next, devices were dipped in BOE solution to remove surface  $SiO_2$  formed during oxidation of SiGe film. After 1000Å  $SiO_2$  deposited by PECVD served as gate dielectic, 600°C anneal with  $O_2$  was performed to cure the defects in gate oxide for 30min. Then, 2000Å poly-Si was deposited by LPCVD to work as control gate. After gate region and channel length defined by lithography, poly-etcher and BOE solution were used for etching. Cross sectional view and top view are shown in figure 2-3 and figure 2-4, respectively. Then, Boron was doped heavily with  $5 \times 10^{15}$  cm<sup>-2</sup> at 10 keV. Activation annealing at 950°C in N<sub>2</sub> flow was then employed for 30 minutes after ion implantation.

 $3000\text{\AA}$  SiO<sub>2</sub> by PECVD was deposited as passivation layer. Lithography comes next to form contact hole. SiO<sub>2</sub> was then etched by BOE solution for about 55 seconds. Finally, a 500 nm Al film was deposited by evaporation and then contact pad were patterned. Al sintering was performed at 430°C for 30minutes.

The detailed fabrication process flow is as follows:

- 1. (100)  $P^+$  Si wafer
- 2. RCA cleaning
- 3. Wet oxidation at 980°C for 3000Å
- 4. RCA cleaning
- 5. 150Å amorphous-Si by LPCVD
- 6. RCA cleaning
- 7. 800Å SiGe(Ge content at 7% and 11%) by UHVCVD
- 8. Mask #1:Define active region
- 9. Dry etching by TCP poly etcher
- 10. PR removing and RCA cleaning
- 11. SiGe dry oxidation
- 12. BOE dipped to remove  $SiO_2$
- 13. RCA cleaning
- 14. PECVD SiO<sub>2</sub> 1000Å deposition
- 15. Gate oxide annealing at  $600^{\circ}$  with  $O_2$  for 30min
- 16. LPCVD poly-Si 2000Å deposition

- 17. Mask #2:Define gate region and channel length
- 18. Wet etching by poly-etcher solution for 90 seconds
- 19. Wet etching by BOE solution for 25 seconds
- 20. PR removing
- 21. Boron doping of  $5 \times 10^{15}$  cm<sup>-2</sup> at 10 keV
- 22. Activation at 950°C for 30 minutes
- 23. STD cleaning
- 24. PECVD SiO<sub>2</sub> 3000Å deposition
- 25. Mask #3:Define contact hole
- 26. Wet etching by BOE solution for 50 seconds
- 27. PR removing
- 28. Al coating for 5000Å
- 29. Mask #4:Define contact pad
- 30. Al etching
- 31. PR removing
- 32. Sintering of Al at 430°C for 30 minutes

### 2.2 Methods of Measurement and Analysis

#### 2.2.1 Current-Voltage Characteristic Measurement

Current-Voltage characteristics were measured by a semiconductor parameter analyzer HP4156A at room temperature. Five devices of each sample were measured in order to choose a most reliable one to serve as result.

Id-Vg characteristic was measured first. Then  $g_m$ , on/off ratio and Vt could be calculated through Id-Vg characteristic data.  $G_m$  was defined as  $dI_d/dV_g$ , and the maximum value of  $g_m$  in each device was analyzed. Id-Vd characteristic was also



measure at a constant (Vg-Vt) value.

#### 2.2.2 The Introduction of HP4156A

HP4156A is an electronic instrument for measuring and analyzing characteristics of semiconductor devices. This instrument is useful for performing both results measurements and analyses. Id-Vg and Id-Vd characteristics of our experiment were measure by HP4156A.

There are four highly accurate source/monitor units (SMUs), two voltage source units (VSUs), and two voltage measurement units (VMUs) in HP4156A. HP4156A is designed for Kelvin connections and has high-resolution SMUs (HRSMUs), so HP4156A is especially suitable for low resistance and low current measurements. Voltage values with a resolution of  $0.2 \,\mu$  V can be measured by using the differential measurement mode of VMUs.

Stress testing can be also performed by HP4156A. That is, a specified dc voltage or current can be forced on the device for a specified duration. Also, we can force ac stress by using pulse generator units (PGUs), which were installed in HP41501A SMU/Pulse Generator Expander. HP41501A is attached to HP4156A, and was equipped with a ground unit (GNDU), high power SMU (HPSMU), two medium power SMUs (MPSMUs), or two PGUs.

HP4156A can print and store, in addition to perform measurement and analysis. Measurement setup information, measurement data, and instrument setting information can be stored in a 3.5-inch disk. The setting information and measurement results can be printed on a plotter or printer which is connected with HP4156A.

HP4156A can be controlled by an external controller via HP-IB by using remote control commands. These commands are based on Standard Commands for

Programmable Instruments (SCPI), so measurement programs can be easily developed. HP4156A has internal HP instrument BASIC, so we can develop and execute measurement programs by using the HP4156A only, without using an external controller.



#### Chapter 3

#### **Results and Discussions**

In the following chapter, all devices were normalized to  $W/L = 1 \ \mu \ m/1 \ \mu \ m$ . In Id-Vg and g<sub>m</sub>-Vg measurement, Vd was applied at -5V. While Id-Vd was measuring, |Vg-Vt| was set at 3V.

#### **3.1 Influence of Oxidation Temperature on Electrical Properties**

Figure 3-1(a) shows Id-Vg characteristic of three different devices: un-oxidized, oxidation at 950°C, and oxidation at 1000°C.  $Si_{0.89}Ge_{0.11}$  film was used in this experiment, and both of the oxidized devices were oxidized for 16 minutes with 3750 sccm  $O_2$  flow.

It can be seen that both oxidized devices show superior electrical performance than the un-oxidized one by higher on/off ratio. The device of 1000°C-oxidized shows even higher on current than the one of 950°C-oxidized one while both devices have roughly the same off state current.

Transconductance is shown in figure 3-1(b). Obviously,  $g_m$  of 1000°C-oxidized device is higher than that of 950°C-oxidized one and of course than the un-oxidized one. They are about 3.78  $\mu$  S, 2.64  $\mu$  S, and 0.33  $\mu$  S, respectively.

Figure 3-1(c) shows Id-Vd characteristic which is consistent with the prediction: the device oxidized at 1000°C has highest Id of the three devices -2.58  $\mu$  A at Vd=-6V. 950°C- oxidized device has 1.83  $\mu$  A, and the un-oxidized one has 0.40  $\mu$  A.

It is known that since SiGe got oxidized, concentration of Ge would be increased, and then mobility would also get enhanced which resulted in a higher transconductance and on state current. As the improvement of on state current is higher than the increasing of off state current, on/off ratio gets improved then. With higher oxidation temperature, the rate of oxidation would be higher, which makes more Si in SiGe film oxidized. Then the mobility would be even higher, and better performance is achieved. Table 3-1 shows the normalized data which is divided by the results of un-oxidized device.

#### **3.2 Influence of Oxidation Time on Electrical Properties**

Figure 3-2(a) shows Id-Vg characteristic of four different devices: un-oxidized, oxidized for 4 minutes, oxidized for 16 minutes, and oxidized for 36 minutes, respectively. Si<sub>0.93</sub>Ge<sub>0.07</sub> film was used in this experiment. All of the oxidized devices were oxidized at 1000°C with 3750 sccm O<sub>2</sub> flow. From the diagram, the trend of the curves indicates that longer oxidation time results in higher on state current and also higher transconductance (from Figure 3-2(b)). Besides, after calculation, the on/off ratio of 36min-oxidized, 16min-oxidized, and 4min-oxidized is 3.3, 1.6, 1.5 times higher than un-oxidized device, respectively. Due to the same reason as described in the previous section, in the longer oxidation time devices, more amount of Si was oxidized and then higher Ge concentration was achieved. So, similar trend of I-V characteristic as the previous section would be found in the diagram.

Figure 3-3(c) depicts Id-Vd characteristic for devices with different oxidation time. As predicted, 36min-oxidized device has highest on current, which is  $5.08 \,\mu$  A at Vd=-8V. For the other devices, they are  $1.52 \,\mu$  A,  $0.75 \,\mu$  A, and  $0.61 \,\mu$  A for 16min-oxidized, 4min-oxidized, and un-oxidized device, respectively. Table 3-2 shows the normalized data which is divided by the results of un-oxidized device.

It is presumed that unless Si in the SiGe layer is fully oxidized, the performance of the SiGe-based p-MOSFET would always be improved with the increasing of oxidation time since the positive correlation between the amount of oxidized Si and the mobility of SiGe channel.

#### **3.3 Influence of Oxygen flow on Electrical Properties**

Si<sub>0.89</sub>Ge<sub>0.11</sub> film was used in this experiment. Devices were oxidized at 1000°C for 16minutes. Figure 3-3(a) depicts Id-Vg characteristic of devices with different oxygen flow during oxidation: 3750 sccm, 2500 sccm, and un-oxidized. The device under larger oxygen flow shows larger on state current, lower off state current, and higher transcondutance, which is undoubtedly better than the device oxidized under 2500 sccm oxygen flow and the un-oxidized one. It is supposed to be contributed by more Si being oxidized, as discussed already. The on/off ratio of 3750sccm-device and 2500sccm-device are 5.89 and 1.86 times higher than the un-oxidized one. G<sub>m</sub>-Vg characteristic is shown in figure 3-2(b).

As to Id-Vd characteristic, which is shown in figure 3-3(c), the device with 3750 sccm  $O_2$  flow has 2.56  $\mu$  A at Vd=-6V. The device with 2500 sccm  $O_2$  flow and un-oxidized has 1.83  $\mu$  A and 0.43  $\mu$  A, respetively. Table 3-3 shows the normalized data which is divided by the results of un-oxidized device.

From 3-1 to 3-3, the amount of oxidized Si explans the improvement of electrical charateristic performance well. In the following three sections, several different oxidation condition were applied, and some other results would be achieved.

#### **3.4 Influence of the Thickness of Pre-oxide on Electrical Properties**

As it is known that oxidation rate would decrease as oxidation proceeds with time, a new experiment is designed to investigate the influence of the initial oxidation rate on electrical performance of SiGe-based p-MOSFET. SiO<sub>2</sub> was deposited onto SiGe film first by PE-CVD right after UHV-CVD SiGe film was deposited. The thickness of SiO<sub>2</sub> was 300Å, 500Å, and 1000Å, respectively, and a non-pre-oxide device was also fabricated. Si<sub>0.93</sub>Ge<sub>0.07</sub> was used in this experiment and oxidation was performed at  $1000^{\circ}$ C for 36 minutes with 3750 sccm O<sub>2</sub> flow.

Figure 3-4(a) depicts Id-Vg characteristic of devices with different thickness of pre-oxide. The on/off ratio is getting higher while the thickness of pre-oxide getting larger, which is 2.6, 1.9, 1.7 times higher than non-pre-oxide device, respectively. But in the other hand, the on state current decreases. In figure 3-4(b), it is found that  $g_m$  also gets lower with the increasing of pre-oxide thickness.

It is presumed that less amount of Si in SiGe film would be oxidized owing to thicker pre-oxide exists, so that results in lower on current and transconductance. As shown in figure 3-4(c), Id at Vd=-8 of non-pre-oxide, 300 Å, 500 Å, and 1000 Å are 5.46  $\mu$  A, 2.79  $\mu$  A, 1.07  $\mu$  A, and 0.95  $\mu$  A, respectively. Table 3-4 shows the normalized data which is divided by the results of un-oxidized device.

But there is still benefit from depositing pre-oxide. Since the oxidation rate was lowered by pre-oxide, a high quality channel was formed, which lowers the off state current, and results in a higher on/off ratio. The trade off between the on/off ratio and on state current should be considered case by case of the different use of the devices.

#### **3.5 Influence of Oxidation Rate on Electrical Properties**

In the previous section, it is concluded that under lower oxidation rate, SiGe-based p-MOSFET would achieve better on/off ratio performance. But the amount of oxidized of Si in last experiment was still a variable. In this section, the factor of the amount of oxidized of Si was removed by a new designed method. Several oxidation conditions were performed first and the thickness of  $SiO_2$  was measured. Three oxidation conditions of roughly the same thickness of  $SiO_2$  were selected. They are 950°C 15 minutes, 900°C 30 minutes, and 850°C 75 minutes, respectively, which indicates same amount of Si was oxidized.  $Si_{0.93}Ge_{0.07}$  was used in this experiment and oxidation was performed with 3750 sccm  $O_2$  flow.

Figure 3-5(a) depicts Id-Vg characteristic at Vd=-5V. On state current of the three devices almost equals, but the device with lower oxidation rate has lower off state current, which supports our conclusion from the previous section. As to  $g_m$ -Vg characteristic in figure 3-5(b), three devices also have about the same gm,  $0.5 \sim 0.6 \mu$  S. In figure 3-5(c), it is shown that the on state current of the three devices are also almost the same at about 2.1 ~ 2.3  $\mu$  A. Table 3-5 shows the normalized data which is divided by the results of un-oxidized device.

As prediction, same amount of oxidized Si results in same gm and Id. Slow oxidation makes channel quality higher and then lower the off state current, resulting in a higher on/off ratio.

#### Chapter 4

#### Conclusion

In this experiment, SiGe-based p-MOSFETs were fabricated and the electrical performance of the devices was improved by dry oxidation of SiGe channel through different recipes.

First, it is found that all the electrical characteristics such as on/off ratio, on state current, and transconductance would get improvement after the SiGe channel was oxidized. This is because after oxidation, Si atoms in SiGe channel would be combined with O atoms to form SiO<sub>2</sub> while Ge atoms would be separated from that. This is the so called Ge segregation mechanism. The more amount of Si in the SiGe film was oxidized, the more Ge atoms would exist in the SiGe channel and then makes Ge concentration higher which results in higher hole mobility in SiGe-base p-MOSFET. With the amount of oxidized Si increasing, the hole mobility also increases and then better electrical performance would be achieved. Experiments of oxidation temperature, oxidation time, and oxygen flow already proved this phenomenon.

Oxidation rate was also considered in our experiment. The results show that the devices under slower oxidation rate have lower leakage current and better on/off ratio. It is conjectured that under slow oxidation process a high quality channel was formed. This is why the devices have lower leakage current.

#### Chapter 5

#### **Future Work**

Since it is known that as the amount of oxidized Si increases, the electrical performance of SiGe-based p-MOSFET would get better, higher temperature like  $1050^{\circ}$ C, higher O<sub>2</sub> flow such as 5000 sccm, and longer oxidation time could be applied in the same experiment to find the optimum situation of oxidation. Whether dry oxidation or wet oxidation differs is also a way of research. Same experiment can be also repeated in some other SiGe film with higher Ge content. Thermal oxide instead of PE-SiO<sub>2</sub> serves as gate oxide may also bring expected improvement. As the best condition is found, fabrication of SiGe layer with high Ge concentration may be more economical and convenient through this Ge condensation method.



Alloy concentration	Oxidation condition	Variable: Temperature
	16 minutos	1000°C
Si <sub>0.89</sub> Ge <sub>0.11</sub>	$\frac{10 \text{ minutes}}{2750 \text{ scarm } O_{10} \text{ flow}}$	950°C
	$5750$ secili $O_2$ now	un-oxidized

Table 2-1(A). Experiment of the Different Oxidation Temperature.

Alloy concentration	Oxidation condition	Variable: Time
		36 minutes
Si Co	1000°C	16 minutes
S10.93 Ge0.07	3750 sccm O <sub>2</sub> flow	4 minutes
	1896	un-oxidized

Table 2-1(B). Experiment of the Different Oxidation Time.

Alloy concentration	Oxidation condition	Variable: O <sub>2</sub> Flow
	$1000^{\circ}$ C	3750 sccm
Si <sub>0.89</sub> Ge <sub>0.11</sub>	1000 C	2500 sccm
	10 minutes	un-oxidized

Table 2-1(C). Experiment of the Different Oxygen Flow.

Alloy concentration	Oxidation condition	Variable: Thickness
	$1000^{0}$ C	1000Å
Si Ca	1000 C	500Å
S10.93 Ge0.07	30  minutes	300Å
	$5750$ secili $O_2$ now	no pre-oxide

Table 2-1(D). Experiment of the Different Pre-oxide Thickness.



Alloy concentration	Oxidation condition	Variable: Rate	
		950 °C 15 minutes	
Si <sub>0.93</sub> Ge <sub>0.07</sub>	$3750 \text{ sccm O}_2 \text{ flow}$	900 °C 30 minutes	
		850 °C 75 minutes	

Table 2-1(E). Experiment of the Different Oxidation Rate.

Fixed oxidation condition	Temperature	Id Vd=(-6V)	gm(max)	On/Off ratio
Si <sub>0.89</sub> Ge <sub>0.11</sub>	1000 °C	4.58	8.02	4.14
$3750 \text{ sccm } O_2 \text{ flow}$	950 °C	6.45	11.51	6.02

Table 3-1. I-V characteristics with different oxidation temperature. Data was normalized by being divided by un-oxidized device.

Fixed oxidation condition	Time	Id Vd=(-8V)	gm(max)	On/Off ratio
Si <sub>0.93</sub> Ge <sub>0.07</sub>	36 minutes	8.32	8.21	3.33
1000°C	16 minutes	2.51	2.79	1.61
$3750 \text{ sccm O}_2 \text{ flow}$	4 minutes	P1.19	1.47	1.49

Table 3-2. I-V characteristics with different oxidation time. Data was normalized by being divided by un-oxidized device.

Fixed oxidation condition	O <sub>2</sub> flow	Id Vd=(-6V)	gm(max)	On/Off ratio
Si <sub>0.89</sub> Ge <sub>0.11</sub>	3750 sccm	6.02	11.28	5.89
16 minutes 1000°C	2500 sccm	4.31	5.64	1.86

Table 3-3. I-V characteristics with different oxygen flow. Data was normalized by being divided by un-oxidized device.

Fixed oxidation condition	Thickness of pre-oxide	Id Vd=(-8V)	gm(max)	On/Off ratio
Si Ca	1000Å	1.48	1.08	6.92
S1 <sub>0.93</sub> Ge <sub>0.07</sub> 1000°C 36 minutes 3750 sccm O <sub>2</sub> flow	500Å	1.71	2.88	5.20
	300Å	4.53	6.98	4.49
	No pre-oxide	8.92	8.81	2.74

Table 3-4. I-V characteristics with different thickness of pre-oxide. Data was normalized by being divided by un-oxidized device.



Fixed oxidation	Oxidation Rate	Pre-oxide	Id	gm(max)	On/Off
condition		thickness	Vd=(-6V)		ratio
Si <sub>0.93</sub> Ge <sub>0.07</sub> 3750 sccm O <sub>2</sub> flow	950°C 15	257Å	3.92	3.43	4.11
	minutes				
	900°C 30	261Å	3.69	3.30	6.51
	minutes				
	850°C	266Å	3.51	3.61	13.47
	75minutes				

Table 3-5. I-V characteristics with different oxidation rate. Data was normalized by being divided by un-oxidized device.



Fig. 2-1. Cross sectional view of SiGe-based P-MOSFET after mask #1.



Fig. 2-2. Top view of SiGe-based P-MOSFET after mask #1.





Fig. 2-3. Cross sectional view of SiGe-based P-MOSFET after mask #2.





Fig. 2-4. Top view of SiGe-based P-MOSFET after mask #2.



Fig. 3-1(b).  $g_m$ -Vg characteristic of different oxidation temperature.



Fig. 3-1(c). Id-Vg characteristic of different oxidation temperature.



Fig. 3-2(b).  $g_m$ -Vg characteristic after different oxidation time.



Fig. 3-2(c). Id-Vd characteristic after different oxidation time.



Fig. 3-3(b).  $g_m$ -Vg characteristic with different oxygen flow.



Fig. 3-3(c). Id-Vd characteristic with different oxygen flow.



Fig. 3-4(b).  $g_m$ -Vg characteristic with different thickness of pre-oxide.



Fig. 3-4(c). Id-Vd characteristic with different thickness of pre-oxide.



Fig. 3-5(b).  $g_m$ -Vg characteristic of different oxidation rate.



Fig. 3-5(c). Id-Vd characteristic of different oxidation rate.

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A Study of Electrical Properties of SiGe Film with Various Oxidation

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